

# 1 Cathode materials for ceramic based microbial fuel cells (MFCs)

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## 22 23 Abstract

24 This study showed the electrochemical performance of different cathode electrodes tested  
25 on a ceramic separator functioning as a cation exchange membrane. Particularly, three  
26 different carbonaceous-based materials (carbon cloth (CC), carbon mesh (CM) and  
27 carbon veil (CV)) have been used as an electrode and as the current collector. When used  
28 as electrode, CC outperformed the others. The carbonaceous materials have been  
29 modified using conductive paint (PA) and micro porous layer (MPL). With these  
30 modifications, the current output was two-three times higher. Generally, the current  
31 produced was slightly higher with MPL treatment compared to PA except in the case of  
32 CV-MPL that had lower output probably due to the negative effect of the heat treatment  
33 on the mechanical strength of the CV. In the case of PA, the current collectors do not  
34 seem to affect the output. The same consideration can also be done for the MPL except  
35 for the CV. The surface morphology seems to explain the results. Linear correlation was  
36 found between current produced and nanoscale roughness and skewness. The results  
37 indicated that those morphological parameters increased the contact between the cathode  
38 and the ceramic surface, thus enhancing the current generated. The further addition of the  
39 inorganic non-platinum group catalyst (Fe-AAPyr) on the surface significantly enhanced  
40 the performances. Following MPL modification and MPL-Fe-AAPyr addition, CM was  
41 the most cost effective support. CV was the most cost effective support with PA  
42 modification.

43  
44 **Keywords:** carbonaceous materials, ceramic separator, current production, morphology,  
45 Fe-AAPyr catalyst

## 1. Introduction

Water availability has been an important global challenge and consequently, water treatment is critical in successfully addressing this problem. The common technologies available for wastewater treatment are very efficient in degrading organics, but very expensive to operate, mainly due to the electricity used for mixing and pumping oxygen. Alternative approaches, such as Microbial Fuel Cells (MFCs), have been explored in order to keep the whole process efficient and significantly decrease the costs [CITATION]. This emerging bio-electrochemical technology is able not just to degrade organics but also to transform the chemical energy stored in the chemical bonds of various compounds including water pollutants into useful electricity [1,2]. The electricity generated could be used for small-scale real applications [3-5], but besides the advantages this technology has, it is still at the lab scale of development.

Major concerns related to MFC utilization are: i) the relative high cost of the electrode materials [6-7]; ii) the low power output caused by the unoptimized both anodic [8] and cathodic [9] conversion processes.

Anodic processes are mainly dominated by the electrically active bacteria, which adhering to the electrode can degrade organics releasing and transferring electrons to the conductive electrode. Electrons moving through the external circuit generate electricity that can be usefully harvested and successfully utilized [3-5]. The understanding of bacteria attachment [10], electron transfer [11], biofilm formation and development [12,13] bacteria selection [14] is still a matter of ongoing investigation.

Cathode processes are mainly affected by the high overpotential caused by the low electrochemical activity of the catalysts used (inorganic or biotic) at neutral pH [15]. Although it has been shown [16, 17] that specific enzymes can significantly lower the overpotential of oxygen reduction reaction (ORR) at neutral pH, utilization of carbonaceous [6, 18] or transition metal-based [19, 20] catalysts is preferred due to the higher availability, low cost and durability. Platinum has been also traditionally used as a cathodic catalyst in MFCs [21] but it has been shown to suffer from rapid decrease in performance due to the fast poisoning effect of sulfide presence in the wastewater [22]. Two different avenues are currently being exploited aiming at a trade-off between cost decrease and effective ORR: i) catalysts based on utilization of carbon-based materials, having high conductivity and high surface area [6]; ii) inorganic catalysts such as iron (Fe) [23], cobalt (Co) [24, 25] and manganese (Mn) [26].

Moreover, the current collector design is also very important for guaranteeing high cathode performances. Several current collectors have been used in MFCs mainly based on carbonaceous materials and particularly carbon veil [27, 28], carbon cloth [29], carbon paper [30] and carbon felt [30]. Also, metallic meshes have been used, based on corrosion proof stainless steel [31]. An understanding of the best performing and cost-effective material is still necessary.

Some studies have shown that the formation of biofilm due to the direct exposure of the anodic solution in membraneless MFC configuration lead to an enhancement in the cathode performance as a result of a biocathode formation [32, 33]. The OH<sup>-</sup> production during the ORR leads to cathode alkalization [34, 35] and calcium and sodium carbonate precipitation on the cathode [36], lowering its long-term operation [32]. Consequently, the option of using a solid separator able to decrease the negative effects of cathode

92 alkalization seems to be reasonable for preventing cathode deactivation and keeping the  
93 anode chamber under strictly anaerobic conditions. Anionic and cationic exchange  
94 membranes have been used previously in a single chamber or double chamber MFCs  
95 [37]. It has been shown that cationic membranes are preferable than the anionic  
96 membranes most likely because they prevent an accumulation of protons at the anode that  
97 inhibits bacterial metabolism [38]. The main problem related with solid polymeric  
98 separators is the high cost of the membrane that makes them not suitable for a large-scale  
99 operation [39]. Recently, MFCs with ceramic cation exchange membranes, utilized as  
100 physical separator between the anode and cathode have been successfully developed and  
101 explored [39, 40]. The main advantages of ceramic separators are: i) low cost; ii) high  
102 ions selectivity; iii) high mechanical strength and iv) high durability [39, 40].

103 This work focuses on electrochemical analysis of different low-cost carbonaceous  
104 materials suitable for the design of cathodes in ceramic MFCs. Carbonaceous materials  
105 (veil, cloth and mesh) have been tested: i) without any pre-treatment, ii) coated with a  
106 micro porous layer (MPL); iii) coated with a conductive carbon paint. The performance  
107 of low-cost non-platinum group metals (non-PGM) Fe-Aminoantipyrine (Fe-AAPyr) has  
108 also been studied, demonstrating promising results compared to other materials. Cost-  
109 performance analysis of the different options has been also carried out, in the light of  
110 future large-scale applications.

111

## 112 2. Materials and method

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114 INSERT FIGURE 1 HERE

115

116 **Figure 1. Camera Photo of CV (a), CM (b), CC (c) and SEM images (1.2mm x 1.2**  
117 **mm) of CV (d), CM (e), CC (f).**

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### 120 2.1 Cathode Materials

121 Different materials have been tested as cathodes or cathodic support in ceramic-  
122 based MFC. Particularly, three carbonaceous electron collectors have been used  
123 identified as carbon veil (CV), carbon cloth (CC) and carbon mesh (CM) (Figure 1).  
124 Those materials were purchased from PRF Composite Materials (Dorset, UK), Saati  
125 (Legnano, Italy) and Electromar (Milan, Italy), respectively. CV, CC, and CM have been  
126 used as controls during the experiments.

127 Modifications were done with the addition of conductive paint from TIMCAL  
128 Ltd. Switzerland (PA) or a micro porous layer (MPL). The materials with PA and MPL  
129 have been additionally modified with Fe-AAPyr as a catalyst for ORR. PA was applied  
130 on the carbonaceous support using a brush, covering the entire surface with a PA loading  
131 of  $40 \pm 10 \text{ mg cm}^{-2}$ . MPL was done similarly as previously described [41]. In summary,  
132 0.7 g of TIMCAL carbon powder was put in a beacker with 9.1 mL of distilled water and  
133 21.5 mL of nonionic surfactant (Triton X100, Sigma Aldrich) and then mixed for ten  
134 minutes using a spatula. Then, 1 g of PTFE (Sigma Aldrich) was added, and the slurry  
135 has been mixed for another ten minutes. At last, 2.75 g of carbon powder was added, and  
136 the overall content was mixed for an additional ten minutes. The resultant mixture was  
137 then applied on the CV, CC and CM using a brush. Thermal treatment has followed,

138 where the materials coated with the MPL were inserted in an oven and heated up at  
139 250°C for almost 2 hours and cool down at room temperature before utilization. **The**  
140 **MPL loading was 50±10 mg cm<sup>-2</sup>.**

141 Fe-AAPyr has been prepared as previously reported [23] and added on the MPL  
142 surface using a micropipette covering the entire surface utilized. Particularly, Fe-AAPyr  
143 has been mixed with Nafion and isopropanol and put into a ultrasonic bath for 15  
144 minutes. Fe-AAPyr loading on the cathode surface was 0.3 mg cm<sup>-2</sup>.

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## 146 **2.2 Surface Morphology Analysis**

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148 Scanning Electron Microscopy (SEM) images have been taken using a SEM  
149 TESCAN mod. Mira II. Three images for each sample (CV, CV-PA, CV-MPL, CC, CC-  
150 PA, CC-MPL, CM, CM-PA, CM-MPL) at two different magnifications (100x and  
151 10000x) were acquired. The morphological features of the surfaces were obtained using  
152 previously reported image processing tools in Matlab [42]. A high-pass filter was applied  
153 to remove the low-frequency component, and low-pass filter to remove the high-  
154 frequency component from the images in order to produce roughness and waviness image  
155 components, respectively [43]. At 100x magnification, the high-frequency component  
156 images correspond to roughness in the range of 10-33 μm, and the low-frequency  
157 component images correspond to 100-400 μm. At 10000x magnification, filtering  
158 separates the images into low-frequency component at 1.3-5.5 μm scale and a high-  
159 frequency component at 60-500 nm. From all these waviness and roughness images, we  
160 have extracted roughness (Ra) and skewness (Rsk), which point to the domination of  
161 pores or peaks in the image.

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## 163 **2.4 Chemical Surface Analysis**

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165 Samples were characterized using a Kratos AXIS DLD Ultra X-ray photoelectron  
166 spectrometer with monochromatic Al Kα source operating at 225 W. No charge  
167 compensation was necessary. Survey spectra were acquired at pass energy (PE) of 80 eV,  
168 and C 1s and O 1s high-resolution spectra were acquired at PE of 20 eV. Data analysis  
169 and quantification were performed using the CASAXPS software. A linear background  
170 was used for C 1s and O 1s spectra. Quantification utilized sensitivity factors provided by  
171 the manufacturer. A 70% Gaussian/30% Lorentzian line shape was used for the curve-  
172 fits.

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## 174 **2.2 Ceramic Cell Set-up**

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176 Ceramic **cylinders were** made from porous earthenware material (International  
177 Biological Laboratories, Haryana, India). It was used as a physical separator or a  
178 membrane between the internal volume and the external face. In a typical MFC  
179 configuration, the anode could be inserted in the internal volume with the cathode on the  
180 external face or vice versa the anode on the external face with the cathode inserted  
181 internally. In **the present** study, the experimental setup involved cathode materials with  
182 geometric area of 108 cm<sup>2</sup> being wrapped around the cylindrical ceramic tube (Figure

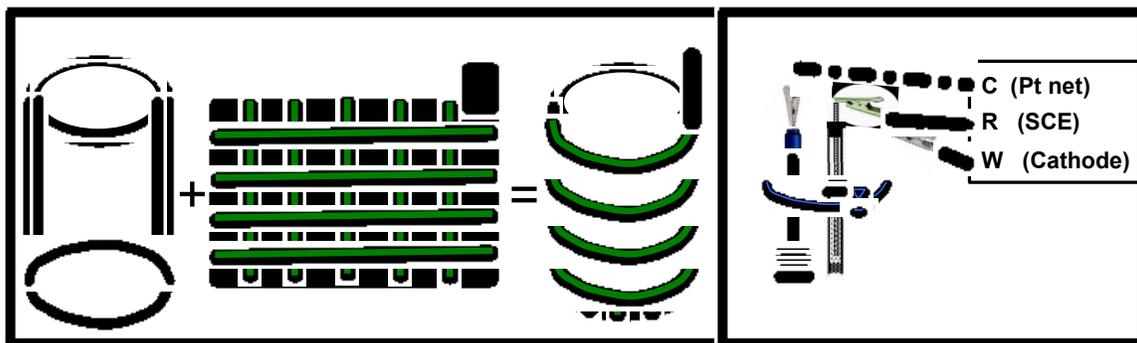
183 2.a). The average dimensions of the ceramic were: external diameter of 4.24 cm, height  
184 of 8.135 cm, wall thickness of 3.45 mm and an internal volume of 80.5 mL.

185 Inside the cylindrical ceramic, phosphate buffer saline (PBS, 50 mM) with 50 mM  
186 KCl was used as electrolyte (Figure 2.b). The pH of the solution was 7.3-7.4 simulating  
187 the typical conditions of a well-buffered wastewater.

188

## 189 2.5 Electrochemical Analysis

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192 **Figure 2. Schematic view of the ceramic support with the electrode in contact with**  
193 **the external face (a). Experimental design for the electrochemical measurements in**  
194 **a three-electrode configuration (b).**

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196 Linear Sweep Voltammetry (LSV) using a three-electrode configuration was used  
197 for materials characterization, with platinum mesh as the counter, saturated calomel  
198 (SCE, + 0.241 V vs SHE) as the reference and the actual cathode as the working  
199 electrode, as shown in Figure 2.b. The working cell has been left at open circuit potential  
200 (OCP) for at least 1 hour and then the LSV was performed. The scan rate used was 0.2  
201  $\text{mV s}^{-1}$  as previously reported [41].

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## 205 2.6 Cost-Analysis

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207 The cost-analysis was done considering the price of the support materials.  
208 Particularly, the cost was 28.5 US\$  $\text{m}^{-2}$  for CC, 16.3 US\$  $\text{m}^{-2}$  for the CM and 15.4 US\$  
209  $\text{m}^{-2}$  for CV. Being the area of a single cathode 108  $\text{cm}^2$ , the cost for each cathode was  
210 0.308 US\$, 0.176 US\$ and 0.166 US\$ for CC, CM, and CV, respectively.

211

212 The cost of the conductive painting (PA) such as that of the carbon particle (MPL)  
213 was not disclosed by the supplier, but it is considered to be negligible compared to the  
214 cost of the cathode materials, as the carbon powder industrial cost is in the order of  
215 magnitude of 1 \$ per 10 kg. The cost of the cathode with the addition of PA or MPL will  
216 increase, especially for the process with MPL, which includes heating treatment and  
217 addition of PTFE. Since the quantity of PA or MPL applied on the support was the same,  
218 the costs due to the process of cathode preparation should be similar to that of the other  
219 low temperature fuel cells already industrialized, and the difference is expected to be  
mainly due to the support cost that has been previously identified.

220 A rough estimation of the catalyst Fe-AAPyr cost considering the materials price  
221 showed by Sigma Aldrich catalog was previously done at 3.5 US\$ g<sup>-1</sup> [23]. Due to the  
222 catalyst loading of 0.3 mg cm<sup>-2</sup>, the additional cost due to the catalyst addition was 0.11  
223 US\$.

### 224 3. Results

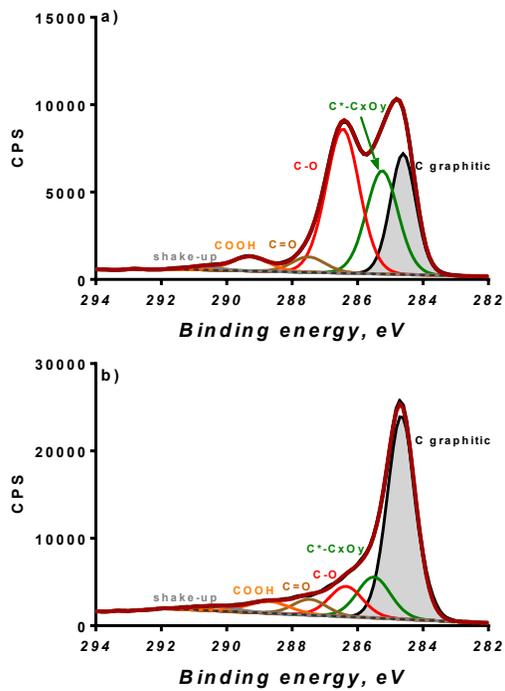
#### 225 3.1 Chemical Analysis

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227 **3.1 Chemical Analysis**  
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229 **Table 1** shows elemental composition and carbon chemical speciation of support  
230 materials as obtained from high resolution XPS analysis. Elemental composition of  
231 support materials is similar with approximately 11-19 at % of oxygen present, with CC  
232 having the lowest, while CM the highest amount. The presence of nitrogen and sodium  
233 (2-3 at.%) was detected in CC and CB supports. Chemical properties of carbon were  
234 evaluated from chemical speciation obtained from high-resolution C 1s spectra shown in  
235 **Figure 3**. CM has lowest graphitic content (peak at 284.6 eV and shake-up at 290.9 eV)  
236 and the highest amount of different types of C-O groups, i.e. C-O at 286.4 eV, C=O at  
237 287.5 eV and COOH at 292.2 eV. These three peaks were summed up to represent total  
238 amount of surface oxides present in **Table 1** - %C<sub>x</sub>O<sub>y</sub>. Peak due to secondary shifted  
239 carbons, such as C\*-C<sub>x</sub>-O<sub>y</sub>, confirms high oxygen functionalization for CM support. The  
240 CC has the highest graphitic content, and a small amount of surface oxides detected. The  
241 CV is similar to CC with slightly higher amounts of surface oxides present. **The surface  
242 chemistry analysis showed a very similar composition that should not influence the  
243 performances output.**

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245 **Table 1.** Elemental composition and carbon chemical speciation of supports  
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	<b>C 1s %</b>	<b>O 1s %</b>	<b>C=C</b>	<b>C*</b>	<b>%C<sub>x</sub>O<sub>y</sub></b>
CM	81.2	18.8	28.4	22.8	48.8
CC	88.3	11.7	62.6	12.3	25.1
CV	87.2	12.8	49.1	16.6	34.2

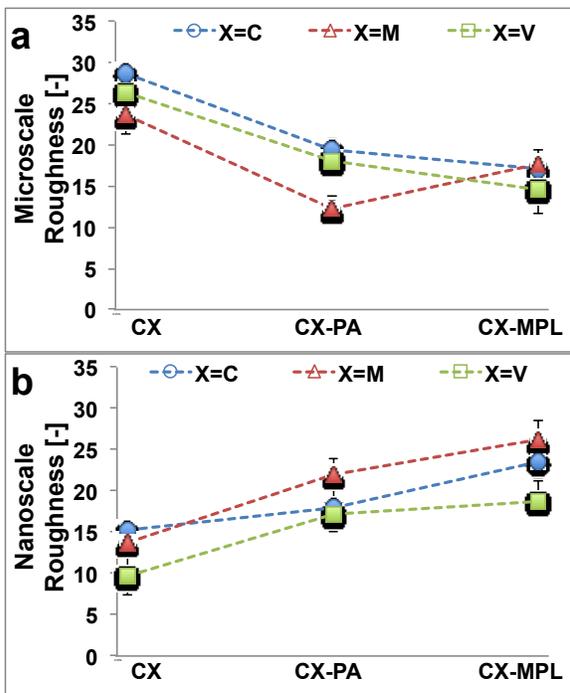
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 249 **Figure 3. High resolution C 1s spectrum for a) CM and b) CC carbon supports**

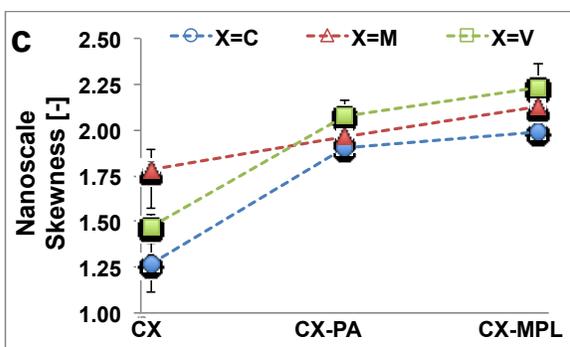
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 251 **3.2 Morphological Analysis**

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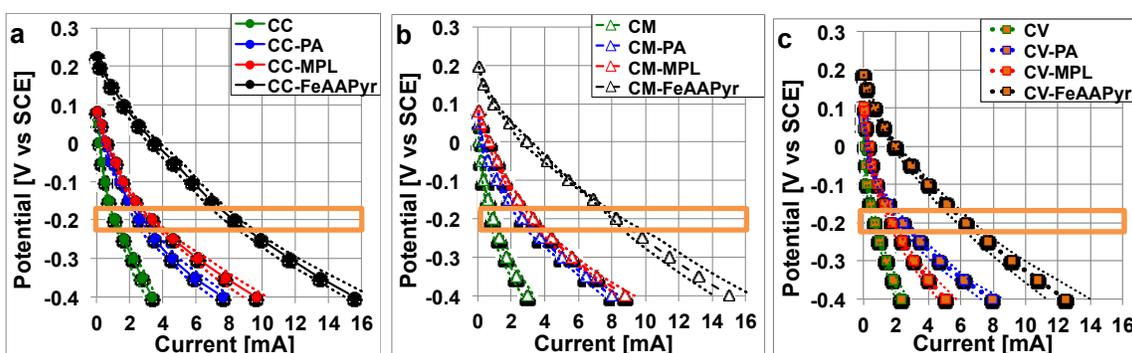
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256 **Figure 4. Roughness at the micro scale between 100 and 400  $\mu\text{m}$  (a), at the nano**  
257 **scale between 60 and 500 nm (b) and porosity at nanoscale between 60 and 500 nm**  
258 **(c).**

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260 The SEM images processing led to important conclusions regarding the surface  
261 roughness and porosity of the materials tested. It can be noticed that CV and CC have  
262 similar roughness at the micro scale (100-400  $\mu\text{m}$ ) that is slightly higher than the  
263 roughness of CM (Figure 4.a). On the contrary, CC and CM have higher roughness at the  
264 nano scale (60-500 nm) compared to CV (Figure 4.b), with CC having the highest  
265 roughness at both scales. CM has larger fibers compared to CV that in a higher resolution  
266 image appear much smoother in comparison to CV, but at nano scale they showed higher  
267 unevenness that is overlooked without the filtering and DIP procedure we have applied  
268 (Figure 4.b). Generally, independently of the support adopted, the addition of PA and  
269 MPL led to an increase in the roughness in the nano scale 60-500 nm (Figure 4.b) and  
270 decrease of the roughness at the micro scale 100-400  $\mu\text{m}$  (Figure 4.a). These results can  
271 be explained by the coating and/or the PA and MPL are filling the gaps between the  
272 carbon fibers making the surface more uniform at the micro-scale (Figure 4.a), while  
273 increasing the roughness at smaller nano-scale due to the intrinsic roughness of the  
274 coating itself and/or irregular particles distributed on the substrate. Also the skewness, a  
275 metrics of porosity, was calculated for all scales while only the porosity at nanoscale of  
276 60-500 nm showed important information. In addition, to the increase in nano-roughness  
277 as discussed above, the application of coating led to a substantial increase in the porosity  
278 at the nano scale (Figure 4.c). The increase in small pores with addition of coating on top  
279 of the support is causing an increase in nano-roughness that is important characteristic's  
280 in the fuel cell design as will be discussed below.

### 281 282 3.3 Electrochemical Results



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**Figure 5. Cathode polarization curves of CC-based cathode (a), CM-based cathode (b) and CV-based cathode (c). Green line indicates the performances of the support only; blue line indicates the modification with PA, red line the addition of MPL and black line the addition of MPL and Fe-AAPyr.**

Single electrode linear sweep voltammetry was done on each material (Figure 5). The results showed two main important aspects: i) the addition of PA and MPL increased the cathodes performance compared to the raw materials despite similar open circuit potential (OCP); ii) the addition of Fe-AAPyr boosted up both OCP and current generated. More specifically, in the case of CV, the current produced at -0.2V for CV only was  $0.69\pm 0.11$  mA and increased 3.6 times with the addition of PA ( $2.45\pm 0.28$  mA) and 2.6 times with the MPL coating ( $1.78\pm 0.22$  mA) (Figure 5.c). When CM was used as current collector, the current produced at -0.2V was  $0.89\pm 0.21$  mA for CM only and increased 3 times with the addition of PA ( $2.68\pm 0.23$  mA) and 3.7 times with the MPL coating ( $3.29\pm 0.29$  mA) (Figure 5.b). CC produced a current of  $1.07\pm 0.15$  mA that increased 2.5 times ( $2.62\pm 0.15$  mA) and 3.1 times ( $3.34\pm 0.32$  mA) due to PA and MPL addition, respectively (Figure 5.a).

A further significant increase in the cathodes performance was obtained with the utilization of Fe-AAPyr catalyst. This catalyst has been used previously in hydrogen PEM fuel cell [44] and double chamber MFC [23] and now for the first time has been characterized by half-cell configuration, simulating a single chamber MFC with ceramic membrane separator. It has been shown previously that Fe-based catalysts possess high electrochemical activity at very acidic or basic pH levels [44]. Despite the utilization of a well-buffered solution (50 mM PBS) with 0.1 M KCl as electrolyte, it has been shown also that severe basic conditions (pH up to almost 14) takes place on the catalytic sites [34, 35] that have also been seen previously utilizing wastewater and sodium acetate [45]. The Fe-AAPyr cathodes demonstrated almost doubled current production at -400 mV vs. SHE independently of the support used.

It has to be noted that at pH equal to 7, the theoretical OCP is  $\sim 820$  mV vs SHE ( $\sim 570$  mV vs SCE) and in this particular study, Fe-AAPyr cathodes showed the highest OCP that was  $\sim 205\pm 5$  mV vs. SCE (Figure 5). Lower OCP was recorded when the materials with and without additional coating were investigated. Those values were in the range between 50 and 100 mV vs SCE (Figure 5). Those results underlined a massive overpotential of approximately 360 mV with Fe-AAPyr utilization and approximately 470-520 mV without catalysts added. The high overpotentials are in line with the literature [19] and seem to be the main problem of the cathodes working at neutral pH.

### 3.4 General Trend and Cost Considerations

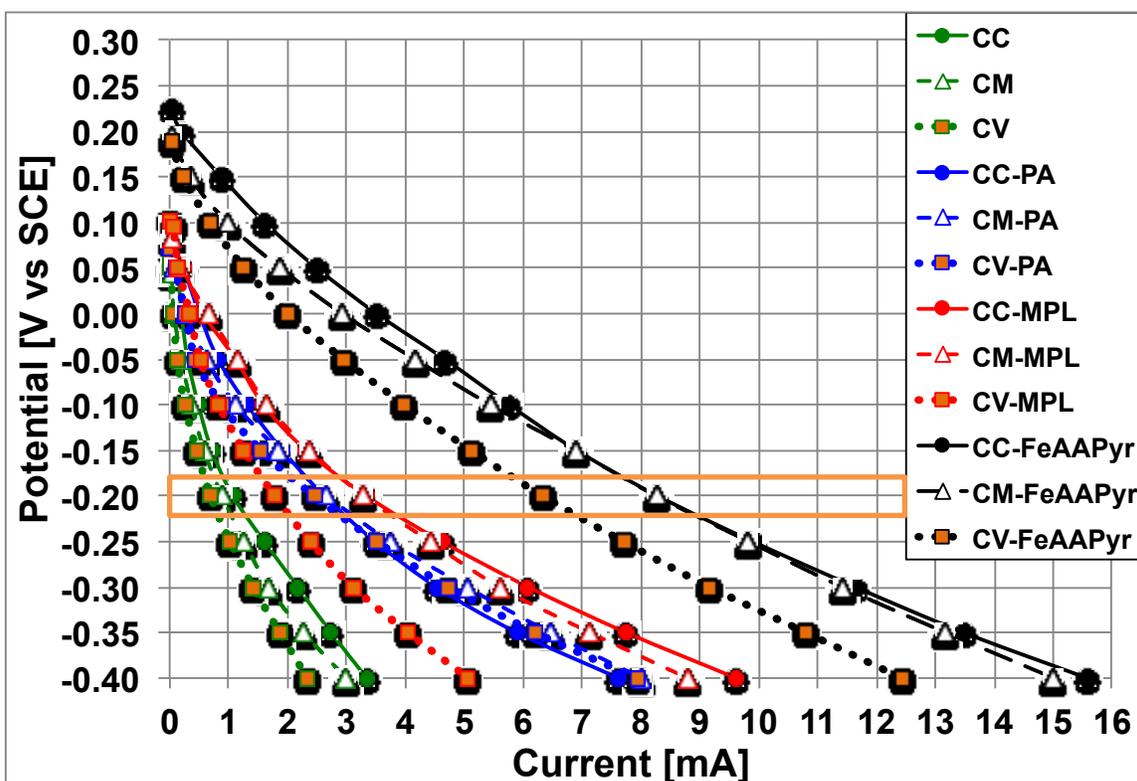
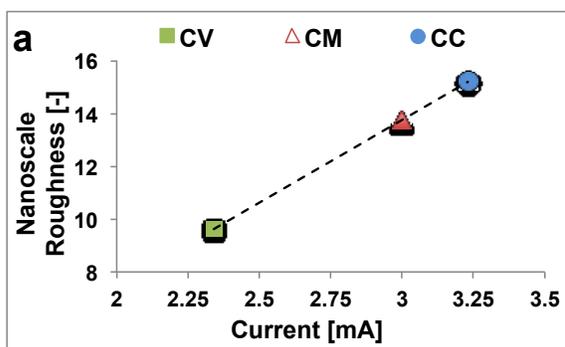


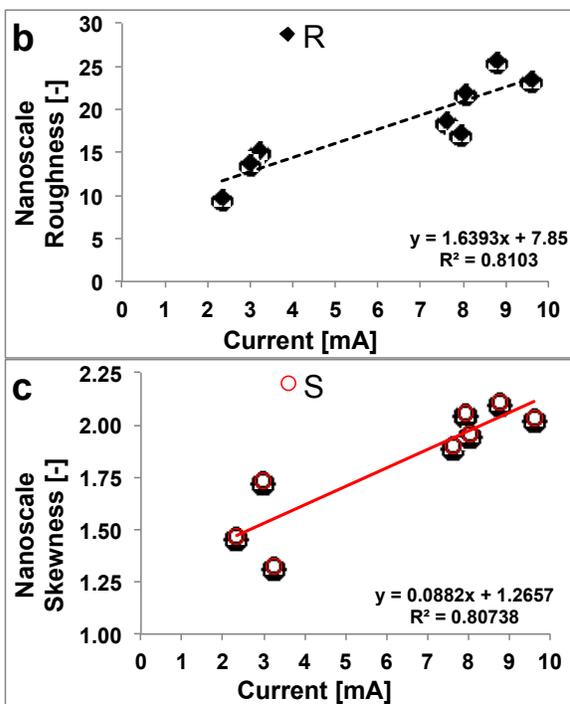
Figure 6. Overall electrochemical cathode performances trends.

In order to have a complete view of the results, all the polarization curves from the materials investigated have been overlapped (Figure 6). It can be noticed the significant difference in the electrochemical performance between the raw carbonaceous materials CV, CC, CM and the same materials after treatment. Considering the raw materials, a comparable electrochemical behavior remains till -0.2 V for all materials. Between -0.2 V and -0.4 V, CC performed better than CM and the latter outperformed the CV. At potential tested of -0.4 V, the CC produced  $1.07 \pm 0.15$  mA that was 20% higher than CM ( $0.88 \pm 0.21$  mA) and 54% CV ( $0.69 \pm 0.11$  mA).



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342 **Figure 7. Correlation between current produced by the support and the roughness**  
343 **of the support at nanoscale (60-500 nm) (a). Nanoscale roughness (b) and skewness**  
344 **of the cathodes tested as a function of the current produced.**

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**These results** may be explained by the differences in surface morphology of the carbonaceous supports, especially the roughness and porosity at the nano scale, which correlates linearly with the current produced by the supports themselves (Figure 7.a). CV had the lower current produced and the lower roughness at nanoscale. The latter may lead to a lower contact between the cathode and the ceramic support and consequently lower performance (Figure 7). In contrast, both CC and CM are rougher than CV at nanoscale and the much better contact increased the current generated (Figure 6). **On the other hand, surface chemistries of the raw materials were similar (Section 3.1) and consequently did not affect the performances output of the raw materials.**

All three carbonaceous materials have been modified by adding PA and MPL, and this modification has increased the performance significantly (Figure 6). This, at least, doubling of the current output can be due to the fact that the addition of PA or MPL, first of all, increases the conductivity of the material by filling the gaps between the fibers, and, second, dramatically increases the materials roughness at nanoscale (60-500 nm) (Figure 4.b) that is beneficial for a better contact of the cathode with the ceramic membrane. The current produced from all materials investigated (except with Fe-AAPyr) was plotted as a function of the roughness and skewness at nanoscale and a linear dependence with  $R^2 > 0.8$  was found (Figure 7.b and Figure 7.c) indicating that both morphological nanoscale parameters are central for increasing the current output.

Moreover, it can be noticed that the addition of MPL on the surface of the CC and CM is increasing the current output more than the addition of PA (Figure 6). Opposite result was obtained when CV was used, and this can be due to the thermal treatment that

368 the materials were subjected to after the application of the MPL slurry. In fact, CC and  
369 CM have much higher thickness and mechanical strength compared to CV and, evidently,  
370 they better withstand temperatures above 200°C. A single layer of CV seems not to have  
371 enough mechanical strength to support stress for long-term operation but this problem  
372 could be solved by utilizing several layers packed one on each other or reinforcing the  
373 fibers with addition of low amount of PTFE.

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### 375 **3.5 Cost – Morphology – Performance Correlation Analysis**

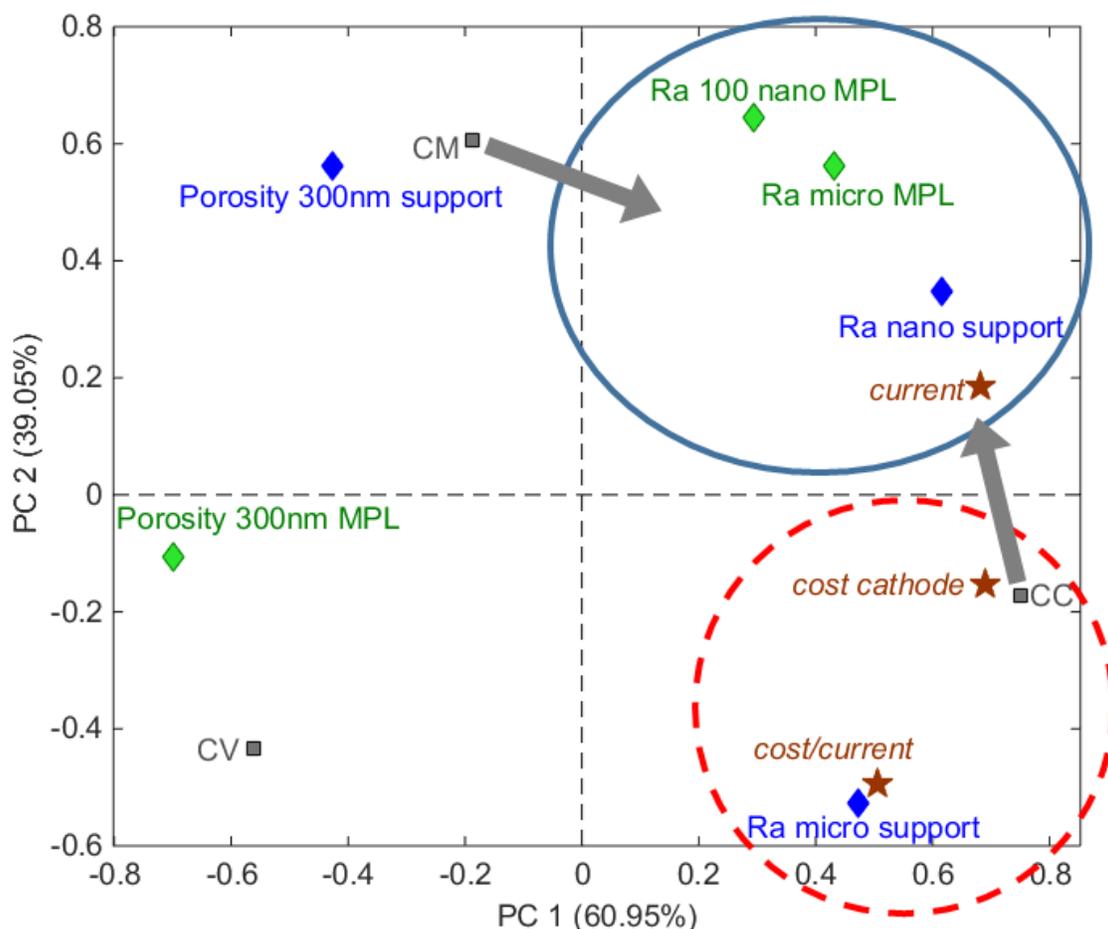
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377 The material cost is another fundamental factor to consider for large-scale practical  
378 applications. In fact, despite CC performed 20% and 54% higher than CM and CV  
379 respectively, CC had the highest cost, which was 43% and 46% higher than CM and CV,  
380 respectively. This indicates that CM is the most cost-effective material studied with the  
381 lowest cost to produce 1 mA, precisely 0.198 US\$ mA<sup>-1</sup>, followed by the CV (0.24 US\$  
382 mA<sup>-1</sup>) and CC (0.287 US\$ mA<sup>-1</sup>). **The cost was calculated considering the current**  
383 **produced at -0.2V vs SCE.**

384

385 The addition of PA resulted in similar output independently from the  
386 carbonaceous support (CV, CC, CM) (Figure 6). Consequently, the carbon support with  
387 the lowest cost, i.e. CV is the most cost-effective material when PA is utilized. Similar  
388 conclusions can be drawn when MPL is added, with the exception of the CV-MPL  
389 (Figure 6). Therefore, CM is also the most cost-effective material when MPL is utilized.

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**Figure 8. PCA bi-plot of current, cost, cost/current generated by the Fe-AAPyr and morphological properties of support and support with MPL. Circles are samples, diamonds and stars are variables.**

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The performance was significantly enhanced when non-PGM Fe-AAPyr electrocatalyst was utilized (Figure 8). Similarly, in this case, the support materials did not play an important role in the current generated (Figure 6). CM was, again, the most cost-effective support to be utilized.

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For understanding the effects the morphological properties of the support and the MPL on the performance of Fe-AAPyr based cathodes, principal component analysis was applied to the data table combining the following variables: nanoroughness Ra of the support and MPL, micoroughness Ra of the support and MPL, nanoporosity (Rsk) of the support and MPL, cost of the cathodes, current as produced by the system combining Fe-AAPyr, MPL, PA and support and cost/current ratio. Figure 8 shows that the current generated is dependent on both nano- and micro-roughness of MPL. If the support has initially high nano-roughness it is enhanced even more by the addition of MPL. In terms of performance of the Fe-AAPyr based cathodes, both CM and CC showed similarly good performance, but CC is much more expensive resulting in a high cost/current ratio, making CM the optimal support. CM in combination with MPL results in a morphology

411 with the highest nano and microroughness providing good contact and, therefore, good  
412 current collection efficiency.

413 These results bring to the conclusion that generally, among the materials studied,  
414 the support or current collector is not the key factor for the current output, and it is  
415 actually the support modification that lead to efficient current densities. Consequently, a  
416 lower cost of the support is necessary. In several cases, CC outperformed CM and CV,  
417 but due to its high cost, it does not seem suitable for large-scale applications where cost is  
418 as important as the performance. Functionalizing the cathode surface is essential for the  
419 achievement of improved current generation. In general, Non-PGM Fe-AAPyr has shown  
420 to be a promising catalyst that could be applied on various low-cost, carbon-based  
421 **substrates**. Therefore it is important to explore further its suitability for ceramic MFC  
422 applications in real operating conditions.

423 **Despite those considerations, the current produced was generally low probably**  
424 **due to the scarce contact between the cathode and the ceramic. Further studies should go**  
425 **in the direction of incorporating the catalytic layer directly onto the ceramic surface in**  
426 **order to minimize ~~the-both~~ ohmic ~~resistance~~-and contact resistance.**  
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#### 428 **4. Conclusions**

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430 Carbon cloth (CC), carbon mesh (CM) and carbon veil (CV)) were studied as an  
431 electrode and current collector. CC performed better than CM and CV but CM was the  
432 most cost effective material. Addition of conductive paint (PA) and micro porous layer  
433 (MPL) boosted up the performances significantly with a two/three-fold increase. A linear  
434 relationship between roughness and skewness at nanoscale **and** the current was found  
435 indicating that the modification reported probably enhanced the contact between the  
436 electrode and the ceramic separator. CM was the most cost effective materials with MPL  
437 addition. The addition of low cost platinum-free Fe-AAPyr as inorganic catalyst  
438 increased further the current generated.

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