

# TOWARDS MONOLITHICALLY PRINTED MFCS: DEVELOPMENT OF A 3D-PRINTABLE MEMBRANE ELECTRODE ASSEMBLY (MEA)

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## Abstract

Additive manufacturing (3D-printing) and microbial fuel cells (MFCs) are two rapidly growing technologies which have been previously combined to advance the development of the latter. In the same line of work, this paper reports on the fabrication of novel membrane electrode assemblies (MEAs) using materials that can be 3D printed or extruded from the EvoBot platform. Materials such as air dry terracotta, air dry Fimo™ and standard terracotta were tested against conventional cation exchange membrane (CEM) material. The MEA was fabricated by painting the materials with custom made graphite coating. The results showed that the MFCs with the printable materials outperformed those using conventional CEM. Economic analysis showed that the utilization of ceramics-based separator can reduce significantly the overall costs. These findings suggest that monolithically printed MFCs may be feasible, as printable MEAs can improve MFCs performance, and help realise mass manufacturing at lower cost.

**Keywords** – MFC, EvoBot, Membrane electrode assembly, MEA, 3D-printing

## 1. Introduction

Renewable energy production from waste using microbial fuel cell (MFC) technology is attracting growing attention as the ever-increasing worldwide energy demand is projected to increase 28% by 2040 [1]. MFCs are bio-electrochemical devices that use microorganisms as biocatalysts to convert chemical energy stored in organic matter into electrical energy, and date back to 1911 [2,3]. MFCs consist of two electrodes, a positive cathode and a negative anode, which are separated by a semi-permeable membrane or salt bridge [4]. Microorganisms are inoculated in the anodic compartment and anaerobically are capable of oxidizing the substrate. Frequently in MFC studies, anaerobic sludge effluents are used to inoculate the cells due to their very diverse microbial community presence [5], however microorganisms that can transfer electrons via metabolism to an external electrode can be also found in pond water and soil samples [6]. The electrons are released, via a process called extracellular electron transfer (EET), to the anode electrode either directly or indirectly (using mediators). The two electrodes are connected by an external circuit, which allows the flow of electrons from the anode side to the cathode side where the reduction of oxygen occurs. However, MFCs are not only considered as infinite-life biological batteries [7] but also as promising wastewater (WW) treatment solutions [8] which can complement or substitute the existing WW treatment technologies. The latter is due to the bacterial decomposition of the

41 organic matter or waste (eg. urine) in the MFC anodes which lowers the chemical oxygen  
42 demand (COD) in the effluent even by 95% in real life field applications [9].

43 Electricity production, COD removal, cost and durability of MFC systems is largely affected by  
44 and dependent on the materials used to build those units (or stacks) [10]. Anode materials  
45 have to satisfy different requisites in order to be suitable for MFCs applications. The material  
46 has to be electrically conductive, chemically and mechanically durable, low-cost and have high  
47 surface area to enhance the bacteria-electrode interface. Carbonaceous materials certainly  
48 are the suitable materials to fulfil these characteristics due to their cost efficiency and inertness  
49 towards bacteria [11–13]. Hence materials such as carbon fibre veil [11], carbon cloth [14,15],  
50 graphite felt [16,17] and carbon paper [18] have been extensively used in the MFC research.  
51 Similarly, the cathode materials must possess specific characteristics [19,20] and ideally must  
52 have high redox potential and the ability to readily capture protons in order to facilitate the  
53 oxygen reduction reaction (ORR). Materials that fulfil those requirements  
54 are carbonaceous materials and non-corrosive metals, same as those mentioned above  
55 which are used as anode electrodes. Moreover, due to the sluggish ORR reaction occurring  
56 at the cathode, a catalyst needs to be integrated within the cathode structure [21]. Due to that,  
57 platinum group metals (PGM), platinum group metals-free (PGM-free) and metal-free high  
58 surface area carbonaceous materials have been investigated in the past as catalyst [22].  
59 However, PGM are expensive and have limited lifetime, as they are prone to  
60 deactivation/poisoning reducing their activity and efficiency [23], therefore are not suitable for  
61 MFC applications. PGM-free catalysts based on transition metals instead, are more reliable  
62 and low cost compared to PGM. However, improvements are needed if they are to be  
63 employed for large scale applications [22]. Metal-free high surface area carbonaceous  
64 materials such as activated carbon have been recently heavily utilized with high durability, low  
65 cost and commercial availability at large scale [9,24]. Another similar material which has  
66 recently attracted an increasing interest due to its biocompatibility and economical nature, is  
67 the bamboo charcoal granules and tubes [25–27]. Lastly a material that plays a crucial role in  
68 high power performances is the current collector which is usually of the same nature as the  
69 electrode or can be a different material such as stainless steel mesh which provides  
70 mechanical strength and is corrosion proof [28].

71 In parallel, membrane materials are an important element within the microbial fuel cell system.  
72 Polymeric separator based on Nafion or Nafion-derived fluorinated polymers are considered  
73 the bottleneck of the MFC research and the main contributor to high cost and internal  
74 resistance [29]. Additionally, those types of membranes are prone to biofouling after long term  
75 operation (more than 60 days) [30,31] which is a result of microorganisms, microbial  
76 extracellular polymers and salts depositing on the membrane. This, along with possibly contact  
77 resistance, impacts negatively the MFC power performance (up to 37% decrease) due to the  
78 deterioration of the cation transfer which limits the charge transfer and increases the systems'  
79 internal resistance (up to 20%) [32]. Moreover, in the open-to-air configuration, the cathode is  
80 often not well integrated within the membrane, therefore the contact resistance is even higher,  
81 and output is limited. To overcome these issues, alternative MFC architectures and materials,  
82 such as ceramic based [33,34], need to be identified (or examined further) as well as ways to  
83 manufacture and integrate them. One possible design that can benefit the system is the  
84 integrated membrane electrode assembly (MEA) in which the cathode is built on the  
85 membrane itself. It was previously shown that the power output is improved by reducing the  
86 internal resistance [35].

87 MEA is the assembled system comprised of a membrane and an electrode/s attached together  
88 as one through pressing with or without heat treatment in order to minimise the distance  
89 between them. This arrangement has been inherited from traditional chemical abiotic fuel cell  
90 and showed higher power densities compared to the conventional separated membrane and  
91 electrode configurations [35]. However, only a few studies have focused on MEA influence on  
92 MFC systems and even fewer on 3D-fabrication techniques to manufacture these MEAs.

93 The present study looks at 3D printing MFCs using novel extrude-able materials that can be  
94 produced from the EvoBot platform; a Rep-Rap 3D printer turned to robot which can inoculate,  
95 maintain and print parts for MFCs [36]. The focus is on the development of cost-effective MEAs  
96 using extrude-able air-dry membranes coated with conductive paint. Different ceramic and  
97 polymeric based membranes were investigated and compared in terms of chemical  
98 composition and properties. The electrical conductivity, surface morphology and chemistry of  
99 the materials were also analysed. At last, the electrochemical performance in terms of power  
100 generation was measured and reported.

## 101 **2. Materials and Methods**

102

### 103 **2.1 Membrane Materials**

104 For the scope of this experiment, three types of potentially extrude-able membranes were  
105 tested and compared against a conventional cation exchange membrane (CEM) (Membranes  
106 International, USA). The materials tested were Fimo™ air-dry clay (Staedtler, German),  
107 terracotta air-dry clay (Hobbycraft, UK) and red terracotta modelling clay (Tiranti, UK). The  
108 latter was kilned at a temperature of 1070°C prior to use, to allow the structural bonding of the  
109 clay and ensure durability, whereas the other two were dried overnight at room temperature.  
110 All three membranes were prepared using the same process as previously described [37] and  
111 the thickness of the tested membranes was consistent for all the custom made membranes  
112 (2.5mm). The total surface area of the membranes was 25 cm<sup>2</sup>. The control (CEM) membrane  
113 required activation in 5% NaCl at 40°C for 12 hours prior to use. The images of the different  
114 membranes utilised in this study are shown in Figure 1.

### 115 **2.2 Membrane electrode assembly**

116 A conductive graphite coating was applied to each membrane and formed the cathode  
117 electrode. The coating was fabricated using polyurethane rubber coating (PlastiDip), white  
118 spirit and graphite powder as previously described [38]. The membranes were coated  
119 uniformly with the conductive cathode mixture using a brush followed by the Dr. Blade  
120 technique using a spatula. The surface resistance was measured during each coating (Section  
121 2.3.4). After the membrane electrode assembly had dried, a cable was attached to the cathode  
122 using conductive wire glue, to form the cathodic current collector.

123

## 124 **2.3 Characterization of membrane and electrode**

125

### 126 **2.3.1 Chemical composition of the membranes**

127 A qualitative chemical analysis of the membranes was determined using the Oxford  
128 Instruments Aztec energy dispersive X-ray (EDX) system and the main elemental content of  
129 each membrane sample was identified.

### 130 **2.3.2 Morphology of the electrode**

131 Surface morphology of the electrode side of the MEA was acquired through images using FEI  
132 Quanta 650 field emission scanning electron microscopy (SEM) at difference magnifications.

### 133 **2.3.3 Hardness test**

134 The hardness of the membranes materials was tested using the Vickers Hardness Testing  
135 equipment [39] (Buehler, UK). Particularly; the entire 25 cm<sup>2</sup> of the membrane was used for  
136 the hardness testing. Average values were obtained from five readings taken from five  
137 different locations on the membrane and located 5 cm apart.

### 138 **2.3.4 Electrical conductivity of the cathode multiple layers**

139 The in-plane conductivity of the cathode was measured through a handheld digital multimeter  
140 (TENMA, 72-7750). Particularly, crocodile clips were attached on the opposite sides of the  
141 membrane electrode assembly and the resistance was measured. This operation was  
142 repeated for each layer of graphite applied on the membrane. This method was only used to  
143 give an early indication of the in-plane resistance of each conductive coating addition.

## 144 **2.4 MFC architecture**

145 Four triplicates of analytical size MFCs were assembled for this experiment. Each MFC  
146 consisted of a single chamber with an empty volume of 25 mL in which the anode was inserted.  
147 The MFCs were constructed using one side of a covered 15mm thick Polymethyl methacrylate  
148 (Perspex) sheet. The anode electrode was a folded carbon veil fibre (20gm<sup>-2</sup> carbon loading)  
149 with a projected area of 8.45 cm<sup>2</sup> and total surface area of 270 cm<sup>2</sup> (PMF Composites, Dorset,  
150 U.K.). The cathode was integrated with the four different membranes and directly glued to the  
151 anode chamber with an inert aquatic sealant (Aquabits, UK). In order to maintain the moisture  
152 of the membrane electrode assemblies and maintain a liquid 'bridge' for proton transport, the  
153 MFCs were wrapped with Parafilm® which is a highly waterproof material but at the same time  
154 is permeable to oxygen (Figure 1.A).

## 155 **2.5 Inoculation and feedstock**

156 The twelve MFCs were inoculated with a mixture of 50% fresh human urine collected  
157 anonymously from healthy individuals and 50% anolyte derived from already established  
158 experiment operating on activated sludge (Wessex Water, UK) and urine. The mixture was  
159 enriched with a background solution of TYE which was comprised of tryptone (1%) and yeast  
160 extract (0.5%) as nutrient and amino acid supplier. The solution was left in a shaking incubator  
161 (Orbital Incubator S150) for 24 hours at a shaking speed of 130 rpm and at a temperature of  
162 36.6°C. The solution was transferred to 15 mL centrifuge tubes (Corning, UK) and placed into  
163 the centrifuge (VWR Compact Star CS4) for 10 minutes at 5000rpm. Subsequently, the  
164 supernatant was removed and the pellet re-suspended into 5 mL of neat urine. The re-  
165 suspended medium was collected and formed the inoculum for the experiment. After  
166 inoculation period when the biofilm formed, the MFCs were fed manually in batch mode with

167 urine. The experiment was performed at room temperature ( $22 \pm 2$  °C) within a temperature-  
168 controlled environment.

## 169 **2.6 External resistance and polarization experiment**

170 The MFCs operated in a fixed load of 2.7 k $\Omega$  prior to polarization experiment, which was  
171 conducted on the MFCs by connecting them to 8-channel automated *Resistorstat* equipment,  
172 developed as described in Degrenne *et.al.* [40]. The external resistance ( $R_{ext.}$ ) was ranging  
173 decreasingly from 30 k $\Omega$  to 3.74  $\Omega$  with each resistance value held for 3 minutes. During  
174 polarization, voltage output was recorded every 30 s (6 samples per resistance value) to make  
175 it possible to monitor and capture the dynamic response of MFC to changes in  $R_{ext.}$ . The MFCs  
176 were kept in open-circuit voltage for 2 hours prior to polarization testing. Following the first  
177 polarization experiment the  $R_{ext}$  was changed to 1 k $\Omega$ , a value at which maximum power was  
178 generated. This load was kept constant until the end of the experiment.

## 179 **2.7 Data recording and calculations**

180 The MFCs were individually connected to an Agilent Keysight 34970A Data Acquisition / Data  
181 Logger Switch Unit (Keysight Technologies, UK) for data recording of voltage (mV) against  
182 time with readings taken every 3 minutes. The data were processed and analysed using MS  
183 Office Excel where current (I) in amperes (A) was calculated in accordance with the Ohm's  
184 Law;  $I = V/R$ , where V is the measured voltage and R is the known value of the external  
185 resistive load in ohms ( $\Omega$ ). Power (P) values in watts (W) were calculated by multiplying  
186 voltage with current:  $P = I \times V$  [41].

## 187 **2.8 COD measurements**

188 Chemical oxygen demand (COD) removal analysis was carried out on the forty-second day of  
189 the experiment by analysing the influent sample and comparing it with the effluent sample  
190 after 24 hour retention time in the MFC. The analysis were conducted using the potassium  
191 dichromate oxidation method (CamLab, UK) with high range (HR) COD vials in which 0.2 mL  
192 of sample to be analysed (inlet and outlet) was taken from the MFC and filtered before being  
193 added into the vial. The sample was then heated up at 150°C for 2 hours and cooled down for  
194 another 2 hours. At last, the concentration was measured through a spectrophotometer  
195 (Lovibond Water testing).

## 196 **2.9 Breakdown of the experimental procedure**

197 The experimental plan that was followed in this study has been summarised and presented  
198 at Table 1. The table notes in detail the feeding regime followed, the operational conditions,  
199 urine replacements, timings and quantities of feedstock as well.

*Breakdown of the experimental scheme*

Day 0	Inoculation	100% inoculum	Full exchange (25 mL)
Day 1-2	Inoculation	50% inoculum – 50% urine	Full exchange (25 mL)
Day 3-4	-	No feeding	-
Day 5-9	Daily Feeding	100% urine	Full exchange (25 mL)
Day 10-14	Daily feeding	100% urine	5 mL top-up
<i>Day 15 onwards feeding regime: a complete exchange of anolyte early each week, followed by a daily top-up of 5 mL until the end of the week.</i>			
Day 15-25	Daily feeding	Water diluted urine	Feeding regime described above
Day 26-30	Daily feeding	100% urine	Feeding regime described above
Day 30	Polarization	Change of Rext. from 2.7 kΩ to 1 kΩ	
Day 31-58	Daily feeding	100% urine	Feeding regime described above
Day 59-62	Starvation period	No feeding	-
Day 63-73	Daily feeding	100% urine	Feeding regime described above

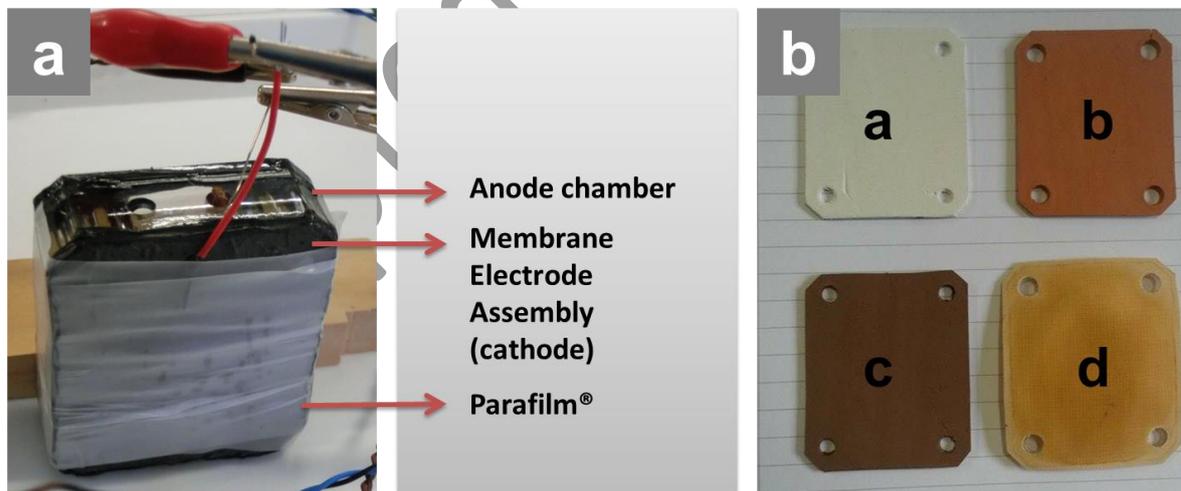
**Table 1. Breakdown of the experimental scheme followed through the entire duration of the experiment.**

200

### 201 3. Results and Discussion

202

#### 203 3.1 Material selection



**Figure 1. Photographs of a) the MFC set-up and b) the four cut-to-shape membrane materials** Air-dry Fimo™ (a), Air-dry Clay (b), Terracotta (c) and Cation exchange membrane (d).

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Four different membranes (Figure 1.B) were selected and investigated as separators in MFCs fed with urine. Two of the membranes were based on air-dry techniques and were Fimo™ air-dry clay and terracotta air-dry clay. These two materials were selected due to the advantage of being extrude-able from an adapted 3D-printer nozzle, which can be incorporated in the EvoBot platform. Furthermore, the air-drying technique allows fabricating ceramics just

209 through normal atmospheric conditions without the utilization of heat treatment. Red terracotta  
210 modelling clay was another membrane used for this experiment and acted as a control to the  
211 investigation. However terracotta was subject to high temperature treatment in controlled  
212 atmospheric conditions which allowed the internal binding of the clay within the structure as  
213 needed [37]. The last membrane utilized was a commercial polymeric-based cation exchange  
214 membrane.

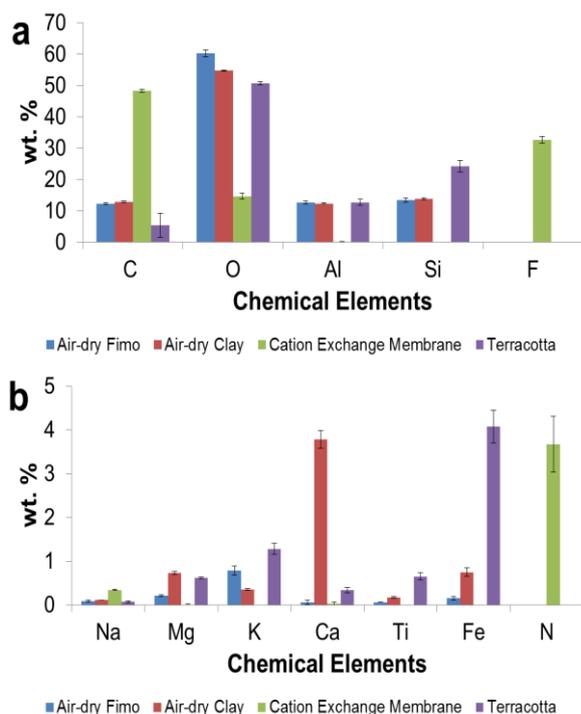
## 215 **3.2 Material analysis**

### 216 **3.2.1 Chemical composition of the membranes**

217 Energy dispersive X-ray (EDX) system was used for the qualitative chemical analysis of the  
218 elements composing each membrane tested during this investigation. C, O, Al, Si and F were  
219 the elements identified with percentages above 10% (Figure 2.A). Carbon and oxygen were  
220 detected in all four samples, however it is notable that aluminium and silicon were detected  
221 only for the ceramic-based membranes. These elements are well known to be generally  
222 integrated within ceramic materials especially in their oxide form [42]. As expected fluoride  
223 was only detected in the polymeric membranes. Generally, polymeric membranes are  
224 composed by a backbone of fluorinated polymer that gives mechanical strength and resistance  
225 to harsh and corrosive environments. Unfortunately, F is not environmentally friendly and  
226 therefore the utilization of fluorinated materials in microbial fuel cells might negatively impact  
227 the environment after long terms operations. In fact, if those MFCs are to be used on-board  
228 low-power robots that are programmed to perform a particular task and then degrade naturally  
229 in the environment [43] then the use of fluorinated membranes needs to be avoided.

230 Additionally, elements with percentage lower than 5% were also reported (Figure 2.B). Traces  
231 of Na, Mg, K, Ca, Ti and Fe were detected within the ceramic-based samples. Only Na and N  
232 were detected on the polymeric membrane. Interestingly, a percentage of ~3.5% of Ca was  
233 measured in the air-dry clay and higher content (roughly 4%) of iron was detected in the  
234 terracotta sample.

235

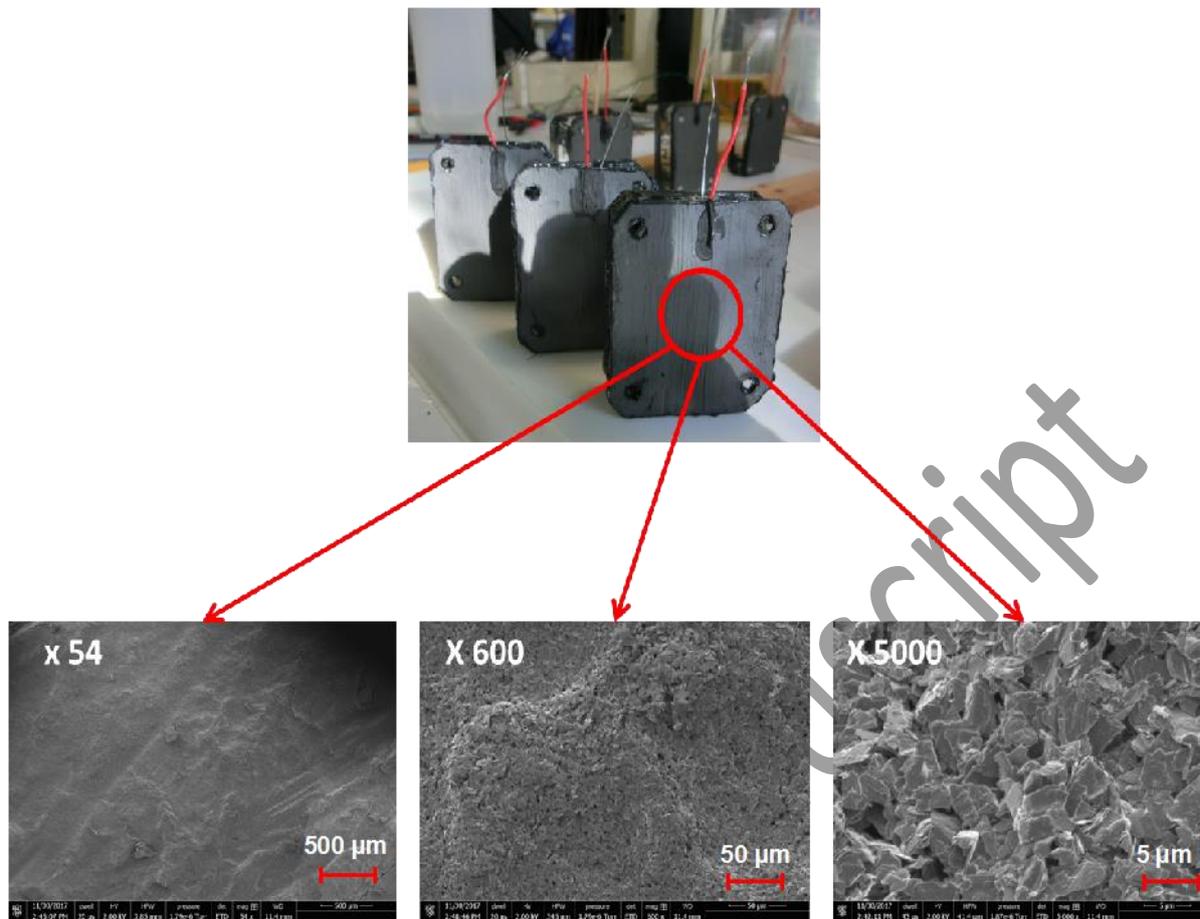


**Figure 2. ED-X analysis results data of the chemical elements between the four tested types of membranes. a) Major and minor components b) trace elements.**

236

### 237 3.2.2 Morphology of the electrode

238 The morphology of the graphite-based coating was observed using SEM images at different  
 239 magnifications (Figure 3). Increasing magnifications allowed visualising the surface of the  
 240 electrode in more detail. The surface of the electrode in fact seems to be fully covered by a  
 241 quasi-uniform coating. At higher magnification, the graphite particles could be clearly detected  
 242 with non-uniform shape and length within the micrometric shape in agreement with the  
 243 manufacturer's specifications [44].



**Figure 3. SEM images at three different magnification of the MEA side surface coated with the graphite ink (cathode side).**

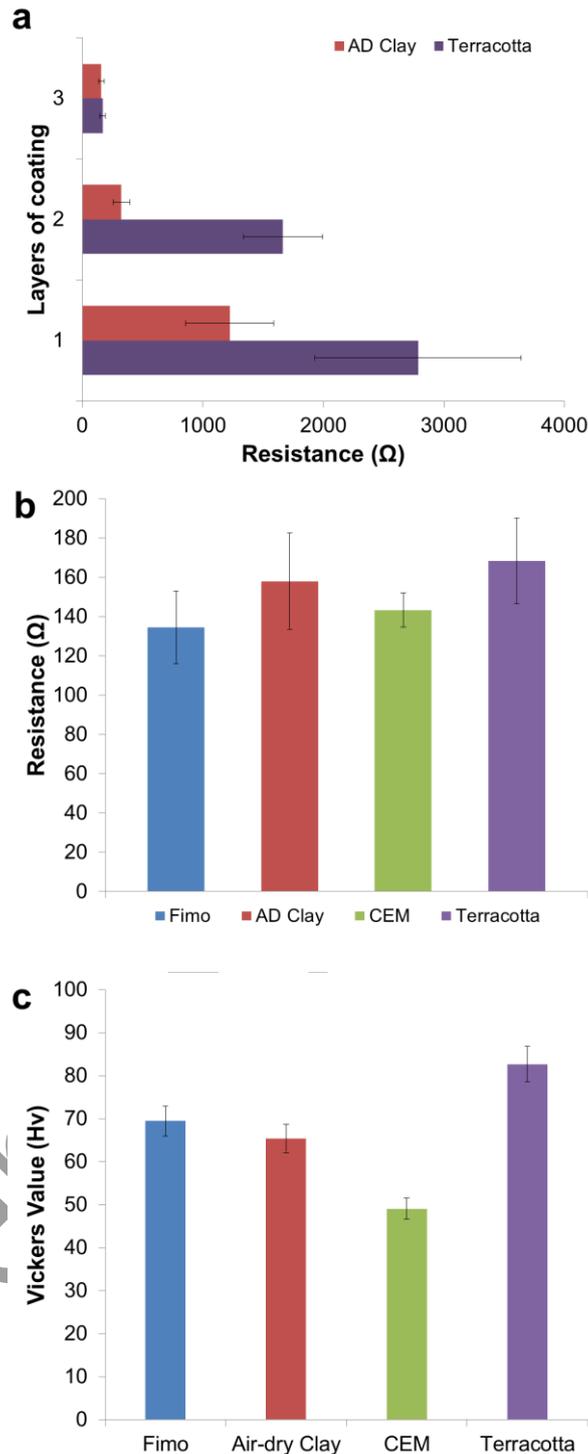
### 244 3.2.3 Electrical conductivity of the cathode multiple layers

245 The MEAs were prepared by applying the conductive ink directly on to the already set  
 246 membranes using a layer-by-layer technique akin to a 3D-printer approach. Each layer was  
 247 left to air-dry before each in-plane measurement of resistance was taken. Figure 4.A. shows  
 248 the in-plane resistance of the air-dry clay and terracotta clay based MEAs after applying the  
 249 first, second and third layer of conductive ink on their surfaces. Those two sets of data were  
 250 chosen to be discussed as they illustrate nicely the initial difference in resistance between the  
 251 two differently made clays (air dry and kilned). Initially the terracotta MEA had almost 2.5 x  
 252 higher resistance compared to air-dry clay. The difference in resistance became larger after  
 253 applying the second layer of coating with terracotta being 5.0 x higher than air-dry clay even  
 254 though the overall resistance decreased for both by nearly 2.0 x and 4.0x respectively.  
 255 Despite the differences and high resistance values initially, by the time the third layer of  
 256 conductive coating was applied and cured, both MEAs showed similar in-plane resistance  
 257 ( $170 \pm 5 \Omega$ ). This suggested that by that time the conductive ink coating covered completely the  
 258 surface of the membrane and bonded with the underlined layers of coating in a quasi-uniform  
 259 manner, in agreement with the SEM micrographs. It is noteworthy that in an attempt to  
 260 decrease the in-plane resistance even more, a fourth layer of coating was applied on the  
 261 membranes however this had an adverse effect on the continuity of the electrode as it caused  
 262 cracking of the upper layer of coating (data not shown). Thus for the scope of this experiment  
 263 only three layers of coating were applied on each membrane and form the MEA, which was in

264 accordance to what has been reported in the literature for similar conductive inks applied on  
265 paper-based MFCs [45]. Following the application of three consecutive layers of conductive  
266 ink on the MEAs the in-plane resistance values were measured and plotted on Figure 4.B  
267 showing that all MEAs had an overall resistance of  $150 \pm 15 \Omega$ . More specifically, Fimo™ had  
268 an in-plane resistance of  $135 \Omega$  followed by CEM with a resistance of  $143 \Omega$ , air-dry clay and  
269 terracotta as mentioned above had a similar resistance of 158 and 168 ohms, respectively.  
270 These results have given an initial indication of the in-plane resistance after each layer of  
271 graphite coating, which can provide information on the current connecting losses of the MEA.  
272 In further investigation of MEA materials, through-plane resistance it is recommended to be  
273 measured in order to provide more comprehensive results on the whole MEA resistance.

274

Accepted manuscript



**Figure 4. Resistivity of the MEA and hardness of the membrane.** a) average resistance values after each coating on the membrane b) average surface resistance of the MEA electrode c) hardness test based on Vickers value (Hv).

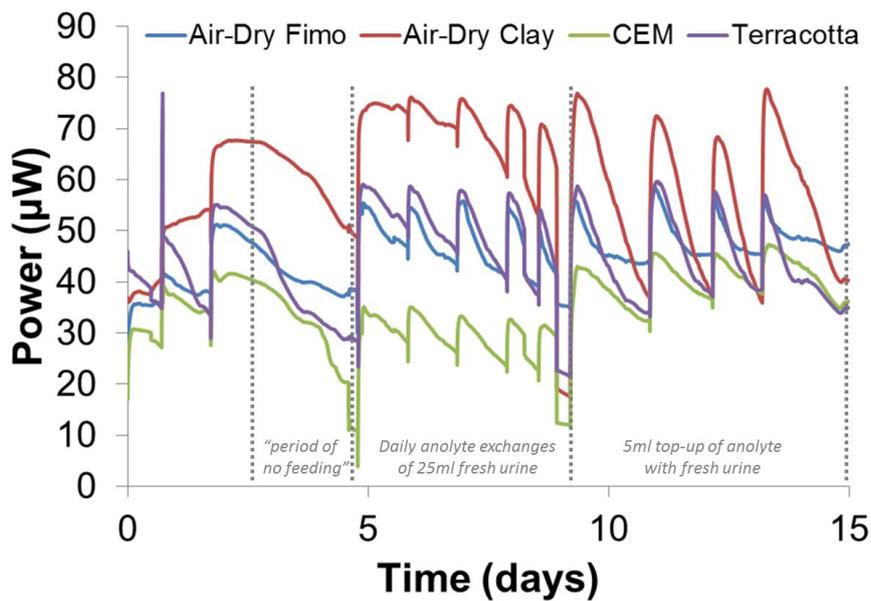
275 It is well known from the literature that ceramics possess many unique characteristics that  
 276 make them suitable for use within MFC systems; one of these advantageous characteristics  
 277 is their structural durability [33]. In order to test the durability of the materials under  
 278 investigation against terracotta, a Vickers hardness test was performed. Even though this  
 279 technique is well used for testing metal materials, it has been reported in the past that it can  
 280 be used for ceramic materials [39] to give an indication of durability between different samples.

281 The unit of hardness given by this test is the Vickers Pyramid Number (HV), the five values  
282 obtained from each ceramic testing were averaged and presented in Figure 4.C. The results  
283 confirmed that terracotta was indeed the most durable/hardest (82.7 HV); however air-dry  
284 Fimo™ and air-dry clay were only 13 and 17 HV units more fragile, respectively. To put the  
285 results into perspective a diamond has a hardness value of 10000 HV. Those data are an  
286 early indication that even though air-dry clays are not as structurally robust as kilned terracotta,  
287 they can still be proven to be durable. The hardness of the materials was tested in order to  
288 observe the properties of the material in question, in terms of deformation from a standard  
289 source (the metal indenter), which would in turn provide an indication of the material's ability  
290 to resist wear, pressure, or damage, which is particularly relevant for shipping systems like  
291 these to other areas.

### 292 3.3 Power Output

#### 293 3.3.1 Initial power output profile of the first fifteen days

294 For this experiment, MFCs using different membrane materials were tested with the ultimate  
295 aim to observe the feasibility of using air-dry clays that can be 3D-printed, as separators for  
296 MFCs. As previously mentioned (Section 2.5), the MFCs were all inoculated with a mixture of  
297 activated sludge, tryptone yeast extract and effluent from established urine fed MFC  
298 experiments. After inoculation, the MFCs were left in open circuit for three hours until the  
299 voltage plateaued. The observed potential difference between the anode and cathode from all  
300 the MFCs was roughly  $600 \pm 50$  mV (data not shown). An external load of  $2.7 \text{ k}\Omega$  was connected  
301 on all MFCs, closing the circuit and initiating the power generation process by encouraging  
302 the formation of an electroactive biofilm on the anode electrode. The MFCs were maintained  
303 in a batch fed mode. Each feeding can be simply identified on the graph as every time a  
304 feeding occurred, an instant increase -followed by gradual decrease- in power output can be  
305 observed. This is reflected by the fact that the energy source availability within the anode  
306 chamber increased, meaning that bacteria have fresh and available organic matter to  
307 metabolise. The first two exchanges in anolyte consisted of replenishing fully the chamber with  
308 the aforementioned inoculum and urine in a 50%:50% ratio, which was sufficient to supply the  
309 bacteria with the much needed carbon energy sources to continue their metabolic activities  
310 during "periods of no feeding", which were beyond the normal feeding cycle (Figure 5). During  
311 this period, the MFCs with air-dry clay membrane decreased in performance by 13.2%, which  
312 was similar with the air-dry Fimo™ (13.8% decrease). The MFCs with terracotta and CEM  
313 were the most affected showing a decrease of 18.6% and 40.0% respectively.



**Figure 5 Power generation recorded in the first 15 days of operations.**

314

315 On the fifth day, the anode chamber of each MFC was replenished completely (25 mL) with  
 316 fresh urine that resulted in increasing the power output by 14.4% and 14.8% on the air-dry  
 317 clay based MFCs, 29.7% for the CEM based MFCs and nearly 50% in the terracotta based  
 318 MFCs (Figure 5). It is assumed that the terracotta based MEA had the greatest improvement  
 319 in power following carbon-energy depletion, because of the higher porosity compared to the  
 320 other materials, which has a direct impact on the cation rate of exchange. Higher porosity, and  
 321 non-selective materials will naturally allow a higher rate of cations to diffuse through – a  
 322 process which is driven by electron-neutrality. This means that for selective or lower porosity  
 323 materials even if the rate of electrons generated by the biofilm communities is at similar levels  
 324 for all tested conditions, the power output will be lower, as a result of the lower number of  
 325 cations, diffusing through the membrane and reacting with the incoming electrons. This is also  
 326 reflected by the fact that the terracotta based MEAs reached maximum power output just after  
 327 5 days of operation[42] whereas the other materials needed more time to reach that. The next  
 328 five days continued with daily anolyte exchanges of 25 mL fresh urine as indicated by the  
 329 spikes on the graph. During this period, the MFCs showed consistent increase in power output  
 330 with a peak at 75 µW for air-dry clay followed by terracotta and air-dry Fimo™, which were  
 331 almost on a par at 58 µW and 54 µW, respectively; the CEM was the least performing with 37  
 332 µW. Nine days after the start of the experiment and the daily complete emptying and refilling  
 333 of the anode chambers with fresh organic matter, the mode of feeding was switched to 5 mL  
 334 top-ups, after manually removing the same liquid volume from each anode. This had as an  
 335 impact the overall increase in power output of the MFCs by 11%-15%, however performance  
 336 would decrease much faster due to the lower amount of fresh carbon-energy available.  
 337 Bacteria were presumably still consuming other by-products available in the suspension -  
 338 within the 24 hour window between each feeding- however the power output at the end of  
 339 each feeding cycle was consistently at the same level of 40 µW each time. In order to  
 340 compensate for the impact of each feeding approach on the power performance, it was

341 decided to have a complete exchange of anolyte early each week, followed by a daily top-up  
342 of 5 mL until the end of the week. This strategy was followed until the end of the experiment.

### 343 3.3.2 Long term power output profile of the entire experiment duration

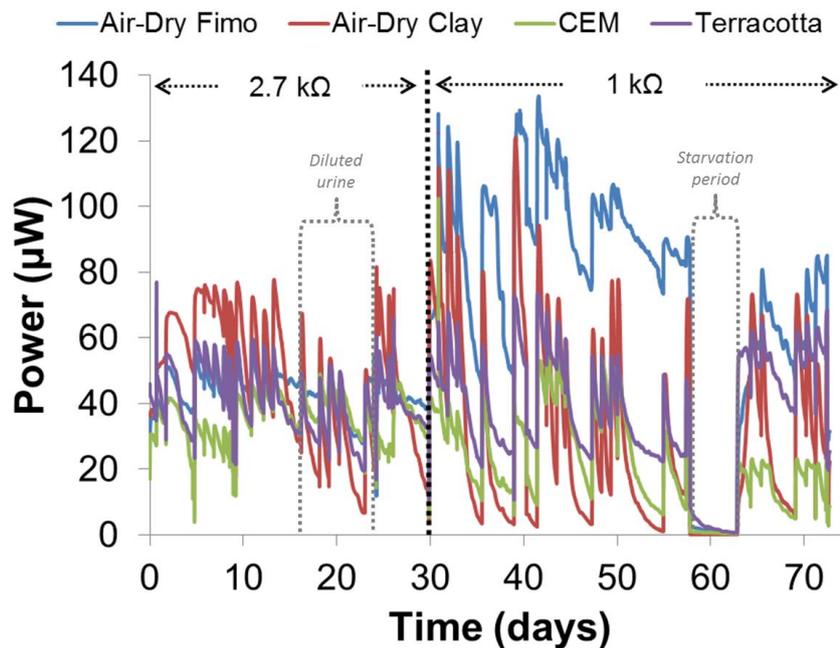


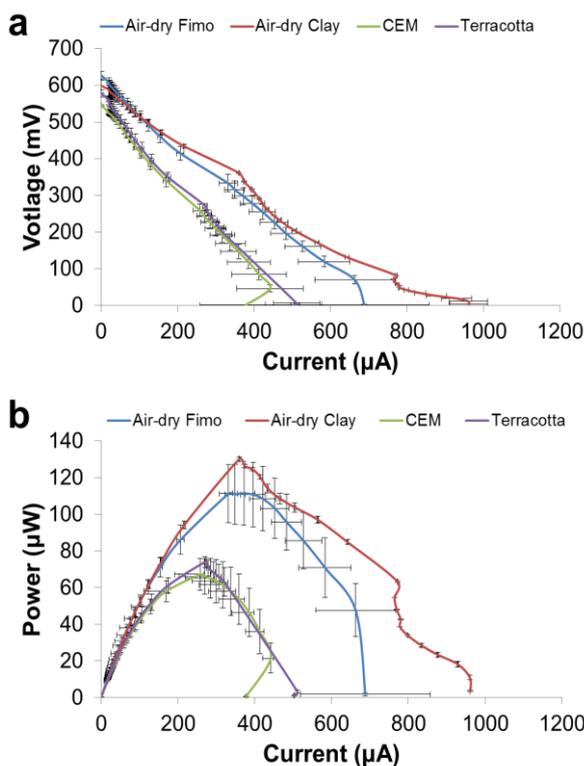
Figure 6. Power generation over 70 days operations

344  
345 As explained earlier, the experiment was maintained at a constant external load of 2.7 kΩ for  
346 the initial 30 days of operation, during which it was clearly observed that the air-dry clay was  
347 the best performing, while the air-dry Fimo™ and terracotta were on a par; in most cases CEM  
348 was the least performing. Although a steady-state was achieved within the first fifteen days of  
349 operation for all the MFCs, this was lost due to urine shortage and having to use diluted urine  
350 (50:50) between the fifteenth and twenty-fourth day of the experiment (Figure 6). It is  
351 envisaged that the steady state was going to be continued at the same levels if neat urine had  
352 been supplied to the bacteria. This hypothesis is confirmed by the data of the twenty-fifth day  
353 where the output of the MFCs recovered to the previous levels once un-diluted neat urine was  
354 supplied. Following a month of operation at a constant load, the MFCs were subjected to  
355 polarization using an automated Resistorstat as explained previously (Section 2.6) in order to  
356 identify the optimum resistance value based on each system that can give the maximum power  
357 output. Although the polarization results are discussed in detail below (Section 3.2), the impact  
358 of identifying and applying the optimum external load on the MFCs will be discussed here. The  
359 results of the polarization experiment showed that those particular MFC systems can operate  
360 at their best when they are subjected to an external loading of 1 kΩ resistance. Once the  
361 external resistance switched to 1 kΩ, Fimo™ outperformed the rest and the overall power  
362 output of all others also increased by 25%-50%. The performance of the MFCs was  
363 maintained at the same levels for the following month until a complete starvation of four  
364 consecutive days brought all the systems to nearly zero (from day 58 to day 62). During this  
365 period, all the anode MFC chambers completely dried out from evaporation (Figure 6).  
366 However, once the MFCs were fed again, the bacterial communities of the already established  
367 anode biofilm switched from inactive mode (carbon limited) to active mode. Therefore, the  
368 power performance recovered immediately back to similar levels as those from the last

369 feeding. More specifically, Fimo™ separated MFCs reached 91.25% recovery, air-dry clay  
 370 and CEM reached full recovery (100%), while interestingly terracotta separated MFCs had an  
 371 increase of 13.4%. However compared to the highest levels of performance recorded during  
 372 days 30-45, the percentage of recovery was 63.5%, 64%, 40.5% and 91% for Fimo™, air-dry  
 373 clay, CEM and terracotta respectively. The recovery profile of the previously dried and inactive  
 374 MFCs adds an extra value to the feasibility of those systems. MFCs are biological entities that  
 375 not only have long-term power production capabilities -for as long as organic matter is  
 376 supplied- but more importantly, can survive elongated periods of starvation with demonstrable  
 377 fast response/recovery. In addition, inexpensive materials such as air-dry clays and terracotta  
 378 can be used to make these MFCs, which is an economic advantage over traditional  
 379 commercially available ion exchange membranes.

380

### 381 3.3.3 Polarization results



**Figure 7. Polarization results** a) voltage vs current curve and b) power curve. Error bars are shown for all conditions

382

383 After a whole month of operation under a constant load (2.7 kΩ), all the MFCs were subjected  
 384 to polarization analysis by sweeping the external resistance value in a gradual manner starting  
 385 from infinite resistance (open circuit) and finishing at a very low resistance value (heavy load).  
 386 Throughout the polarization experiment, the voltage output of the cells was recorded as a  
 387 function of resistance, which made it possible for the automated system to calculate the  
 388 current and subsequently the power output. Using the aforementioned data a polarization  
 389 curve (Figure 7.A) was generated by plotting the voltage versus current. The recordings from  
 390 each triplicate of MFCs were averaged and plotted including the standard deviations. The  
 391 initial values at 1MΩ resistance (no current) show the open circuit voltage (OCV) of all the  
 392 cells at around 600±50 mV. The OCV of the tested MFCs was around 500 mV below the

393 theoretical OCV value (1.1 V) for open-to-air cathode MFCs primarily due to activation  
394 overpotentials, which is a characteristic of MFCs with air-breathing cathodes, which operate  
395 on the oxygen reduction reaction (ORR); this is limiting when occurring in neutral media [22].  
396 These OCV values are in agreement with other open-to-air MFC devices under similar  
397 operating conditions [13,46]. Although in the literature there are also higher OCV values  
398 reported (0.7-1.0 V), the cathode electrodes of those MFCs were either supplemented with  
399 ferricyanide, during the polarization experiment, or were moistened continuously with tap  
400 water [47–49].

401 The scope of the polarization experiment is to understand better the specific characteristics of  
402 the systems under examination in order to measure their power outputs. Figure 7.B shows the  
403 power curves generated, this graph gives us the possibility to assess the maximum power  
404 transfer point (MPP), which is the maximum peak of each power curve and corresponds to the  
405 optimum resistance value that can give this output. Based on the results, it is evident that air-  
406 dry clay had the highest power output at 130  $\mu\text{W}$  followed by air-dry Fimo™ with 111  $\mu\text{W}$ .  
407 These results were in accordance with the real time data of the initial thirty days shown above  
408 (Section 3.3.2) proving that the air-dry membranes generated the highest amount of power  
409 output. The two underperforming MFCs were the control ones with terracotta (73  $\mu\text{W}$ ) and  
410 CEM giving 50% less than the air-dry clay (66  $\mu\text{W}$ ).

411 Comparing the power density of the MFCs reported herein with other MEA-based MFCs in the  
412 literature, the 3D-printable ones are showing promising results. In particular a study on tubular  
413 MFCs with air breathing cathodes based on CEM-MEA had a power density of  $5\text{W m}^{-3}$  (based  
414 on the anodic liquid volume of 200 mL) [50]. This output is  $0.2\text{W m}^{-3}$  less than the air-dry clay-  
415 MEA of this study, calculated based on a reactor volume of 25 mL. Additionally apart from the  
416 advantage over the power density, the MEA used in the aforementioned study [49] was  
417 fabricated using carbon cloth coated with a mixture of Pt powder and carbon black bonded  
418 together with Nafion resin, which inherently increases the cost.

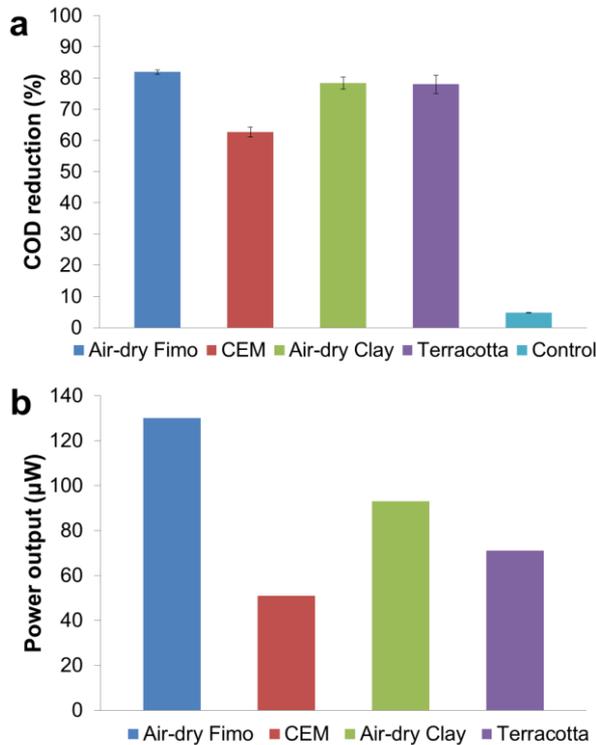
419 Following polarization analysis, the optimal external resistance was identified (1 k $\Omega$ ). Once the  
420 MFCs were connected to this lower resistance value, the performance levels begun to diverge  
421 and Fimo™ was producing the highest power output as can be seen in Figure 6. Although  
422 after the change in external resistance, the results of the polarization differed from the real-  
423 time data (day 30 onwards) and Fimo™ ended up outperforming the air-dry, in all cases the  
424 soft materials were operating better than the conventional cation exchange membrane.

### 425 3.4 COD Analysis

426 Following the increase in performance due to the external loading shift, a chemical oxygen  
427 demand (COD) analysis was conducted to observe the rate of COD decrease within a 24 hour  
428 period. On the eleventh day following the switch in external resistance (forty-second day of  
429 the entire experiment), prior to replenishing the anode chamber with the 25 mL of urine a  
430 sample was taken for COD analysis. In parallel to that, a sample from that urine (25 mL) was  
431 kept in a closed glass bottle on the bench to observe the decrease in COD without being  
432 treated in MFC. The following day, a sample of the effluent of all the MFCs was taken and  
433 analysed. The results of this analysis are presented in Figure 8.A. which shows that MFCs  
434 with air-dry Fimo™ MEA had a decrease of almost  $82\pm 1\%$  in COD which was 4% higher than  
435 air-dry clay and terracotta. CEM based MFCs resulted in  $63\pm 1\%$  COD decrease. The control  
436 COD reduction occurring in the closed glass bottle after 24 hours was 4.7%; this value was

437 deducted from the overall percentage decreases of all MFCs. This was in order to demonstrate  
438 the decrease in COD, which was induced due to the bioelectrocatalytic activity of the  
439 electroactive microorganisms presented in the anode and also by fermentative floating  
440 microorganisms.

441



**Figure 8. COD reduction results and power output at the time of sampling.** a) Percentage of COD reduction of fresh urine within 24 hours in an MFC and in a closed glass bottle (control). b). Power Output of the MFCs at the moment that the samples for COD analysis were taken.

442

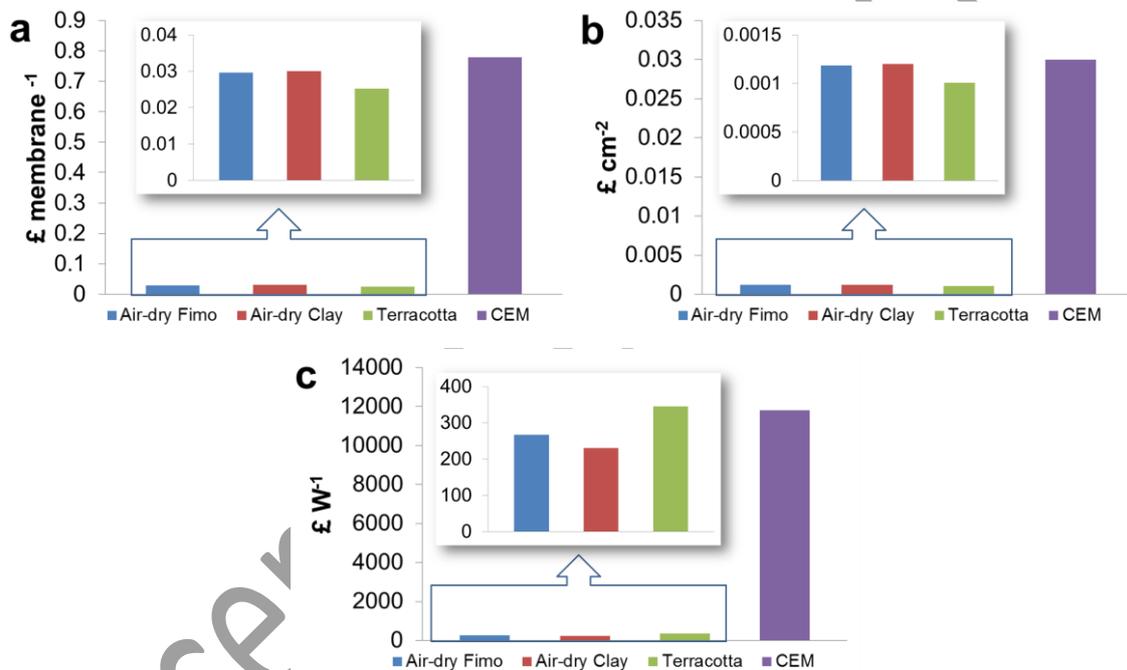
443 The results of the COD analysis are in agreement with the real time data of that period (day  
444 42), which confirm and support the literature which reports that the highest the power output  
445 the highest the COD removal [51]. For clarity Figure 8.B shows the power output of the MFCs  
446 at the time that the samples were taken. Based on that in terms of power output Fimo™ was  
447 at 130 µW, air-dry clay and terracotta performed 29% and 45% lower than Fimo™ whereas  
448 CEM had a 60% less power output than the aforementioned (51 µW). Figure 8 demonstrates  
449 clearly the correlation of power output to COD reduction and adds an extra value to the feasibility  
450 of those MEAs as alternative conductive separators of MFCs.

451 The capability of MFCs in treating wastewater and pollutants [52–54] while generating  
452 electricity is one of the attractive characteristics of the technology that make it a favourable  
453 off-grid source of electricity and sanitation.

### 454 3.5 Cost Analysis

455 As microbial fuel cell is a technology producing low quantity of electricity, particular attention  
456 needs to be given to the overall cost needed in maximizing the performance while minimizing

457 the costs. In this section, the cost of the membranes is illustrated and discussed (Figure 9).  
 458 The cost of the ceramic membranes was calculated considering 1 kilo of raw materials. More  
 459 specifically, at the moment of purchase air-dry Fimo™ cost 4.94 £ kg<sup>-1</sup>, air-dry clay cost 3.75  
 460 £ kg<sup>-1</sup> while terracotta was the least expensive of all as it was 33% cheaper than air-dry clay  
 461 and roughly 50% cheaper than air-dry Fimo™. The cost of the latter was in fact 2.52 £ kg<sup>-1</sup>.  
 462 The CEM cost was considered to be 188 £ m<sup>-2</sup>(250\$ m<sup>-2</sup>) according to the supplier. Each  
 463 membrane was weighed during fabrication and therefore the composition was known. In the  
 464 case of CEM, a total area of 25 cm<sup>-2</sup> was used. In order to fabricate the ceramic membranes,  
 465 6 g, 8 g and 10 g of air-dry Fimo™, air-dry clay and terracotta respectively were used.  
 466 Consequently, the overall cost for each membrane (with an area of 25 cm<sup>-2</sup>) was similar for  
 467 air-dry Fimo™ and air-dry clay at £0.030 per membrane and terracotta being slightly lower at  
 468 £0.025 per membrane mainly due to the lower cost of the raw material. CEM was the most  
 469 expensive costing £0.78 per membrane (Figure 9.A). This means that it is possible to fabricate  
 470 26 air-dry Fimo™/air-dry clay membranes or 31 terracotta membranes for every single CEM  
 471 membrane for the same cost.



**Figure 9. Cost analysis of the membranes.** a) Overall cost of the membranes, b) cost of the membrane for unit of area c) and cost of the membrane for unit of power produced.

472 Another important aspect to consider is the cost of the membrane density meaning the cost of  
 473 the membrane per cm<sup>2</sup>. The results are shown in Figure 9.B. The trend reflects the cost of the  
 474 raw materials with terracotta having the lowest cost per surface area and CEM the highest.  
 475 Particularly, the cost for terracotta was roughly 0.1 pence per cm<sup>2</sup> (10£ cm<sup>-2</sup>) followed by air-  
 476 dry Fimo™ and air-dry clay with a cost of roughly 0.12 pence per cm<sup>2</sup> (12 £ m<sup>-2</sup>) and CEM  
 477 thirty times higher than that (≈300 £ m<sup>-2</sup>).

478 Finally, the cost of membrane for each Watt produced was also calculated (Figure 9.C). The  
 479 power from the peak of power curves (Figure 7.B) was considered in 130 µW, 111 µW, 73 µW  
 480 and 60 µW for air-dry clay, air-dry Fimo™, terracotta and CEM respectively. Due to the higher  
 481 performance, air-dry clay had the higher value of cost per power produced among the  
 482 materials investigated that was quantified in 230 £ W<sup>-1</sup> (Figure 9.C). Slightly higher value was

483 achieved by air-dry Fimo™ with 267 £ W<sup>-1</sup>. Terracotta had a cost per W produced of 345 £ W<sup>-1</sup>  
484 and CEM had an astonishing value of 11818 £ W<sup>-1</sup>. The cost per W produced of CEM was  
485 34, 44, 51 times higher compared to terracotta, air-dry Fimo™ and air-dry clay respectively.  
486 Among the three clay-based membranes, the most cost-effective material tested was air-dry  
487 clay. The cost per W produced was 14% and 33% better compared to air-dry Fimo™ and  
488 terracotta respectively. Air-dry Fimo™, air-dry clay and terracotta seems to be very promising  
489 cost-effective membranes materials that can replace the more expensive and not  
490 environmentally-friendly polymeric membranes. It must be noted that the cost of kilning or  
491 shipping from abroad were not taken into account for these economic calculations.

### 492 **3.6 3D-printing MEAs using EvoBot.**

493 EvoBot (Figure 10.A) is a RepRap open-source 3D-printer modified successfully to operate  
494 like a robot for culturing and maintaining Microbial Fuel Cells (MFCs) based on an established  
495 feedback loop between the MFC systems and the python controlled platform [55]. EvoBot can  
496 host a large number of MFCs (24) on its experimental arena where it can gather data about  
497 the MFCs and react through a feedback loop to set thresholds (voltage output). This improves  
498 the performance of the MFCs by providing nutrient supply only when needed. So far this type  
499 of liquid handling robot-MFC interaction, based on the feedback loop mechanism, has enabled  
500 the study the adaptability and stability of those systems [55].

501 Since EvoBot is a 3D-printer turned to robot it still holds its 3D-printing capabilities thus it can  
502 extrude (3D-print) parts for MFCs with the ultimate aim to monolithically print and nurture those  
503 already made MFCs. Having this in mind, the two air-dry materials tested as alternative  
504 membranes -against conventional polymeric cation exchange membranes and fired  
505 terracotta- come in the form of soft modelling clay, which makes them suitable for extrusion  
506 from the EvoBot platform (Figure 10.B). As the uncured form of the electrode material is also  
507 fluid, it can be applied using EvoBot by incorporating a brush/roller on the actuation layer of  
508 the robot. This will apply the conductive coating onto the dried extruded membranes.



**Figure 10. EvoBot Rep-Rap 3D-printer.** a) EvoBot within its Evo-world enclosure performing experiments on MFCs under controlled conditions b) EvoBot with an adapted extruder 3D-prints Fimo™ membranes. Source: [www.evobliss-project.eu](http://www.evobliss-project.eu).

509

### 510 3.7 Conclusions

511 This study explored the idea of using air-dry based materials such as Fimo™ and clay to  
 512 fabricate membrane electrode assemblies for Microbial Fuel Cells (MFC) which were then  
 513 compared against two of the most popularly used materials in the field, cation exchange  
 514 membrane (CEM) and kilned red terracotta. Through this study it has been successfully shown  
 515 that MFCs with air-dry clay based MEAs can produce up to 50% more power than the controls.  
 516 Since the MFC technology relies on low-cost materials, due to the small amount of electricity  
 517 generated per MFC unit, the fact that terracotta, air-dry Fimo™ and air-dry clay were  $40 \pm 10$   
 518 times cheaper than CEM (per Watt produced), adds an extra advantage in using these  
 519 materials. In addition those MFCs had a COD reduction of nearly 80% which was 20% more  
 520 than what the CEM based MFCs achieved. At last, the capabilities of those air-cured materials  
 521 open another promising avenue to the MFC research as their fabrication can be done with 3D  
 522 printing and/or extrusion techniques using the EvoBot platform. Thus it is envisaged that  
 523 monolithically 3D-printed MFCs are feasible and can potentially emerge from platforms such  
 524 as EvoBot.

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