Appendix 1. Additional Information

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Appendix A. Conductivity of Graphene

Conductivity refers to the ability of a material to conduct electric current. The mechanism of conductivity varies depending on whether the material is a metal, or non-metal (semiconductor, insulator, or electrolyte).

Metals are known for their high electrical conductivity, a property that allows them to efficiently conduct electric current.¹ In case of a metal, the outer electrons of atoms are not tightly bound to the individual atoms but are instead free to move throughout the material. The exceptional conductivity can be attributed to the free movement of electrons within the metallic structure. Since the electrons are delocalized, they can move freely throughout the material. This free movement of electrons facilitates the easy flow of electric charge, making metals excellent conductors. However, when an electric field is applied (i.e., when voltage is applied across a metal), these free electrons drift in response to the field, creating an electric current.

Nonmetals, on the other hand, generally exhibit poor electrical conductivity. This is primarily due to their atomic structure. Unlike metals, nonmetals do not have a delocalized electron sea. Instead, electrons in nonmetals are tightly bound to their respective atoms, limiting their ability to move freely. As a result, nonmetals impede the flow of electric current and are considered semiconductors or insulators. However, semiconductors have an intermediate level of conductivity between conductors (like metals) and insulators (non-conductive materials), which can be altered by introducing impurities through a process called doping.² Doping introduces additional charge carriers (either electrons or "holes" where electrons are missing) into the semiconductor material.³ In the presence of an electric field, these charge carriers move, contributing to the electric current. The mobility of charge carriers in semiconductors is influenced by factors such as temperature and the concentration of impurities. On the other hand, insulators are materials that do not conduct electricity easily. The electrons are tightly bound to atoms, and there are only few free charge carriers. When an electric field is applied, very little current flows through insulators.⁴ When the electric field becomes strong enough (exceeds the material's dielectric strength), insulators can undergo electrical breakdown starting to conduct. Electrolytes are materials that conduct electricity through the movement of ions. Ions are charged particles that can be positively or negatively charged. In an electrolyte solution, ions are free to move in response to an applied electric field. Positive ions (cations) move toward the negative electrode (cathode), and negative ions (anions) move toward the positive electrode (anode). The movement of ions constitutes the flow of electric current in electrolytes.⁵ In summary, metals rely on the free movement of electrons, semiconductors use doped charge carriers, and electrolytes involve the movement of ions. Insulators, on the other hand, generally resist the flow of electric current.



Figure A.1. Top-view schematics of the two layers of graphene flakes and the resulting network graph. Nodes are positioned at overlaps and connected through graphene flakes. Reproduced with permission.⁶ Copyright 2018, American Chemical Society.

However, there are exceptions, and some nonmetals can conduct electricity under specific conditions. For instance, graphite, a form of carbon, can conduct electricity due to its unique layered structure that allows electrons to move along the planes. Macroscopic graphene can outperform copper in electrical conductivity. Literature suggests that there are two main parameters to obtain a high electrical conductivity for macroscopic graphene. It is crucial to have a high in-plane electrical conductivity in most of the graphene flakes. The value for the in-plane electrical conductivity determines the maximum possible conductivity. Nevertheless, the lateral size of graphene flakes compensates a lower out-of-plane conductivity to a certain extent due to the larger overlap area and the lower contact resistance. Rizzi et al.⁶ proposed a network model to investigate the electrical characteristics of layered graphene relying on their microscopic properties to compute the macroscopic electrical conductivity. Graphene flakes were modelled as polygons to reproduce the shape of actual graphene flakes produced by liquid exfoliation. A proper edge-to-edge contact between flakes within the same layer is highly unlikely because of the production methods and would have a negligible effect

compared to that of overlapping flakes. For each overlap, a node is created in the corresponding flakes. Nodes that belong to the same overlap or share the same graphene flake are connected (Figure A.1).

Appendix B. Electrochemical performances of supercapacitor

A series of key parameters, and various techniques are used for the evaluation of the electrochemical performance of a SC. Among them, widely used techniques are: Cyclic voltammetry (CV), galvanostatic charge/ discharge (GCD), and electrochemical impedance spectroscopy (EIS). Three fundamental parameters (voltage, current, and time) for SC can be measured by all these techniques. Additionally, other SC performance metrics including capacitance, equivalent series resistance, operating voltage, time constant, energy and power performance can be derived from those parameters.



B.1 Cyclic voltammetry (CV)

Figure B.1. Typical characterisation parameters of supercapacitors a) CV curves c) GCD curves for ideal supercapacitor; b) CV and d) GCD curve distortion due to faradaic reactions⁷

CV is a powerful and popular electrochemical technique commonly employed to investigate the reduction and oxidation processes of molecular species.⁸ Such technique is used to study the electrochemical properties related to electroactive surfaces, and characterize the electrode materials primarily.^{9, 10} In this technique, a linearly changed electric potential is applied against time to measure the current. The graphical analysis of a cyclic voltammogram provides redox peaks (reduction and oxidation peaks of the material) and predicts the capacitive behaviour of the electrode. Therefore, the potential at which the material is oxidized and reduced can be found in this technique.^{11, 12} A typical cyclic voltammogram for an electrochemically reversible and diffusion-controlled redox process is shown in *Figure*

B.1a. The curves obtained through CV for both EDLCs and pseudocapacitors are evaluated to measure the capacitance (C) of the material deposited over the electrode by using Equation (1). The shape of the resulting CV curves for an ideal SC is rectangular. However, the variation in the shape and size of the plot can occur when the deposited materials over the electrode are dissolved into the electrolyte. It can also happen due to the detachment of the electrode contacts during cyclic repetition, *Figure B.1b.*¹³ The gravimetric capacitance (Equation (4)), lengthwise capacitance (Equation (6)), areal capacitance (Equation (8)), volumetric capacitance (Equation (10)), energy density (Equation (12)) of the electrode or total SC cell can be obtained *via* integration of CV curves.¹¹

B.2 Galvanostatic charge discharge (GCD)

GCD test is considered as the most accurate and versatile approach, and most widely used method for the capacitance assessment (Equation (2)).¹³ The direct current at a constant level is imparted in this method for repetitive charging and discharging of the SC device or the working electrode. A potential *Vs* time plot is obtained from this method, *Figure c*. Additionally, the cyclic stability of SC devices can be studied from GCD. The symmetric curves obtained from the charge discharge through GCD confirms the capacitive behaviour of the device, enlightening capacitance as the function of applied voltage. Additionally, gravimetric capacitance (Equation (5)), lengthwise capacitance (Equation (7)), areal capacitance (Equation (9)), and volumetric capacitance (Equation (11)) for SC materials can also be obtained *via* GCD.¹¹

Techniques	CV	GCD	EIS
Principle	CV is varying the	GCD is applying a	Measuring impedance of a
	potential against	positive or negative	power cell as a function of
	time and measuring	current against time and	frequency by applying
	the current	measuring the voltage	alternating current (AC)
Merits	• Degradation	Capacitance	Resistance calculation
	process	calculation	• Specific capacitance
	• Specific	• Differentiate	calculation
	capacitance	between EDL and	• Differentiate between
	• Differentiate	PC	resistive and inductive

Table B.1. Technical merits and demerits of the CV, GCD, and EIS techniques^{13, 14}

	between EDLC		nature
	and PC		• Nondestructive technique
			• Relaxation time for
			recharging
			• Exhibit Degradation
			behavior
Demerits	• Show only	• Exhibit same	• Evaluation at small voltage
	kinetic aspects;	triangular shape for	only
	thermodynamic	all double layer	• Discrete behavior above
	aspect is	capacitive materials	10 ⁶ Hz
	neglected		

B.3 Electrochemical impedance spectroscopy (EIS)

EIS, an electroanalytical method, measures the impedance of a power cell as a function of frequency by applying the alternating current (AC) instead of the direct current (DC). The fundamental approach of EIS is the application of a spectrum of small-amplitude sinusoidal AC voltage excitations to the system. The frequency of the AC signal is varied, and the overall impedance of the cell is recorded as a function of frequency. The resulting data are usually expressed graphically in two types of plots: a) the Nyquist plot, which shows imaginary V_s real impedance at different frequencies, and b) the Bode plot, which shows absolute impedance V_s frequency. For SC materials, EIS testing can be used to study the impedance, charge transfer, mass transport, and charge storage mechanisms as well as to estimate the capacitance (Equation (3)), energy, and power properties.^{10, 11} A summary of technical merits and demerits of several characterization techniques are presented in *Table*.

B.4 Key metrics for supercapacitor performances

The key parameters used to evaluate the electrochemical performances of a SC are capacitance, operating voltage, equivalent series resistance, power density, energy density, and time constant. Capacitance is defined as the ratio of the charge stored (or separated) to the potential difference between the conductors.¹⁵ The total charge storage ability of a SC device is termed as the capacitance, which is calculated from the formula stated in *Table B.2* ((*Equation (1)-(3)*). It is noteworthy that, while specifying the capacitance of SC, a more intrinsic specific capacitance is measured in terms of the mass of the electroactive materials or length, area and/or volume of the SC device (*Equation 4-11*). The other two important

parameters for evaluating SC performances are: energy density and power density. Energy density, derived from the *Equation (12)*, denotes the amount of energy that can be delivered from a SC. The power density denotes how faster the energy can be delivered by a SC and can be calculated from the *Equation (13)*, *Equation (14)* or *Equation (15)*.

Parameters (Unit)	Information obtained	Measurement formula	Equation
Capacitance (F)	Ability to collect and store	$C = \frac{\int I dV}{V}$	Equation 1
	energy in the form of	V	Equation 2
	electrical charge	$C = \frac{i \Delta V}{\Delta t}$	Equation 2
		$2\pi C = \frac{d (-Z'')}{d \left(\frac{1}{f}\right)}$	Equation 3
Gravimetric	Charge storage ability per	$C_m = \frac{A}{2 s m V}$	Equation 4
capacitance (F g^{-1})	unit mass	$C_{\rm m} = \frac{i \Delta t}{m \Delta V}$	Equation 5
Lengthwise	Charge storage ability per	$C_l = \frac{A}{2 s l V}$	Equation 6
capacitance (F cm ⁻¹)	unit length	$\mathbf{C}_1 = \frac{i \Delta t}{l \Delta V}$	Equation 7
Areal capacitance	Charge storage ability per	$C_A = \frac{A}{2 s a V}$	Equation 8
(F cm ⁻²)	unit area	$C_{A} = \frac{i \Delta t}{a \Delta V}$	Equation 9
Volumetric	Charge storage ability per	$C_{v} = \frac{A}{2 s v V}$	Equation 10
capacitance (F cm ⁻³)	unit volume	$\mathbf{C}_{\mathbf{v}} = \frac{i \Delta t}{V \Delta V}$	Equation 11
Energy density	Amount of energy able to	$E = \frac{1}{2}CV^2 = \frac{QV}{2}$	Equation 12
(Wh kg ⁻¹)	deliver	2 2	
Power density (W	How faster the energy to	P = VI	Equation 13
kg-1)	deliver	$\mathbf{P} = \frac{E}{t}$	Equation 14
		$\mathbf{P} = \frac{V^2}{4R}$	Equation 15
Coulombic	Reversible capacity	$\% E = \frac{C_{charging}}{C_{discharging}} \times 100$	Equation 16
efficiency		usenut ytty	

[C= capacitance, I= current density, V= voltage window, i=discharging current, Δv =discharge voltage, Δt = discharge time, -Z" = imaginary part of the impedance, A= integrated area of the CV curve, s= scan rate (mV s⁻¹), m= mass of the electroactive material on both electrodes, l=length of the electrode, v=volume of the SC, R= resistance]



Figure B.2. Approaches for enhancing energy and power densities of supercapacitor¹⁶

Among the performance metrics for all kinds of energy storage and conversion systems, power density and energy density are the most often used parameters for their performance evaluation for all kind of applications. Compared to batteries, SCs suffer from lower energy density.¹⁷ The energy density depends on the capacitance and working voltage window (V). Therefore, increasing the capacitance or extending operating voltage window will enhance the energy density of a SC. Power density depends on their working voltage window (V) and internal resistance (R). Therefore, in addition to extending the working voltage window, one of the ways to increase the power density is by the reduction of internal resistances of SC components. *Figure B.2* summarizes the approaches to improve the energy and power density of SCs.¹⁶ Additionally, the long cycle life of SC devices is one of the highly desirable characteristics for certain applications. However, the cycle life, when extremely long, is difficult to measure directly. Therefore, the capacitance retention rate is used as an indirect measurement to estimate the cycle life of a SC. By comparing the capacitance after given thousands of cycles with that of the first cycle in GCD test, the capacitance retention value is obtained.¹¹

Appendix C. Various type of electrolytes for supercapacitors

C.1. Aqueous electrolytes

The aqueous electrolytes can be categorized into three types: acidic solution (i.e., H_2SO_4 solution), alkaline solution (i.e., *KOH* solution), and neutral solution (i.e., Li_2SO_4 , Na_2SO_4 , or *KCl* solution). Due to high ionic conductivity of acidic aqueous electrolytes, SC electrode materials show better performance in comparison with the neutral aqueous electrolytes.¹⁸ They get dissolved in water, providing high ionic conductivity, and low internal resistance compared with organic electrolytes. Therefore, SCs with an aqueous electrolyte solution may possess a higher capacitance and power than capacitors containing organic electrolytes. The maximum working voltage of aqueous electrolyte is limited to 1.23 V owing to the thermodynamic decomposition of water. In addition, they can be prepared and employed without much tight control of the production process, whereas organic electrolytes require strict preparation procedures to obtain pure electrolytes. Commonly used aqueous electrolytes are inorganic salts (e.g., *LiCl*, *NaCl*) (for Li-ion & SCs), alkali (e.g., *KOH*) and inorganic acid (e.g., *H₂SO₄*) in water (for SCs).

C.2 Organic electrolytes

Organic electrolytes allow a much wider voltage window of about 3.5 V resulting in a large advantage with respect to higher energy density. Among the organic electrolytes, propylene carbonate (PC) is the most used solvent, because of their environmentally friendly nature and a wide voltage window with good conductivity. The complex purification and preparation procedures may cause safety problems due to the flammability and toxicity of some of the organic solvents. Furthermore, their low conductivity could lead to lower power and smaller capacitance. Several combinations of organic solvents and lithium salts have been examined as electrolytes for ambient-temperature, rechargeable lithium batteries.¹⁹ Inorganic or organic salts e.g. Lithium hexafluorophosphate ($LiPF_6$); Tetraethylammonium tetrafluoroborate (TEABF₄) in organic solvents (carbonates, ethers, sulfones, etc., some of which may be fluorinated) are the common examples of organic electrolytes for SCs.

C.3 Ionic liquids

Ionic liquids (ILs) are salts having uncommonly low melting points, below 100°C, which is usually liquid at room temperature.²⁰ They are inherent and competitive electrolytes due to their ability to overcome many disadvantages of the conventional aqueous and organic electrolytes,²¹ for instance volatility, high thermal and chemical stability, extensive electrochemical stability window between 2 and 6 V, low flammability, nontoxicity and the

wide variety of cation and anion combinations. They are composed entirely of ions, solventfree and liquids at room temperature, making them attractive "green electrolytes". They are usually highly viscous liquids with low ionic conductivity at ambient temperatures, seriously influencing their electrochemical performance. Salts in ILs (for Li-ion batteries) or pure ILs (for SCs); organic cations (e.g., imidazolium, pyridinium, pyrrolidinium, etc.) with inorganic or organic anions (e.g., H_2SO_4) in water (for SCs) are commonly used ionic liquids.

C.4 Polymer electrolytes

In order to meet the safety (for instance the leakage, flammability and toxicity of organic electrolytes), flexibility and multi-functionality requirements for advanced energy-storage devices (ESDs), polymer electrolytes are considered to be the best candidate to replace liquid electrolytes due to their wide electrochemical window, good thermal stability and less risk with electrolyte solution leakage.²²

C.4.1 Dry solid polymer electrolytes (Polymer-salt complex electrolytes)

A solid polymer electrolyte is prepared by dissolving inorganic salts into a polar functional polymer, which form a solid electrolyte with ion conducting after drying. With interactions between metal ions and polar groups inside the polymers, electrostatic forces are generated due to the formation of coordinating bonds. Since the coordination of cations to align along the polymer chain is weak, after applying an electric field the cations in the electrolyte may migrate from one coordinated site to another. Various polymers are used to form dry and solid polymer electrolytes including polycarbonate, poly (methyl methacrylate) (PMMA), poly (ethylene oxide) (PEO) and poly (vinyl alcohol) (PVA). For metal ions, various soluble compounds, such as salts containing lithium (Li), sodium (Na) and potassium (K) can be used. Nevertheless, the room ionic conductivities of SPEs $(10^{-8} \sim 10^{-5} \text{ S cm}^{-1})$ are lower than the required conductivity of $10^{-3} \text{ S cm}^{-1}$ in actual application, limiting their wide practical applications.²²

C.4.2 Gel polymer electrolytes

The gel polymer electrolytes possess a higher ionic conductivity at room temperature. It has attracted research attention because of the combination of the advantages of high ionic conductivity of liquid-based electrolytes, and the high stability of solid-based electrolytes. This combination superiority is embodied in high ionic conductivity and good interfacial properties from the liquid phase as well as good mechanical properties from the solid component. They are safer to use compared to liquid-based electrolytes. The majority of GPEs exhibit outstanding ionic conductivity in the order of 10^{-3} S cm⁻¹ at ambient temperature, which can boost the electrochemical performance of the cells involving GPEs. Consequently, GPEs have become one of the most desirable alternatives for the fabrication of advanced ESDs with enhanced safety and flexibility.^{23, 24}

C.4.3 Plasticized polymer electrolytes

A host polymer with lower molecular weight like poly(ethylene glycol) (PEG), propylene carbonate and ethylene carbonate is used to produce plasticized polymer electrolytes. The rigidity of the polymer structure is decreased with a change in their mechanical and thermomechanical properties. The glass transition temperature of the particular polymer electrolyte system is also decreased. The increase of salt dissociation capability and the reduction of crystallinity results in enhancement of charge carrier transportation.²³ Polymer electrolytes are found to exhibit higher ionic conductivity at higher plasticizer concentration at the cost of their mechanical stability.²⁵

C.4.4 Composite polymer electrolytes

The addition of inorganic fillers in polymer electrolytes increase the mechanical strength and interfacial stability of the resulting electrolytes, providing a new branch of polymer electrolytes (Pes) which are known as composite polymer electrolytes (CPEs).²⁶ By doping different types and amounts of high dielectric constant fillers, especially inorganic inert fillers into the polymer matrix, the electrical properties of polymer electrolytes can be improved. Ceramic materials are one of the most commonly used inorganic dopants. They are fragile and have low dielectric strength. By combining such inorganic dopants with polymers, the new composite electrolytes consist of ceramic particles, they can be regarded as heterogeneously disordered systems, with electrical properties highly dependent on the relative permittivity and conductivity of the dopants. Moreover, electrical performances of these composite electrolytic materials are affected by the size, shape and volume fraction of the dopants.²³

Appendix D. Electrochemical performances of textile supercapacitors reported in literature

		1	D shaped							
Substrate (Reporting year)	Device Configuration	Device	capacitance	Energy density		Power density	Capacitance	retention	Flexibility	Ref
Carbon fibre and Gold coated plastic fibre (2012)	Commercial graphite pen ink used as active material and deposited on fibre surface. Spacer wire evenly twisted onto the surface of one fibre electrode, two fibre electrodes placed parallelly in a flexible plastic tube filled with PVA- H ₂ SO ₄ electrolyte	$11.9-19.5 \mathrm{mF}\mathrm{cm}^{-2}$		$1.76 \times 10^{-6} - 2.70 \times 10^{-6}$	$\mathrm{Wh}~\mathrm{cm}^{-2}$	Up to 9.07 mW cm ^{-2}	Similar after 15,000 cycles		Slight drop at 180° , 360° bending	27
Carbon microfibre bundle (2013)	Carbon microfibre bundle coated with MWCNTs as core electrode in the centre of coaxial SC and carbon nanofibre (CNF) film prepared by electrospinning as outer electrode	6.3 mF cm ⁻¹ (86.8 mF cm ⁻	2)	$0.7 \ \mu Wh \ cm^{-1} \ (9.8 \ \mu Wh$	cm ⁻²)	583 μW cm ⁻¹	94% after 1,000 cycles		Negligible change at 180° bending	28
Carbon fibre (2013)	Electrochemically reduced GO (ERGO) coated carbon fibre followed by acid treatment (ERGO@CF–H) with PVA– H ₃ PO ₄ gel electrolyte	13.5 mF cm ⁻¹ (307 mF cm ⁻²)	at 0.05 mA $\rm cm^{-1}$	1.9 mW h cm ⁻¹ (21.4 mW h	cm ⁻²)	0.74 mW cm^{-1} (8.5 mW cm ⁻²)	85%	after 5,000 cycles	No decay at bending 0° -180°	29
rGO fibre yarns (2014)	rGO fibre yarns deposited on a titanium current collector and separated by a PVDF membrane	409 Fg ⁻¹		14 Wh kg ⁻¹		25 kW g ⁻¹	No decay after	5,000 cycles		30

Table D.1. Summary of Carbon-based supercapacitors or supercapacitor electrodes

Carbon nanotube and rGO composite yarn (2014)	Two SWNTs/rGO electrode mounted on a PET substrate using PVA–H ₃ PO ₄ electrolyte	116.3 mF cm ⁻² , 45	$F\ cm^{\text{-3}}$ at 26.7 mA	cm ⁻³	6.3 mWh cm ⁻³		1085 mW cm ⁻³	93% after 10,000	cycles	>97% after 1,000	bending cycles at	90°	31
GO, CNT and their mixture wet spun filament (2014)	Polyelectrolyte-wrapped carbon nanomaterial (graphene, CNTs and their mixture) core-sheath fibre, RGO@CMC, CNT@CMC, RGO+CNT@CMC, Two coaxial fibres twisted together and coated with PVA electrolyte	$269~\mathrm{mF~cm}$ $^{-2},239~\mathrm{F~cm}^{-3},8.0~\mathrm{mF~cm}^{-1}$ (liquid	electrolyte 1 M H ₂ SO ₄), 177 mF cm $^{-2}$, 158 F cm $^{-3}$,	5.3 mF cm ⁻¹ @ current density of 0.1 mA cm ⁻² (solid electrolyte H_3PO_4/PVA)	5.91 mWh cm ⁻² (liquid) 3.84 mWh cm ⁻² , 3.5 mWh	cm ⁻³ (solid)		No decay within 2,000 times		Dropped 2% at 200 times of bending and rose	persistently up to 111% at 1,000 times of bending		32
GO fibre (2014)	Region-specific reduction of GO fibre by laser irradiation, to prepare rGO/GO/rGO single fibre SC with IL electrolyte of 1-butyl-3-mthylimidazolium tetrafluoroborate	$1.2~{\rm mF}{\rm cm}^{-2}$ at 80 $\mu{\rm A}$	$\mathrm{cm}^{-2},0.45~\mathrm{mF}~\mathrm{cm}^{-2}$ at	$200 \mu\mathrm{Acm^{-2}}$	2-5.4×10 ⁻⁴ Wh cm ⁻²		3.6–9×10 ⁻² W cm ⁻²	No degradation after	1,000 cycles	No decrease after 160	bending cycle		33
Carbon fibre yarn (2015)	Two hybrid AC drop casted carbon fibre yarn electrodes, twisted together, dipped in PVA-H ₃ PO ₄ electrolyte, dried	45.2 mF cm^{-1}	at 2 mV s^{-1}		$6.5 \mu \mathrm{Wh}\mathrm{cm}^{-1}$		27 μWcm ⁻¹	86.6% after	10,000 cycle	68%	after 1,000	bending	34
CNT fibre (2015)	GO sheets coated on CNT fibre, reduced to RGO, forming a core–sheath-structured CNT/RGO composite fibre. Two composite fibres drawn into gel electrolyte consisting of PVA, phosphoric acid, and water, followed by twisting	$68.4~{ m F~cm^{-3}}~(126.7~{ m F~g^{-1}})$ at 31	$mA cm^{-3}$		2.4 mW h cm^{-3} , $3.8 \mu \text{W h cm}^{-2}$		$0.016 \text{ W cm}^{-3}, 0.025 \text{ mW cm}^{-2}$	No decay after 10,000		No decrease at bending 180°			35

Graphene fibre (2015)	Spun-rGO//rGO coaxial all graphene fibre SC with PVA gel coating as				ı kg ⁻¹),	n ⁻² with			le		ending			36
	separator, a dip-coated cylinder sheath fibre as the other	[82 F g ⁻¹)			(15.5 Wł	14 μWh cr	lectrolyte		,000 th cyc		times of b			
	electrode followed by reduction, coating of H ₂ SO ₄ -	F cm ⁻² (Wh cm ⁻²	ed to 10	c 10nic e		ay at 10		ter 100			
	PVA gel electrolyte	205 mł			17.5 μ ^ν	increas	organic		No dec		92% af			
Graphene	Two MWCNTs-rGO fibre	,1	at		1 ⁻³							60		37
fibre (2015)	electrodes with PVA-H ₃ PO ₄	^r cm	:m ⁻³ ;	cm ⁻³	h cm			-3	er	cycle	ease	ottin		
	electrolyte	5 mF	8 F c	mA	mW) V cm	% aft	000	decr	ar kn		
		0.3	38.	50	3.4			700 mV	939	10,	N_{O}	afte		
Porous	PGRs freeze-dried for 24 h to										0°,	ile		38
graphene	obtain dried porous graphene	or									5°, 9) cyc	glove	
ribbon (PGRs)	ribbons (DGRs) followed by	cm ⁻²							es		ng 4	r 100	to g	
(2015)	immersion in H ₃ PO ₄ -PVA	mF							cycl		endi	afte	over	
	electrolyte, two pairs of	(78.3	[_]						,000		e at b	95%	en w	
	electrodes pressed together	ao' ⁻	F cm						ter 5		rease	80°,	g wh	
		8.7 F	2 m]						% af		o dec	5°, 1	nding	
		20	3.1						66		Ž	13	bei	
GO fibre	Region specific reduction of	Ļ									ter			39
(2015)	wet spun GO fibre by laser	n ⁻² a	ė						1,00(se afi	SS		
	irradiation to prepare alternate	ıF cr	v cm⁻						fter		creas	cycle		
	rGO-GO electrolyte-free fibre	4.3 n) mA						3% a	/cles	o de	000		
		1	5(6	S	Z	1,		40
PVA/KGU	into a populiquid arrestalling CO				Μh			g ⁻¹)			ు	80		10
(2016)	dispersion before wet spinning				32 m			mW	les		cycli	und 1		
(2010)	and chemical reduction two	¢,			3 (5.3			23.9	cyc		fter (n 0 a		
	bundles of PVA/RGO fibres	cm			cm			n ⁻³ (2	,000		ion a	twee	nes	
	with PVA-H ₂ SO ₄ -H ₂ O gel	(41 F			Μh			W cr	fter 1		stenti	g be)0 tir	
	electrolyte	bre 2			97 m	-1)		.9m)	5% a		1% r€	ndin	r 10(
		Ξ			5.	ad		26	85		67	þe	fo	41
Graphene and	Hybrid fibre based on graphene	_				8 to								41
iew-walled	and rew-walled CN1s (G_{10}	0 m∕			l6 to	84.6			00					
nanotubes	KOH aqueous electrolyte	at 20			23.4	-1 at	ы З		10,0					
composite	KOIT aqueous electrolyte	8 ⁻¹			rom	h kg	ł N C		ufter					
varn (2018)		2.6 F			ied f	6 W]	54.50		6% 5	les				
Jun (2010)		312	aa^-		Var	9.6	II		89.	cyc				

		2D shaped					
Substrate (Reporting year)	Device Configuration	Device capacitance	Energy density	Power density	Capacitance retention	Flexibility	Ref
Cotton fabric (2010)	Dipping and drying of cotton with SWNT ink electrodes with LiPF ₆ electrolyte to form fabric SC	140 Fg ⁻¹ at 20 μA cm ⁻² , 0.48 F cm ⁻²	20 Wh kg ⁻¹ at 10 kW kg ⁻¹		98% after 130,000 cycles	No decrease @100 th 120% strain	42
Cotton Stretchable fabric (2010)	Dipping and drying of cotton with SWNT ink electrodes with LiPF ₆ electrolyte to form fabric SC	62 Fg^{-1} at 1 mA cm ⁻²			<6% decrease after 8,000	Similar after stretched 120% strain 100 times.	42
Cotton fabric (2010)	Conformal coating of SWNTs ink with 2 mol L^{-1} Li ₂ SO ₄ electrolyte	70–80 Fg ⁻¹ at 0.1 mA cm ⁻²			Coulombic efficiency		43
Cotton cloth (2012)	Brush-coating of cotton cloth with GO suspension ink as electrode, nickel foam current collector, pure cotton cloth separator with 6 M KOH electrolyte	81.7 Fg ⁻¹	7.13 Wh kg ⁻¹	1.5 kW kg ⁻¹	93.8% after 1,500 cycles		44
Cotton woven (2017)	Screen printing of GO, followed by electrochemical reduction to produce rGO- cotton electrode, with PVA- H ₂ SO ₄ gel electrolyte	2.5 mF cm ⁻² , 257 F g ⁻¹			97% after 10,000 cycles	95.6% after folding 180° for 2,000 cycles	45
Cotton fabric (2019)	Dry-coated and subsequently two step reduced GO coated cotton fabric electrode, raw cotton fabric separator, PVA- H ₃ PO ₄ electrolyte sandwiched	$464~\mathrm{F}~\mathrm{g}^{-1}$ at 0.25 A g^{-1}	27.05 W h kg ⁻¹		91.6% after 1,000 cycles	No apparent performance degradation after 0° and 180° bending	46

Cotton fabric	Graphene ink screen-printed on														47
(2022)	cotton									,000					
	textiles with PVA-H ₂ SO ₄ gel	m^{-2}				\mathbf{h}		n^{-2}		t 10					
	electrolyte	nFc				шW	at	W cr		afte	SS				
		3.2 I				0.28	cm ⁻²	3 mV		95%	cycle				
Polyester	CNT dip-coated onto PET	ш													48
fabric (2014)	fabrics electrode	10^{-4}													
		$1.4\times$	cm^{-2}												
Polyester	Sandwiching 2 dip-dried														49
fabric (2019)	graphene nanoparticle/											of			
	polyester electrodes and 1 h-											cles o	50		
	boron nitride/polyester) cyc	guibu		
	dielectric layer in between to	m^{-2}										\$ 100	l ber		
	form flexible textile-based	oF ci										tains	eatec		
	capacitor (FTC)	261										Sus	repe		
Poly-cotton	Graphene coated textile	2									es		0	0°	50
(65/35) 2/1	electrodes with PVA-H ₂ SO ₄ gel	cm	de)							ufter	cycl	ter	g 15(at 18	
twill fabric	electrolyte	mF	ectro							3% a	000	% af	guibi	les a	
(2020)		2.7	(ele							86≈	15,	986	ber	cyc	
Cotton and	Cotton and polyester screen-		0		1										51
polyester	printed with AC as SC	3 F	cm	-95	ΩVs [−]					000					
fabric (2011)	electrode, polyester separator	e 0.4	шA	, 85	10 n					r 10,					
	and Li ₂ SO ₄ and Na ₂ SO ₄	trode	at 5	SO4	at 1-					afte	SS				
	electrolyte	Elec	cm ⁻²	(Na_2)	Fg^{-1}					92%	cycle				
Polypropylene	Reactive inkjet printing of rGO		1								s				52
fabric (2019)	layers on PP fabric as electrode	-2	at 0.			m^{-2}			2	%	sycle				
	with PVA-H ₃ PO ₄ gel	^t cm	g_1)	2		Vh c			cm	100	000				
	electrolyte to form flexible	3 mF	9 F	cm		3 mV			шW	lost	r 5,0				
	solid-state SC	13.3	(79.	шA		1.18			4.6	Aln	afte				
Carbon fabric	Entangled carbon nanofibres														53
(2012)	(CNFs) synthesized on flexible														
	carbon fabric (CF) via water-	1								cle					
	assisted chemical vapor	V s ⁻) cy(
	deposition to form CNF/CF	5 m								2,000					
	electrode with	-1 at								fter 2					
	0.5 M Na ₂ SO ₄ aqueous	Б								% a					
	solution	140								≈95					

Carbon fibre Carbon fabric screen-pr	ted										t			54
knit and with activated carbon w	ı	cm ⁻²	<u>-</u> -	ven							ing a			
woven fabric solid polymer electrolyt		1 F	6 Fg	Ň							endi	puq		
(2013)		, 0.5	d), 6	cm ⁻²							ter b	5°, 8		
		F_{9}^{-1}	nitte	19 F							% af	, 13	$^{\circ}0$	
		88	Â	0.1							80	90	18	
Carbon cloth Electrochemically active	ed	'n		ſŦ.	-									55
(2014) carbon cloth electrode w	h	лF сı	t 10	3 mF	Fg ⁻¹)				00					
PVA-H ₂ SO ₄ electrolyte		88 n	<u></u> з-1) а	15.	5 m]				20,0					
		ode	mFg	, SC	0.76				ufter					
		lectr	8.8	Vs^{-1}	n ⁻² (5% 8	/cles				
		Ξ	6	В	cı				6	ં				56
Activated CNTs and graphene (GN									000					30
Carbon fibre modified composite AC	-	m^{-2}	00			cm^{-2}		l^{-2}	it 1,0					
felt (2015) textile electrodes, non-v	ven	nFc	2,70	5		V h c		V cm	ay a					
fabric separator, with K	H	50 r	vice	-tcm		2 μV		0 μV	dec	cles				
electrolyte		а, С,	de	mF		112		49	No	cyc				
Stainless steel Two chemically convert	1					.2								57
fabric (SSF) graphene (CCG) on SSF		<u>ц</u>	- mF	-2		386			0			ള		
(2016) electrode with 1 M H_2S	to	1^{-2} at	80.4	∧ cm		- ² at			7,500		800	ndir		
form flexible solid-state		Fcm	1 ⁻² , 1	m∕		u cm			fter		fter 8	g-be		
symmetrical SC		8 ml	A cm	at 1		WF	m^{-2}		% af	sə	% af	chin	SS	
		730.	2 m/	cm		19.2	W CI		96.8	cycl	96.4	stret	cycl	
Silver fibre Electrophoretic depositi	n of	6)	5											58
fabric (SFF) graphene on SFF, electro	le	cm ⁻	cm ⁻						er	<i>i</i> cles				
(2017) with KOH 3 M electroly	2	mF	mA						afte	00 cy				
		172	at 4						97%	5,00				

			1D shap	ed					
Substrate (Reporting year)	Device Configuration	Device	capacitance	Energy density	Power density	Capacitance	retention	Flexibility	Ref
Cotton	Cotton yarns coated with	Гт.		Ч				er Jes	59
yarn (2017)	PPy nanotubes	74.0 mF	cm^{-2}	7.5 μWl cm ⁻²				97% aft 200 cyc	
Carbon fibre thread (2015)	Carbon fiber thread (CFT) @ PANI as positive and functionalized carbon fiber thread (FCFT) as negative electrode, coated with PVA- H ₃ PO ₄ gel electrolyte and twisted together	High operating voltage	(1.6 V).	2 mWh cm ⁻³	11 W cm ⁻³			Unchanged at 100% strain	60
			2D shap	ed	1	1		I	1
Substrate (Reporting year)	Device Configuration	Device	capacitance	Energy density	Power density	Capacitance	retention	Flexibility	Ref
Cotton fabric (2013)	PPy-coated cotton fabrics electrode, prepared in mixed surfactants: cetyltrimethylammoni-um bromide (CTAB) and sodium dodecyl benzene sulfonate (SDBS), with NaCl solution	51.7 mAh g^{-1}				Negligible decay after 100	cycles		61

Table D.2 Summary of Conducting polymer-based supercapacitors or supercapacitor electrodes

Cotton	In situ oxidation												62
fabric	polymerization of pyrrole												
(2013)	in presence of												
	lignosulfonate as both	Å g⁻¹											
	template and dopant to	0.1 i											
	prepare PPy/ lignosulfonate	-1 at											
	(PPy/LGS) coated cotton	Ч Ч											
	fabric electrode	304											
Cotton	PPy nanorods deposited on				at			_					63
fabric	cotton fabrics via in-situ				6 ⁻ -	\mathbf{l}^{-2}		after					
(2015)	polymerization	aa^{-1}			Vh k	A cn		ω^{-1}	/cles				
		25 F			Λ.7 V	6 m		00 F)0 cJ				
<u> </u>		32			57	0.		2(5(64
Cotton	Pry coated cotton as		en					5					01
	working, Pt sheet as		WOV	2				ycle					
(2016)	counter, Ag/AgCI as		non	cm				00 c					
	reference electrode with	7,	and	mF			-	5,0					
	$1M H_2SO_4$ electrolyte	4,117	,191	,905	kg ⁻¹		V kg	ithin					
		ted 2	en 2	cs 2	Wh		55 V	le w					
		Knit	wov	fabri	5.94		259.	Stab					
Cotton	In situ polymerization		-										65
woven	coating of PPy		nF	m^{-2}				0					
and knit		48,	3481	nAc				5,00					
(2019)		1,7	d 4,8	t l n				fter					
		over	nitte	n^{-2} a				3% a	cles				
	x	W	\mathbf{K}	cn				88	cy				66
Cotton	In-situ chemical			_	_					00			00
knit fabric	polymerization of PPy on		y,	∧s_	tota			te),	SI	110	l to		
(2019)	fabric with PVA-H ₂ SO ₄ gel	n^{-2} at	tivel	5 m]	d on			rolyi	neor	cheo	nced		
	electrolyte	Fcm	spec	⁻² at	ased			elect	; (aq	stref	nha		
		33 m	- ² re:	cm	g^{-1} t			gel e	sycle	fter	nce e		
		1,43	Vcm	0 m]	/h k	es)	kg^{-1}	les (Oth c	8 %	citar	-2	
		and	l m∕	d 45	.3 W	trod	5 W	cyc	500(.0~0	capa	cm	
		481	I put	1 an	-2 (2	elec	(57.	500	% at	78%	ice (mA	
		ode	\mathbf{s}^{-1} $\mathbf{\hat{c}}$	e 10	/hm	of 2	m^{-2}	after	539	n <u>yu</u>	dev	at 5	
		lectr	шV	evic	4 M	lass	0 M	; %0	bove	lectr	mes,	60%	
		Щ	S	q	0.	π	-	3	<u>,</u> al	р Ц	÷	1(

Cotton	PPy-coated fabric		-		at								67
knit fabric	electrodes via chemical		$^{-2}$ at	vice	${ m m}^{-2}$ ϵ	7		nce	ycles				
(2019)	polymerization with 1 M	e	F cm	², de	Vh ci	/ cm		acita	00 ci				
	H ₂ SO ₄ electrolyte	trode	3 mł	cm^{-2}	4 μ	МШ		cap	: 2,0(
		Elec	5,07	mA	102.	0.39		%06	afteı				
Nylon/lyc	Chemical polymerization	>											68
ra (80/20)	synthesis of PPy on fabric	0 m'								thed	00		
knitted	with 1.0 M NaCl electrolyte	at 1			÷					tretc	r 1,0		
fabric		ao ^T			ı kg		W kg			fter s	% fo		
(2012)		3.3]			7 Wł		3.4 1			% af	100	nes	
		12	\mathbf{s}^{-1}		6.7		75			90	to	tin	
Polyester	Repeated spray-coating of		of										69
fabric	PEDOT:PSS solutions	g_1 at	rate	_									
(2018)	containing 5 wt% DMSO	$0 F_{g}$	scan	JVs ⁻									
		75.3	the 9	20 n									
Polyester	In situ polymerization of		1										65
knitted	PPy on fabric	щ	1 m/										
fabric		l3 m	- ² at	2									
(2019)		1,2	cm	cm									
Polyester	Electrospun PEDOT: PSS		'n										70
fabric	nanofibres deposited on	and	JA C		at 5			000					
(2020)	flexible PET substrates to	≥m ⁻²	at 5µ		kg^{-1}		- - -	r 1,0					
	obtain electrodes with	nFс	ω ⁻		Wh	:m ⁻²	Wkg	afte	SS				
	PVA-H ₃ PO ₄ polyelectrolyte	1.81	3.6I	7	0.32	μA c	11.8	92%	cycl				
Polyester	Conductive PET fabric		t		~								71
fabric	electrode is prepared by in-		n ⁻² a		cm ⁻³		ς,		~				
(2021)	situ polymerization of	le	uF cr	m^{-2}	Wh		/cm ⁻	ufter	ycles				
	aniline and pyrrole	ctroc	46 m	IA CI	43 m		05 W	2% 8	00 c				
		Ele	1,0	2 m	0.0		0.0	54.	1,0				
Cotton/po	DMSO-doped	r-	ť	_	ĥ	-	M						72
lyester	PEDOT:PSS-coated cloth	64 F	n^{-2} a	man	16 W	1 ⁻²),	9.70	cles,	cles				
(55/45)	as electrode and sweat as	it (7.	uF cn	al hu	lt 1.3	h cn	ut 32	10 cy	00 cy				
fabric	electrolyte	wea	45 m), reį	wea	μW	swea mW	4,00	5,00				
(2020)		cial s	d 8.4	ve_1.	zial s	1.63	cial s 0.40	fter	ufter				
		rtific	-1 and	07 A	rtific	зд_1 С	rtific _z -1 ((5% a	5% a				
		A	ad	0.	A	k	A k	75	4,				

PP non-	Reactive inkjet printing to				_				73
woven	fabricate PPy layers on	- ⁻¹ at	l at		ı kg		fter /cles		
textile	textile with direct freezing	3 F g	A g.	°C	2 Wł	Wk	4% a 00 cy		
(2021)	of inks	72.3	0.6	-12	6.12	139	55. ² 2,0(
Silk	In situ polymerization		A						65
woven	coating of PPy on fabric	Ц	1 m/						
fabric		49 m	⁻² at	7					
(2019)		1,32	cm	cm					
Wool	In situ polymerization		Ā						65
gauze	coating of PPy on fabric	Ц	1 m/						
fabric		07 m	- ² at	7					
(2019)		1,0(cm	cm					
Fibre	Conductive fibreglass cloth								74
glass	(CFC) derived from gas-								
cloth	phase polymerization of							S	
(2020)	pyrrole, followed by							ycle	
	electrochemical						s	ng c	
	polymerization of a layer of						ycle	endi	
	PPy attached to the surface						000	9 00 P	
	of the conductive fibreglass	n^{-2}			cm^{-2}		10, (r 1,0	
	cloth, sandwiching two	LF cn			Wh e		ufter	afte	
	PPy/CFC composites with	.6 m			35 µ'		4% а)8%	
	PVA-H ₂ SO ₄ electrolyte	549			48.8		92.4	96.(
Carbon	PEDOT nanofibre								75
fabric	electrode, carbon cloths as						cles		
(2010)	current collector and) cyc		
	electrospun PAN						00,00		
	nanofibrous membranes as						er 1(
	separator with ionic liquid	"ag_1					6 aft		
	electrolyte	20]					606		
CNT fibre	Two PANI deposited CNT							Se	76
woven	fibre textile stacked with	ao' ⁻					000	cycle	
textiles	PVA-H ₃ PO ₄ gel electrolyte	: 1 A					ter 2	200 50°	
(2014)		3 ⁻¹ at					iy afi	fter at 1.	
		.7 Fg					deca les	4% a ding	
		272					No cyc	96. [,] ben	

			1D shap	ed				
Substrate (Reporting year)	Device Configuration	Device capacitance		Energy density	Power density	Capacitance retention	Flexibility	Ref
Kevlar fibre (2011)	Kevlar fibres and flexible plastic wire substrates for ZnO NW arrays, Plastic wire/ZnO, Kevlar/ZnO/Au electrodes with KNO ₃ and PVA-H ₃ PO ₄ electrolyte	0.21 mF cm^2 at 100 mV s^1	(aqueous) and 2.4 mF cm ⁻² and 0.2 mF cm ⁻¹ (gel electrolyte)	2.7×10^{-8} Wh cm ⁻² (gel)	1.4×10 ⁻⁵ Wcm ⁻² (gel)			77
Carbon nanotube (CNT) yarn (2014)	CNT CNT, CNT@MnO ₂ CNT, CNT@MnO ₂ CNT@MnO ₂ with aqueous electrolyte	Asymmetric SC CNT@MnO2 (positive) and CNT (negative) possess a	capacitance of 12.5 F $\rm g^{-1}$ at a current density of 0.14 A $\rm g^{-1}$	1 to 2.12 Wh kg ⁻¹ (CNT CNT), For CNT@MnO ₂ CNT, up to 42.0 Wh kg ⁻¹ (low power density), and 28.02 Wh kg ⁻¹ (at high power density)	241.8 to 10 000 W kg ⁻¹ (CNT CNT), for CNT@MnO ₂ CNT, low power density of 483.7 W kg ⁻¹ , high power density	For CNT@MnO ₂ CNT 98% retention after 500 cycles, in comparison with 99% for CNT CNT	For CNT@MnO2 CNT, specific capacitance suffered only 0.5% reduction after 200 cycles of folding and unfolding actions	78

 Table D.3 Summary of Metal-based supercapacitors or supercapacitor electrodes

Carbon nanotube (CNT) yarn (2015)	NiO and Co ₃ O ₄ deposited on spun CNT yarn, two PVA-H ₂ SO ₄ coated CNT, CNT@NiO and CNT@Co ₃ O ₄ yarns placed together and coated with electrolyte again	CNT @ Co_3O_4 yarn based SC 52.6 mF cm ⁻² at 0.053 mA cm ⁻² ,	87.6 Ag^{-1} , CNT based SC 7.4 mF cm ⁻² , 13.4 Ag ⁻¹ , CNT@NiO based SC 15.2 mF cm ⁻² , 25.9 Ag ⁻¹	CNT@Co ₃ O ₄ yarn based SC 1.1 μ Wh cm ⁻²	CNT@Co ₃ O ₄ yarn based SC 0.01 mW cm ⁻²	Pure CNT, CNT@NiO and CNT@Co ₃ O ₄ maintain 96%, 94% and 91%, respectively, of original capacitance after 1,000 cycles	No decrease after 100 cycles bending at 90° and 180°	79
Carbon nanofibre (CNF) (2015)	CNFs containing Co ₃ O ₄ nanoparticles electrodes with 6 M KOH	586 Fg ⁻¹ at 1	Ag^{-1}			74% after 2,000 cycles		80
			2D shap	ed			1	
Substrate (Reporting vear)	Device Configuration	Device	capacitance	Energy density	Power density	Capacitance retention	Flexibility	Ref
Polyester (2020)	Ni nanoparticle in situ synthesized, two Ni-plated polyester electrodes with PVA-KOH gel electrolyte	450 mF cm^{-2}	at 7.5 mA $\rm cm^{-2}$					81

Silk fabric (2021)	Graphite coated Berlin (silver coated silk) as negative electrode and Nora Dell (Ni/Cu/Ag coated) fabric as positive electrode with biocompatible PVA- KCl gel electrolyte	32 mF cm ⁻² at 25 mV s ⁻¹ , 41 mF	cm^{-2} at 0.75 mA cm^{-2}		2.8 μ Wh cm ⁻² at 25 mV s ⁻¹ , 3.6 μ Wh cm ⁻² at 0.75 mA cm ⁻²			45% retention after 1,000 cycles		82
Carbon	Hydrogenated single-	ĹL	0							83
cloth	crystal ZnO @ amorphous	25 m]	A cm		^					
(2013)	ZnO-doped MnO ₂ core-	- ² (32	5 m∕		r cm			u.	cles	
	snell nanocables (HZM) on	cm.	at 0.		hWn	c	cm ⁻³	ó afte	0 cyc	
	with PVA-LiCl electrolyte	ie ml	:m ⁻³)	0	0.04 1	.44	ММ	37.5%	0,00	
Carbon	Electrochemically activated	(1	5		0	(1	_	8	Π	84
cloth	carbon cloth (EACC) as	it 6		ç	n I				~	
(2015)	anode and TiN@MnO ₂ on	m ⁻³ 8	0		u cm	cm^{-3}		y at	ycles	
	CC as cathode with LiCl) F c	cm		IM MI	M (deca)00 c	
	electrolyte	2.69	mA	1	1.5	1.7]		No	70,0	
Carbon	Uniform large-area									85
fabric	manganese oxide (MnO ₂)							les.		
(2016)	nanosheets on carbon fabric							cyc]		
	oxidized using O ₂ plasma	Vs^{-1}						,000		
	treatment (MnO ₂ /O ₂ -carbon	5 m						er 10		
	tabric) via electrodeposition	g ⁻¹ at						of aft		
	sulfate (Na-SQ) electrolyte	75 Fg						0% с		
Carbon	In-situ electrodeposition of	2				 		×		86
cloth	MnO_2 on carbon cloth with	at								
(2016)	$1 \text{ M Na}_2 \text{SO}_4 \text{ aqueous}$	г в_1	\log^{-1}							
	electrolyte	275	0.2 <i>F</i>							
Carbon	Zinc sulfide (ZnS)		-2)		at					87
textile	nanospheres		^r cm		kg_ 'g'g			fter	vcles	
(2016)	hydrothermally grown on	F g	.25 F mV		Wh Wk			6% a	00 c)	
	flexible carbon textile (CT)	540	(56 at 5	i	51 205			94.	5,0	

Carbon	KOH activation of		ſŦ			0						88
fibre	commercial CF threads		Jm (n^{-3}	⁻¹ or	-3 4		ter				
fabric	followed by coated with	-1 at	, 30	Fcr	Vh g	h cm		% af	ī t			
(2017)	PVA- H ₃ PO ₄ electrolyte and	Fg	$V s^{-1}$	² , 26	8 mV	[Mu		tr 80)00 a			
	woven to SC fabric	18.6	2 m	cm	2.58	3.6		Ove	10,0			
Carbon	Fabrication of Co ₃ O ₄	F						0				89
fabric	nanowires on flexible	290	s-1					5,00				
(2017)	carbon fabric electrode	le 3,	МV		kg ⁻¹		- - -	lfter				
	(CoNW/CF) with PVA-	ctroc	at 5 1		Wh		W kg	3% a	les			
	KOH gel electrolyte	Ele	ad [_]		6.7		5 K	95.	cyc			
Carbon	Anodic deposition of	1										90
cloth	MnOx on Pd coated carbon	-1 at	5									
(2018)	cloth electrode with 0.5 M	E E B	_cm_									
	Na ₂ SO ₄	186	шA									
Carbon	Viscose fibre woven fabrics											91
fibre	carbonized and activated											
fabric	(ACVF), CeO ₂ , ZnO	s ⁻¹								200		
(2020)	hydrothermally deposited,	тV				10		ş		ufter		
	CeO ₂ -ACVF & ZnO-ACVF	t 0.2				×10-'		sycle		4% 8		
	electrode, polyester	n ⁻² at				3.31>		00 0		13.4	les	
	separator, carbon cloth	LF cn			6	² at 3		it 5,0		ss by	cyc	
	current collector with PVA-	24 m			×10 ⁻	cm_		5% a		rease	ding	
	H ₃ PO ₄ electrolyte	13.			4.6	Wh		87.		Inc	ben	
Carbon	Crystalline nano-flowers							10				92
fibre	structured zinc oxide (ZnO)							s at				
textile	directly grown on carbon							ycle				
(2021)	fibre textile (CFT) via							00 c				
	hydrothermal process and	l Ag						r 3,0				
	fabricated with a binder-	¹ at j						afte				
	free electrode (denoted as	$\mathrm{F}_{\mathfrak{S}}$						32%	-			
	ZnO@CFT) for SC	201						90.3	Ag			

Substrate (Reporting year)	Device Configuration	Device	capacitance	Energy density	Power density	Capacitance retention	Flexibility	Ref
Cotton fabric	Dipping and drying of cotton into MXene	82.70 F	-					93
Cotton fabric (2020)	3D knitted cotton yarn coated with Ti ₃ C ₂ T _x and 1M H ₃ PO ₄ -PVA electrolyte	519 mF cm ⁻² at 2 mV s ⁻¹ , 1	(707 mF cm ⁻² at 2 mV s ⁻¹ g with 1 M H ₃ PO ₄ electrolyte)	25.4 μWh cm ⁻²	0.47 mWcm ⁻²	>100% over 10,000 cycles		94
Carbon cloth (2018)	Hydrothermal growth of MoS ₂ nanosheets on carbon fabrics, with Lithium metal counter electrode, and 1.0 M LiPF ₆ solution with mixture of ethylene carbonate and dimethyl carbonate (EC/DMC, 1:1 in volume) electrolyte	$159.38 \text{ mF cm}^{-2} \text{ at } 0.5 \text{ mA cm}^{-2}$				80.6% after 15,000 cycles		95
Carbon fabric (2018)	Stacking RuO ₂ coated carbon fabric (CF) as positive and $Ti_3C_2T_x$ coated CF as negative electrode with 1 m H ₂ SO ₄ electrolyte	Electrode 416 mF cm ⁻	² , 200 F g ⁻¹	Device 37 μWh cm ⁻² at 40 mW cm ⁻²		86% after 20,000 cycles		96

Table D.4 Summary of 2d material-based supercapacitors or supercapacitor electrodes

Carbon	Hierarchical flower-like			μ/	es	97
cloth	Mn ₃ O ₄ @N, P-doped carbon			65 W	cycl	
(2021)	(NPC) composite cathode	ad		32.0	000	
	with an electrochemically	t 1 A		g^{-1} af	r 10,	
	reduced porous carbon (PC)	00_1 1	/h kg	W kg	afte	
	anode and a PVA-Na ₂ SO ₄	97 F	96 M	02 k'	71%	
	hydrogel electrolyte	81.9	76.9	26.0 kg ⁻	92.	
Carbon	MoS ₂ drop casted on	2				98
cloth	functionalized carbon cloth;	at 0.			000	
(2021)	MoS ₂ /FCC electrodes	ы С П			er 1,	
	soaked with 1 M H ₂ SO ₄	525]			aft les	
	electrolyte	56.5 A g			29% cycl	

	1D shaped											
Substrate (Reporting year)	Device Configuration	Device	capacitance	Energy density	Power density	Capacitance	retention	Flexibility		Ref		
Cotton thread (2013)	Two PPy-MnO ₂ -CNT- cotton thread electrodes, separated with cotton textile wrapping with 0.5 M Na ₂ SO ₄ electrolyte, transparent silicone pipeline as a package shell	1.49 Fcm^{-2} at 1 mVs^{-1}	(electrode)	33 μ Wh cm ⁻² at 0.67 mW cm ⁻²	$13\ mW\ cm^{-2}$ at 14.7 $\mu Wh\ cm^{-2}$	87% after 2,000 cycles				99		
Cellulose yarns (2015)	Yarns (cotton, linen, bamboo, viscose) welded with activated carbon and twisting with stainless steel yarn	$120 \text{ F g}^{-1}, 37 \text{ mF}$	cm^{-1} at 2 mV s^{-1}			77% after 3,000	cycles	Some decay at	180° bent, curled, and crumpled	100		
Cotton yarn (2015)	Electroless deposition of Ni and electrochemical deposition of graphene on commercial cotton yarns (RGO/Ni cotton composite electrodes) with PVA-LiCl electrolyte and separator	0.11 F cm ⁻¹		6.1 mWh cm ⁻³	1,400 mW cm ⁻³	82% after 10,000 cycles		No decrease at 45° , 90° , 180°	bending, 95% after 4000 cycle at 180°	101		
Cotton thread (2016)	Twisting 2 strands of PVA-H ₃ PO ₄ gel electrolyte coated carbon nanoparticles /rGO-cotton thread (CNPs/rGO-CT) together	3.79 mF cm^{-3} at 50 mV	S ⁻¹	0.084 μWh cm ⁻³		95.23% after 10,000	cycles	92.30% after 2,000	bending cycles	102		

 Table D.5 Summary of Hybrid material-based supercapacitors or supercapacitor electrodes

Cotton	Twisting 2 strands of					s	50	102
thread	PVA-H ₃ PO ₄ gel	$V s^{-1}$.		ycle	guibu	
(2016)	electrolyte coated	2 m		n cm		00 c) ber	
	graphene	1 ⁻¹ at		nWh		r 8,0	r 50(
	hydrogels/MWCNT-	F cm		0 ⁻³ I	Vcm	afte	afte	
	cotton thread	73 µ		9×1	2 mV	51%	75% les	
	(GHs/MWCNTs-CT)	97.7		4.79	1.2	95.2	90. [°] cyc	
Cotton	Two cotton/Ni/Co-Ni							103
thread	layered double hydroxide	at 5	1 ⁻² at			s		
(2018)	(CT/Ni/Co-Ni LDH)	≥m ⁻²	F cn			ycle		
	hybrid yarn electrode	6 F (21 m	5	1 ⁻²	00 c		
	twisted together and	1.2	C 22 cm ⁻²	cm	V cn	r 2,0		
	painted with PVA-KOH	trode	s ⁻¹ , S mA	aWh	Vm (afte		
	gel electrolyte	Elect	mV s 0.04	9.3 n	43.99	79%		
Cotton yarn	Cotton electrode by dip-							104
(2018)	dried in MWCNT							
	followed by interfacial							
	polymerization of PPy,				-			
	two parallel electrodes			Vhg	M BB			
	embedded in a thin layer	ao ⁻⁷		3 mV	33 m			
	of PVA-H ₃ PO ₄ layer	30 I		2.63	11.3			
Cotton	Short-staple length SSFs							105
fibres (2018)	blended with cotton fibres							
	to spin SSF/cotton							
	blended yarn, PPy							
	deposited on PEDOT:PSS							
	coated composite yarn,					es		
	followed by coating with					cycl		
	PVA-H ₃ PO ₄ electrolyte,			:m ⁻²		000		
	placed in parallel and	m^{-2}		Wh c		er 5,		
	twisted together, followed	5Fc		6 m/		6 0V		
	by coating	1.3		0.1		80%		

Cotton (2019)	GO nanosheets (NSs) modified with ultrathin and large area MoS ₂ NSs followed by reduction with PVA-H ₃ PO ₄ electrolyte	$134.38 \text{ Fg}^{-1}, 332.85 \text{ mF}$	cm^{-2} and 221.9 F cm^{-3}	at a current of 50 mA					100% when bent by 30	and 60 degrees	106
Cotton fibre (2021)	In-situ growth of PPy and MXene composite, on cotton fibre to prepare fibre electrode	506.6 F $\mathrm{g}^{-1},$ at 1 A g^{-1}	and $455.9 \text{ mF} \text{ cm}^{-2}$ at	$0.9 \mathrm{~mA~cm^{-2}}$			83.3% after 2,000	cycles			107
Polyester fibre (2018)	PPy electrochemically deposited on rGO painted SnCl ₂ modified polyester yarn electrode with PVA- H ₂ SO ₄ electrolyte for SC	Electrode 175.7 mF cm ⁻¹ , 699.6 mF cm ⁻² , 239.6 F	$\rm g^{-1}, 35.0 \ F \ cm^{-3}$ at 0.13 mA cm^1, device 85.3 mF	cm^{-1} , 339.7 mF cm^{-2} , 116.4 Fg ⁻¹ , 17.0 F cm^{-3}	0.0472 mWhcm ⁻²	26.5 mWcm ⁻²			Unchanged after 1,000 bending cycles		108
Polyaniline fibre (2013)	Two PANI composite fibres incorporated with aligned MWCNT twisted	274 Fg ⁻¹ , 263 mF	$\rm cm^{-1}$ at 2 A $\rm g^{-1}$				99% after 1,000	cycles	>97% after 50	bending cycles	109

Nylon fibre	MWCNT helically									110
(2015)	wrapped around nylon	8 F					ring			
	fibres, followed by	⁻² , 3,					é du			
	electrochemical	F cm					50%			
	deposition of MnO ₂ . Two	6 m]	-				rain,			
	coiled MnO ₂ /CNT/nylon	, 40.	-S VI	-7	-2		% st			
	fibre electrodes placed	cm ⁻¹	10 m	h cm	V cm		at 12	ain		
	parallel and coated with	mF	³ at	μWJ	ν μ 9		8% 8	ge sti		
	PVA-LiCl gel electrolyte	5.4	cm	2.6	66.		90.	larg		
Elastic fibre	Elastic fibre/CNTs/PANI	n ⁻¹				es	les			111
(2014)		ıF cı				cycl	cyc			
		19 n		-		000	,000	$^{\circ}0$		
		-1, 0.		h kg	kg ⁻¹	r 10	ter 1	at 18		
		$5 F_{g}$	A g-1	5 W.	t W	afte	% af	ling		
		255.	at 1	12.7	149	%69	93.8	benc		
Shape-	Wrapping aligned CNT	Ц								112
memory	sheets onto shape-	.3 m								
polyurethane	memory polyurethane	ld 42				SS	cles			
(SMP)	(SMP) substrate as	1 ⁻¹ ar				cycle) cyc			
substrate	electrode, coated with	F cn				000	r 50(
(2015)	PVA gel electrolyte	69 m				r 12,	afte			
	followed by winding	0.26				afte	ease	ation		
	another layer of aligned	Fg_1,	ς			loss	decr	ormâ		
	CNTs as outer electrode	24]	cm			No	No	def		
Urethane	CNTs dipping and PPy							at		113
stretchable	electrodeposition on	3m ⁻²						ged (ain	
yarn (2016)	urethane elastic fibre core	mF c					arly	hang	% str	
	spun yarns (UY)	69					Neć	unc	809	

Stretchable	Electrochemical							114
substrate	activation of pristine							
(2017)	CNT fibres, coating of							
	PEDOT:PSS followed by						S	
	electrochemical						ycle	
	deposition of MnO ₂ to						000 0	
	form						r 3,0	
	MnO ₂ @PEDOT:PSS@O						in fo	
	CNTF positive electrode			cm ⁻²			stra	
	and hydrothermal			μW			%0 (
	synthesis of MoS ₂ to			540			at 1(
	form MoS ₂ @CNTF			² at			ning	
	negative electrode with	n^{-2}		ı cm			retch	
	LiCl-PVA electrolyte	LF CT		4Wh			er st	
	placed on stretchable	.6 m		.37			6 aft	
	substrate	278		125			929	
Carbon	PANI nanowire arrays in-						°	115
nanotube	situ deposited on CNT	yarn	NT				, 18(
(CNT) yarn	yarn, PVA gel coated on	ANI	ire C				135°	
(2013)	CNT yarn or CNT/PANI	T/P/	² , pu F cm				00,00	
	yarn, two CNT or	CN	cm ⁻ .3 m			/cles	5°, 5	
	composite yarns twisted	cm	mF SC 2)0 cy	at 4	
	together	mA	C 38 sed 5			er 8(ease	
		0.01	ed S n-ba:			6 aft	ding	
		At (basi yarı			919	No ben	
Carbon	Electrochemical							116
nanotube	deposition of MnO ₂ onto	/ s ⁻¹					000	
(CNT) yarn	CNT yarn, two PVA-	/m (sr 1,(
(2014)	KOH coated CNT/MnO ₂	at 1(:m ⁻³	1^{-3}		afte 0°	
	composite yarns placed	m^{-3}		Wh c	N cn		ease ; at 9	
	on top of each other, and	4 F c		2 m/	m		deci	
	coated by PVA-KOH	25.4		3.5′	127		No ben	

Carbon	MnO nanosheets in situ					11	17
nanotube	grown on CNT fiber					lat	
CNT fibre	(positive), polyimide		2 ⁻² ,	cm ⁻²		ngec 80)	
(2017)	deposited on CNT fiber		JW G	Wh c		ichai (0-1	
	(negative electrode)		78 n	.2 μ'	les	ns ur	
			at 0.	ut 30) cyc	s deg	
			cm ⁻²	m ⁻² 8	,000	es re rious	
			Wh c	M C	pto 2	g vai	
			-4 μ'	.6 m	lu lle	paci	
			36	15	We	Ca bei	
Carbon	Twisting CNT yarns (n)	- - -				S 11	18
nanotube	with Pt filament as	.7 F			00	nd sycle	
(CNT) yarn	current collector, PANI	217 -1	V kg		5,00	ing c	
(2017)	nanowires further	SSC 2 Ag	88.		after	cycl flex	
	deposited in-situ to form	NI F. at 0.5	lt 91		lges	000,	
	the Pt/n-CNT @ PANI	¢PA] m ⁻³	6 ⁻¹ 3		chan	er 5 er 5	
	with alkaline electrolyte	NT @ nF c	Vh k		ible (6 aft 6 aft	
		5-CJ 27 n	22 V		gligi cles	.17 9	
		Pt/ 48.	30.		Ne cyc	98. 95.	
Carbon	Coating electroactive	2_	g_1	-1	S	ອ ສູ່ມາ	19
nanotube	MoS ₂ nanoflakes on a	⁻ g ⁻¹ , F cm	⁷ h K	Wkg	ter ycle	ay 000 sendi	
(CNT) fibre	CNT fibre backbone with	2 m]	4 W	00	% af 00 c	dec er 1,	
(2021)	rGO as adhesion layer	19(93.	26.	4,0	85 ⁹ 5,0	No afte cyc	
Carbon	Spun yarn composed of					12	20
nanotube	SWCNTs and PANINWs,					135	
(CNT) and	coated with PVA-H ₂ SO ₄	∆ g ⁻¹				,°00	
Polyaniline	electrolyte, twisted	0.6 /			vcles	5°, 5	
Nanowire	together	-2 at	-2	5	00 cJ	g g	
(PANINW)		cm,	h cm	cm	er 8(ease	
composite		7 mF	μWl	Ψų	ó aft	decr º° be	
yarn (2014)		2.6	0.8	150	86%	No 180	

Carbon	Yarn electrodes by										121
nanotube	biscrolling MXene with	523		n^{-2} ,		n^{-2} ,					
(CNT)	CNTs, freestanding	cm ⁻³ ,		Vh cı		W cī					
/MXene	asymmetric yarn SC	3 F		8 μV		8 m		ycle			
yarn (2018)	prototypes by pairing	1,08	n^{-2}	3, 16		°, 14,)th c			
	with biscrolled RuO ₂	:m⁻²,	LA CT	cm	n^{-1}	cm_	\mathbf{n}^{-1}),000			
	yarns with 3.0 M H ₂ SO ₄	mF c	t 2 m	۱Wh	Vh cı	шW	V cn	at 1(
	electrolyte	188	g ⁻¹ af	.6 n	4 μV	428	⊦1 μV	%06			
		З,	Щ	61	×.	5,	74	5́~			122
Graphene	2 intertwined hierarchical	at							s		122
fibre (2013)	hybrid core–sheath fiber	cm ⁻¹							ycle		
	electrode [a core of	μF							00 c		
	graphene fibre (GF)	¹ , 20							ng 5		
	covered with a sheath of	Ч 1		m^{-2}		5		/cles	endi		
	3D porous network-like	5-40		Vh c		cm		00 c)	ter b		
	graphene framework,	- ² , 2,	0	0-7		-6 W		er 5(se afi		
	denoted as GF@3D-G],	t cm	cm ⁻²	7 imes 1		$\times 10$		r aft	creas		
	solidified in the H ₂ SO ₄ -	7 mF	μĄ	4-1.7		100		mila	o dec		
	PVA electrolyte	1.7	17	0.4		-9		Siı	ž		
Graphene/P	Intertwining two G/PPy	24								iles	123
Ру	electrodes pre-coated	at 0.2							fter	cyc	
composite	with H ₂ SO ₄ –PVA	m ⁻² 8		cm ⁻²					ase a	guib	
fibre (2014)	polyelectrolyte	nF ci	m^{-2}	Wh e					ecrea) ben	
		07 n	nAc	μ 7.0					Vo de	,000	
GO/MXene	Wet spinning of GO	-	<u>н</u>	5					~	Π	124
(~88 wt%)	liquid crystal-assisted	:m ⁻² ,	, 341								
fibre (2017)	MXene fibre with 1 M	nFc	οο σο								
	H ₂ SO ₄ electrolyte	233 1	257] ⁵ cm								
			. –			1			1		1

rGO/MXene	Electrolyte Mediated	at									125
hybrid fibre	hybrid fibre made of rGO	a		cm ⁻³	-3						
(2020)	and MXene, assembled	89 F		Wh e	V cm						
	into fibres via wet	110.3		5 m'	7.1 V						
	spinning with PVA-	and		1 9.8	2 pure		/cle				
	H ₂ SO ₄ electrolyte	m^{-2}		² and	m^{-2} ()0 cy				
		nF c	-	cm	W c		10,0(
		196.	nV s	۲Wh	.8 m		o at				
		550	201	12	at 8		85%				
rGO/10 wt%	Wet spun rGO/10 wt%										126
MoS_2	MoS ₂ composite fibre	Щ		νh		8					
composite	with PVA- H ₂ SO ₄	.3 m	0	8 mV	ς.)36 V					
fibre (2021)	electrolyte	185	cm	3.38	cm	0.0	CIII				
rGO/20 wt%	Wet spun rGO/20 wt%					3					126
MoS_2	MoS ₂ composite fibre	Ц		Λh		/ cm	ter	cles			
composite	with PVA- H ₂ SO ₄	.6 m	4	2 mV	ς	51 W	% af	0 cy			
fibre (2021)	electrolyte	282	cm	4.9	cm	0.0	87.	100			
MXene	Wet spun fibres using	, ,								in	127
(70%)	hybrid formulations of	cm	m ⁻³				0		lly	stra	
/PEDOT:PS	Ti ₃ C ₂ T _x MXene	í mF	Γc			n^{-3}	00,0		clica	%00	
S hybrid	nanosheets and	676	, 615	h cm		W ci	er 1		r cyc	to 1	
fibre (2019)	PEDOT:PSS with PVA-	trode	г о	3 M]		-9 m	6 aft		afte	ched	
	H ₂ SO ₄ electrolyte	Elect	258]	≈7.1		≈824	~959	cycle	96%	streto	
Carbon fibre	Activation of pristine										128
tow (CFT)	CFTs by oxidative								5°		
(2015)	exfoliation by								, 13		
	KMnO ₄ /H ₂ SO ₄ , annealing								t 90°		
	in air and reduction by a								ng a		
	mixture of hydrogen								endi		
	iodide (HI) and acetic	s ⁻¹					les		00 b		
	acid (AcOH), twisting	Vm () cyc		r 1,0		
	two activated electrodes	at 10		m^{-3}		m^{-3}),00(afte		
	together after coating	m ⁻³ (Vh c		M c	er 10		ease		
	with PVA-H ₃ PO ₄	5 F c		5 mV		0 m	5 aft		decr	tting	
	electrolyte	2.55		0.35		300	91%		No	kno	
Carbon fibre	Manganese oxide						129				
--------------	--	-------------------	---------------------	---------------------	-------	------------------------	-----				
(2015)	nanosheet grown on										
	carbon nanoparticle					, 36(
	coated carbon fibre					180°					
	(CF@CNPs) and					g at					
	functionalized CF@CNPs	<i>с</i> у			cles	guibr					
	are employed as the	cm			00 cy	r ber					
	positive and negative	mA	1 ⁻³	ά	0,00	afte					
	electrode respectively	at 2	h cm	cm'	fte 1	ease					
	with LiCl-PVA	cm ⁻³	mW	22 W	2% a	decr					
	electrolyte	5 F	2.1	10.2	81.2	No					
Carbon fibre	PPy deposition on MnO ₂						130				
(2013)	nanoflakes coated carbon										
	fibre (CF/MnO ₂ /PPy),										
	two PPy-MnO ₂ -CFs fixed										
	on preservative film	2m ⁻³			cles	d					
	substrate and assembled	1 A 6			0 cy	n gu					
	into SC by sandwiching	at 0.	m ⁻³		1,00	rolli					
	PVA-H ₃ PO ₄ membrane as	m ⁻³ 8	Vh c	cm ⁻³	fter	fter					
	separator and electrolyte	3 F C	5 mV	M t	7% a	3% a					
	between electrodes	69.3	6.16	0.0	86.7	3.66					
Carbon fibre	Carbon fibre bundle @						131				
(2016)	CNT-NiCo(OH)x		n ⁻³			50 5					
	(CF@CNC) as positive	n ⁻¹)	h cn	cm		nding afte					
	and carbon fibre bundle	LF CT	шW	mW	cles) ber ntion					
	@ activated carbon	.0 m).84	9.1	0 cy	1000 reter nes					
	(CF@AC) as negative	, (64	n ⁻² , (n ⁻² ,]	8,00	lfter 17% ng tii					
	electrode, immersed in	cm ⁻²	/h cr	V CL	fter	ay a d 10 ristir					
	PVA-KOH electrolyte,	mF	Μμ	m i	% a	dec s an 00 tw					
	dried, twisted together	111	33.0	0.75	103	20% time 1,00					

Carbon fibre	Braided carbon fiber						132
(2017)	electrodes coated with	·m ⁻²			cles	ad	
	MWNT/V ₂ O ₅ nanowires	nA c			0 cyc	ndin	
	(NWs), a cellulose-based	0.5 r			00,0	0 be	
	separator, and an ionic-	² at			>1	1,00	
	liquid electrolyte of	^r cm			after	ifter	
	[EMIM][TFSI]/LiCl/Al ₂	5 mF			%0	7% a les	
	O ₃ nanoparticles	10.6			~ 10	98.3 cyc	
Carbon fibre	Sheath-core PANI						133
yarn (CFY)	nanowire array grown on						
(2018)	aligned carbon						
	nanofibres/ carbon fibre						
	yarn electrode	m ⁻²					
	(CFY@CNFs@PANI	ıA c			es		
	NWA), two fibre-shaped	0.1 n			cycl		
	electrodes parallelly	² at (m^{-2}	cm ⁻²	000		
	placed on PET substrate	cm	/h cı	Mu	er 8,		
	and immersed in PVA-	mF	4 μW	.52 1	ó aft		
	H ₂ SO ₄ electrolyte	234	21.4	at 0	606		
Carbon fibre	Dip-coating of					e	134
(2019)	mixture of ionic liquid 1-					im (
	ethyl-3-					at 6(
	methylimidazolium					tion	
	bis(trifluoromethylsulfon					eten	
	yl) imide					lce r	
	([EMIM][TFSI]),					citar	
	carbon nanotubes, and					capa	
	electro-polymerization of					6%	
	PPy onto Au coated					, 95.	
	carbon fiber with	cm ⁻²				vcles	
	propylene carbonate-	mA				lg cy	
	poly(methyl	t 0.6				ipua	
	methacrylate)-	n ⁻² at	m^{-2}	1 ⁻²)0 bé	
	[EMIM][TFSI] gel	LF cn	/h cı	V cn		er 1(
	electrolyte to form wire-	49 m	7μW	2 mV		6 aft vater	
	type SC	38.4	24.3	3.52		99% w ni	

Carbon fibre (2020) Platinum	PEDOT:PSS- rGO drop coating and MnO ₂ electrodeposition on carbon fibre, MnO ₂ /PEDOT:PSS-rGO and PEDOT:PSS-rGO as positive and negative electrodes with Na ₂ SO ₄ - CMC electrolyte PEDOT/MWNT	Electrode 2.92 F cm ⁻² (194 F cm ⁻³ , 550 mF cm ⁻¹) at 5 mA cm ⁻²	295 μWh cm ⁻² (19 mWh cm ⁻³ , 55 μWh cm ⁻¹)	2900 μW cm ⁻² (190 mW cm ⁻³ , 545 μW	96% after 5,000 cycles	135
wire (2013)	biscrolled yarn with Pt wire with liquid electrolyte (H ₂ SO ₄) or, with 1 M H ₂ SO ₄ /PVA gel electrolyte.	At 0.01 Vs ⁻¹ , ~ 167 F cm ⁻³ (liquid electrolyte), ~ 180 F cm ⁻³ (solid).	1.4 mWh cm ⁻³ (Solid)	40 W cm ⁻³ (Solid)	98% after 2,000 cycles	98% after 2,000 bending, 92% after 10,000 winding, 99% after 10,000 cycles woven into gloves
Platinum yarn (2014)	PANI nanowire solution coated on CNT wrapped Pt yarns (Pt wire/CNTs/PANI) with PVA-H ₃ PO ₄ electrolyte	$86.2 F g^{-1}$, 0.24 mF cm ⁻¹ , 52.5 mF cm ⁻² at 5 mV s ⁻¹	35.27 Wh kg ⁻¹	10.69 kW kg ⁻¹		No decrease after folding/unfolding 1,000 cycles
Au wire (2018)	Two wire electrode prepared by layer-by- layer (LbL) assembly of MWCNTs, vanadium oxide (VO _x), wetted with organic electrolyte of propylene carbonate (PC)–acetonitrile (ACN)–lithium perchlorate (LiClO ₄)– PMMA and twisted together	5.23 mF cm^{-2} at 0.2 mA cm $^{-2}$	1.86 μ Wh cm ⁻²	$8.5 \mathrm{~mW~cm^{-2}}$	94% after 10,000 cycles	138

Stainless steel yarn (2015)	Two PPy-wrapped Fe ₃ O ₄ deposited SS magnetic electrodes coated with PVA-H ₃ PO ₄ gel electrolyte, dried, twisted and coated with PU	61.4 mF cm ⁻² at 10 mV	s^{-1}					77% after 1,000 cycles		Self healing 71.8% at	4 breaking &	reconnecting	139
Stainless steel fibre yarn (2015)	PPy@MnO ₂ @rGO- deposited conductive yarns as both active materials and current collectors, two parallel yarn electrodes coated with PVA-H ₃ PO ₄ electrolyte	36.6 mF cm^{-1} & 486 mF cm^{-2} in aqueous Na ₂ SO ₄	electrolyte (3-electrode cell) or 31 mF cm $^{-1}$ and	411 mF cm ⁻² in PVA / H_3PO_4 (2-electrode cell)	SS SC 0.0092 mWh cm ⁻² & 1.1 mWh cm ⁻³	5 111 U 1 C 111 C 1	1.25 mw cm ⁻ , 100 mw cm ⁻	SS SC 92%	over 4,950 cycles	80% after 1,000 cycles at 90 bending, 91% after	1,000 cycles knotting, 103% after 1,000 cycles	twisting	140
Stainless steel spring (SSS) (2018)	In-situ synthesis of hierarchical carbon tubular nano structures (hCTNs) and PANI composites onto SSS substrate to form stretchable SC electrode	277.8 F g^{-1} at 1 A $g^{-1},$ and	$402.8\ \mathrm{mF}\ \mathrm{cm}^{-1}$ at $1\ \mathrm{mA}\ \mathrm{cm}^{-1}$					75% over 3,000 cycles		100% stretchable			141
Stainless steel filament (2020)	Core-sheath single yarn produced by PVA-H ₂ SO ₄ electrolyte-mediated rGO & MXene fibres twisted around the SS filament core, two-ply yarns made from plying of two single core-sheath yarns, two yarn electrodes plied to obtain SCs	Dual-core YSC 253.01 mF cm ^{-2} (43.6	mF cm ⁻¹ at 20 mV s ⁻¹)		27.1 μWh cm ⁻² at 2502.6 μW cm ⁻²		$2,302.0 \mu$ w cm ⁻² at 27.1μ Wh cm ⁻²	18% deterioration after 10,000 cycles		90% after 1,000 bending cycles			142

Stainless steel filament (2020)	Hydrothermal deposition of nickel-cobalt oxide on stainless steel cables, stainless steel cables @nickel-cobalt oxide	<10 ⁻³ mAh cm ⁻¹ at	nA cm ⁻¹					143
	electrode	113>	0.3 r					
		 	2D shape	ed			1	
Substrate (Reporting year)	Device Configuration	Device	capacitance	Energy density	Power density	Capacitance retention	Flexibility	Ref
Cotton	Deposition of MnO ₂ after							42
fabric	dipping and drying of					nge cles		
(2010)	cotton with SWNT ink					chai 0 cy		
	electrodes with 2M	¹ cm ⁻²				gible 15,00		
	electrolyte	.41 F				Jegli ver 3		
Non-woven	Dip coating of nonwoven	0				Z 0		144
cloth (2011)	wiper cloth in the						ас	
	SWCNT ink followed by					ş	ndin	
	deposition of PANI					cycle)° be	
	nanowire arrays on to	l A g				000	at 9(
	obtain the	¹ at]		ı kg ⁻¹	'Kg ⁻¹	er 3,(ease	
	PANI/SWCNT/ cloth	го ГЦ		5 Wb	00 M	ó afte	decr	
	composite electrode	390		26.0	7,00	606	No	
Cotton t-	MnO ₂ / ACT (activated							145
shirt (2012)	carbon textiles from							
	cotton t-shirt) as positive,							
	ACT as negative							
	electrode, copper foil as	n^{-2}				ycles		
	current collectors, and	ıA cı				00 0		
	Whatman filter paper	1 m				r 1,0		
	soaked with 1M Na ₂ SO ₄	g ⁻¹ at		/h kį	W k	afte		
	aqueous solution as the	0 F 5		N.7 W	97 K'	.5%		
	separator	12		99	4.5	76		

Cotton	Electrochemical						146
stretchable	deposition of PPy on pre-	-				train re	
knit fabric	strained Au coated cotton	IV S)% s ailu	
(90%	fabric with 1.0 M NaCl	10 m				o 14(trric f	
cotton/10%		⁻¹ at				up t elec	
Lycra)		а Ц				tain hout	
(2013)		255				Sus wit	
Cotton	Electrodes prepared by						147
(2015)	dip coating of cotton on		hA				
	GO dispersion, thermal		0.6 r				
	reduction, and PPy		¹ at (
	deposition. SC device	÷	kg				
	with 2.0 M NaCl aqueous	່ວວ 	Wh 2				
	solution electrolyte	336	21.1 cm ⁻				
Cotton	PPy coated cotton fabric						148
(2015)	electrode through in situ	m^{-2}					
	chemical polymerization	nA c			ycles		
	by using CuO	0.6 r			00 ci		
	nanoparticles template,	⁻¹ at			er 2(
	with 2.0 M NaCl	ы Д			6 aft		
	aqueous solution	225			92 9		
Cotton	PPy conductive polymer						149
fabric	coated on top of MnO ₂					y ífter s	
(2015)	nanoparticles deposited-					pacit 3% a ycle	
	CNT textile with					о са <u>ј</u> 96.3 00 с	
	PEO/Na ₂ SO ₄ gel					n, n rain, 50,0	
	electrolyte				s	strai ig sti of 7	
					ycle	isile indir ling	
					00 c	6 ten % be benc	
			-	÷	10,0	21% 139 clic	
		-	h kg	V kg	over	upon ug cy	
		50 LI	1 W]	1 kV	8% (5% 1 nge vosin	
		461	31.	22.	93.	98. cha imp	

Cotton	Binder-free ternary				S		S							150
fabric	composites of MnO ₂	SS			T:PS		T:PS							
(2015)	nanoparticles, SWNT,	T:PS			[DO		DO							
	PANI and PEDOT:PSS	EDO			T/PE		T/PE							
	deposited layer-by-layer	[]/T			MN:		MN							
	by dip coating to prepare	MMS			02/S	1	02/S	-						
	cotton electrode with gel	02/5	-		Mn(hkg ⁻	Mn(Vkg ⁻			80°			
	electrolyte composed of	Mn	$6 F_{g}$		kg¹,	2 W	kg ⁻¹ ,	0.5 V			of 1			
	cetonitrile (ACN):	Fg'.	: 24		Wh]	:: 60.	5 W]	: 64(ungle			
	propylenecarbonate (PC):	294	vices		66.4	vices	746.	vices			ing a			
	PMMA: tetra-	N:	e dev		NI:	e dev	NI:	e dev			bend			
	butylammonium	Γ/PA	osit		Γ/PA	osit	Γ/PA	osit			oat			
	hexafluorophosphate	MN	imo		WN	funo:	MN	somp			wn t			
	(TBAPF6) in a ratio of	S∕2C	ary c		S/2C	ary c	D₂/S	ary c			ó, do			
	70:20:7:3 by weight	Mn(tern		Mn(tern	Mn(tern			<3%			
Cotton	Direct electrospinning of													151
fabric	MWCNTs on nickel-	Ц	2.5	\dot{c}^2					ıy	000				
(2016)	coated cotton fabrics (Ni-	.5 m	⁻² at	_ cm_					deca	х 3,(
	cotton)	973	cm	mA					N_0	afte				
Cotton	CNT/rGO-coated fabric								S		ler	th		152
fabric	as negative electrode and								sycle		/ nuc	s wit		
(2017)	PPy-coated fabric as the	at			n^{-2}				000 c		bility	time	80°.	
	positive electrode	cm^{-2}			/ hcn				r 1,0		t stal	100	of 1	
		mF (A cm		тW				afte		ellen	ling	ngle	
		570	1 m		0.26				91%		Exce	benc	an a	
Cotton	PPy electrochemically		0											93
fabric	deposited on the surface	Σ	43.2(
(2018)	of dip- dried cotton into	th 1	n) 3,			-	3 Mr							
	MXene $(Ti_3C_2T_x)$	(wit	lutio			h g'	1.1 n							
	nanosheets with	rode) 4 SO		ce	тW	ce 4							
	H ₂ SO ₄ -PVA electrolyte	Elect	H ₂ SC	ao T	Devi	1.30	Devi							
		_	_	_							1			

Cotton	PPy/TiO ₂ -coated cotton	t								153
fabric	fabric electrode via sol-	$g^{-1}a$								
(2018)	gel and in-situ chemical	3 F			¹ at					
	oxidation method with	le 73	1^{-2}		l kg	aa^{-}				
	2.0 M NaCl aqueous	ctrod	A cn		ł Wł	Wk				
	electrolyte	Elec	0.6		44.4	555				
Cotton	GO deposited by 'dip and									154
fabric	dry' method, chemically	Ц								
(2020)	reduced into rGO/cotton	444								
	fabric, MnO ₂	ω ⁻								
	nanoparticles	25 A								
	accumulated on	and								
	rGO/cotton fabric by in	a^{-1}						∫ g ^{−1}		
	situ chemical deposition,	t 1 A						15 /		
	PANI layer coated on	5 ⁻¹ at						es at		
	rGO/MnO ₂ /cotton fabric	2 F g						cycle		
	by in situ oxidative	d 25.						000		
	polymerization technique	-1 an						er 3,(
	with 1 M H ₂ SO ₄	പ പ	5					afte		
	electrolyte solution	888	cm					70%		
Cotton	PPy/rGO nanocomposite		-2							155
fabric	cotton fabric (NCF) by	-	н сш		m^{-2}		m^{-2}	_	00	
(2021)	chemical polymerization	NCF	0 m]		Wh c		ιW c	ntior	10,0(
		GOJ	9,30	m^{-2}	7 μV		20 n	rete	fter	
).5/r ^r	ode	лА с	ce 16		ce 1.	rode	% a:	
		Py-(lectr	t l n)evic		Jevic	llecti	4.47	
		Ц	e	а	П		L	Щ	6	

Cotton	GO nanosheets fixed on									156
fabric	the cotton fabric by	C								
(2022)	vacuum filtration, pyrrole	-3, S								
	monomers and silver ions	cm							S	
	(Ag+) adsorbed on the	7.0 I							ycle	
	surface of GO/CF by π - π	$ ^{-2}, 2$	-3)						ing c	
	and electrostatic	F cm	F cm						endi	
	interactions, respectively	0 m]	4.7]		n^{-2}	m^{-2}			9 00 P	
	to form flexible	664.	n ⁻² (h cn	LW C			10,0	
	PPy/Ag/GO/CF	le 1,	uF cr		Ψų	9.5 µ			ufter	
	electrodes with PVA-	ctroc	.6 m		25.5	1149			7% а	
	H ₂ SO ₄ gel electrolyte	Ele	286		SC	SC			89.	
Bamboo	MnO ₂ -NiCo ₂ O ₄ printed	<u> </u>								157
fabric	bamboo fabric as positive	ц Ц								
(2020)	electrode, and rGO	766				1 ⁻³	;000			
	printed bamboo fabric as	² (1,	m^{-2}		cm ⁻³	Wcn	ter 5			
	negative electrode with	Fcm	ıАс		лW	4 m	of af	S		
	LiCl-PVA gel electrolyte	2.12]	at 2 n		37.8 1	2678.	92%	cycle		
Polyester	Graphene/MnO ₂ -textile		•				es			158
fabric (2011)	as positive and SWNTs-		tiles				cycl			
	textile as negative	hene	d tex	g^{-1}	Ţ.	_	,000			
	electrode in aqueous	grap	base	5 F	h kg	kg ⁻¹	er 5,			
	Na ₂ SO ₄ electrolyte	brid	10^{-1}	to 31	5 WI) kW	% aft			
		Hy	M	dn	12.	110	959			
Polyester	MnO ₂ nanoflowers	/ s ⁻								159
fabric (2011)	electrodeposited onto	Vm (00			
	CNT-enabled conductive	0.05					10,0			
	textile fibres, MnO ₂ -	l ⁻² at					after			
	CNT-textile as positive	cm					3 %0			
	electrode, reduced MnO ₂ -	2.81					es, 6	<i>i</i> cles		
	CNT-textile as negative	and					cycle)0 cy		
	electrode, Whatman filter	V_{S} -1				_	200 (50,00		
	paper as separator and 0.5	5 m			kg ⁻¹	/ kg ⁻	irst	fter :		
	M Na ₂ SO ₄ in water as	t ⁻¹ at			Wh	M 00	fter f	% al		
	electrolyte	$0 F_{g}$			5-20	13 0(% al	d 50		
		41	1		²	2	80	an		

Polyester	Directly growing CNTs												160
film (2013)	along graphene fibres	1 ⁻²											
	with Fe ₃ O ₄ nanoparticles	A cn								cles			
	for CVD of the	0 m.								g cyc			
	nanotubes, PET films	at 2								guibu			
	coated with Au layer as	cm ⁻²) ber			
	supporting substrates and	mF								r 20(
	current collectors, filter	0.98								afte			
	paper separator soaked	^{г_1} , (ease			
	with 1 M Na ₂ SO ₄	.4 F								decr			
	aqueous electrolyte	200								No			
Polyester	CNT dip-coated onto Cu												48
fabric	metallized PET fabrics	<10 ⁻³	n^{-2}										
(2014)	electrode	8.5>	Fcr										
Polyester	PANI film deposited on	5							S		e		48
fabric	the CNT/Au/PET; with	cm	ode),		10^{-4}	n ⁻³		ter	sycle	ent	nanc		
(2014)	PVA-H ₃ PO ₄ electrolyte	ШF	ectro	$\mathbf{\hat{N}}$	t12×	h cn		% af	500 c	celle	rforr	der	
		7.1	(el	9	2.4	M		89	2,5	Ex	pe	un	
Polyester	Strip shaped electrode	Ц	at	e),				0		ų			161
fabric	prepared by PANI	43.6	m ⁻² ;	trod				0,00		ersio	ų		
(2015)	deposited on aligned	-3 (3	ΕLC	elec	cm ⁻³		9	in 1(mme	48		
	CNTs, with PVA-H ₃ PO ₄	t cm	47.8	m ⁻³ (Wh		cm	with		ter i	r foi		
	gel electrolyte	1.7 F	and	Ac	6 m		1 W	ble	cles	% af	wate		
		42	ao' ⁻	0.5	6.5		2.9	Sta	cyc	816	in		
Polyester	Two pieces of MnO ₂												162
fabric	deposited conductive	t 2								0	а		
(2016)	graphene/polyester	$g^{-1}a$								зr 10	ndin		
	composite fabric with	8 F	\mathbf{s}^{-1}							afte	s bei		
	PVA-NaNO3 electrolyte	265.	mV							87%	time		

Polyester fabric (2018)	PPy electrochemically deposited on rGO painted SnCl ₂ modified polyester textiles with PVA-H ₂ SO ₄ electrolyte	Electrode 1,117 mF cm ⁻² at 1	mA cm ⁻² , 329.5 F g ⁻¹ at 1 mA cm ⁻¹ . Device 474 mF cm ⁻²	$0.0658 \text{ mWh cm}^{-2} \text{ at } 1 \text{ mA cm}^{-2}$	$25 \mathrm{ mW cm}^{-2}$	100% after 10,000 cycles	98.3% after 1000 bending cycles	108
Polyester	Dipping and drying of	at						163
fabric	PET with GO, followed	cm^{-3}						
(2019)	by chemical reduction of	.5 F c						
	chemical growth of PPv	;-1, 5						
	on PET, composite	mVs				ycles		
	electrodes of	at 1			2 -	000 c		
	PET/rGO/PPy	cm^{-2}	cm^{-3}	cm^{-2}	/ cm	er 6,0		
	sandwiched using PVA-	mF (mA 6	۱Wh	3 m W	ó afte		
	H ₂ SO ₄ gel electrolyte	230	1.6	11 µ	0.03	76%		
Polyester	Pre-treated polyester dip							164
fabric	coated in GO, followed	a'-						
(2020)	by reduction of GO and	mF	<i>c</i> ₁					
	in situ polymerization of	,300	cm					
	PPy particles on fabric	ode 8	0 ml					
	surface with aqueous 1 M	ectro	d 64					
DI	H ₂ SO ₄ solution	E	an					165
Polyester	Silver paste printed on						00	105
fabric (2020)	PET, MnHCF-MnOx/GO						ng to for 1	
(2020)	ink overprinted and				n^{-2}		endii 80°	
	alactrode DVA L:C1	m^{-2}		cm^{-2}	W cn		ile buund 1	
	electrolyte and paper	nF c		Whe	3 m/		e wh: 0°, a	
	separator	16.8 I		0.5 m	0.002		Stable 60°, 9 cycles	

Silk fabric	Screen-printed current									166
(2016)	collector and active						es			
	material layers [MnO-) tim			
	coated hollow carbon						r 100			
	microspheres, acetylene						afteı			
	black, and binder) mixed						.8%			
	in a weight ratio of 7:2:1						96 p			
	as printing ink] on silk						g an			
	fabrics substrate and a	m^{-2}					ndin			
	PDMS film; after being	nАс					s be			
	pasted with gel	t 1 n				ş	time			
	electrolyte, the PDMS-	n^{-2} a				sycle	100			
	based electrode	F cn				00 c	fter			
	transferred on the top of	23 m			<u>`0</u>	r 2,0	5% a	sting		
	silk fabric electrode	19.2			84%	afte	98.4	twis		
Nylon/PU	Supersonic spraying	5	2							167
(67/33)	rGO/SnO ₂ on fabric with	cm	cm			cles				
(2021)	2 M KOH electrolyte	mF	mA		after	00 cy				
		,00	it 1.5		3%	0,00				
Stretchable	Fully printed	-	0		0,	-				168
textiles	Ag@PPv@MnO ₂ on Ag			² at						
(2018)	cathode electrode and	7-		cm ⁻¹	,000		%0	ain		
	activated carbon on Ag	cm	_	Wh	ter 5		ter 4	g stra		
	anode electrode with	3 mF	(apo	37 m mW	% af	S	% af	ching		
	PVA-Na ₂ SO ₄ electrolyte	126.3	cath).03().38	90.8	cycle	36.29	strete		
Spandex	Assembling vertical PPy	7	<u> </u>	0.0		•	~			169
fabric	nanotube (VPPyNT)						ng	50%		
(2019)	grown on the carbon nano						stchi	n of	\ 0	
	onions (CNO) @ PPy				cles		r stre	strai	100%	
	granula (PPyG)-textile				10 cy		afteı	at a	lof	
	electrodes with a PVA-			- - -	1,00		tion	cles	trair	
	H ₃ PO ₄ electrolyte	1-1		h Kį	ufter		eten	0 cy	ıt a s	
	sandwiched structure	4 F g		W 7.	5% 8		9% 1	or 50	8% 5	
		Ő.		5	8		6	fc	õ	

Carbon	PPy coated on MWCNT-	aď									170
nanofibre	embedded activated	75 F						00			
web (2008)	carbon nanofibers	-2, 2,						ter 5			
	(PPy/ACNF/CNT)	A cm						çe afi			
	using Ni foil as the	1 m/	m^{-2}					hang			
	current collector with 6	-1 at	nA c					ole c			
	M KOH aqueous	ы Б	10 n					gligil les			
	electrolyte	333	¹ at					Neg			
Graphene	Hierarchical graphene			л,				x			171
fibre fabric	fibre fabrics (GFFs) with	ad ⁻	at	un 0				bility	°0	සු	
(2017)	PVA-H ₂ SO ₄ gel	044 I	m^{-2}	f 15	cm^{-2}		m^{-2}	capa 00 cy	0 18	sakir	
	electrolyte	ode 2	mFc	ess o	Wh e		W c	cate 6	ble t	it bre	
		ectro	090	ickne	.5 µ		.3 m	to 5	endal	thou	
		Ē	1,(thi	23		26	dn G	Be	wi	
Carbon	Carbon fabric-aligned							es			172
fabric	carbon nanotube/ MnO ₂ /	/ s ⁻¹						cycl			
(2012)	conducting polymers	/m/						000			
	(CF-ACNT-MnO ₂ -	t 0.1						er 1,0			
	PEDOT) with 1 M	n ⁻² a						afte			
	Na ₂ SO ₄ Composite SC	Fc						loss			
	electrodes	1.3						5%			
Carbon cloth	TiO ₂ NWs grown on		r-								173
(2013)	carbon cloth, hydrogen-		.6 F								
	treated TiO ₂ NWs as core,		(139								
	electrochemically active		m^{-2}						gu		
	MnO ₂ (H-TiO ₂ @MnO ₂ as	el),	nA c				(visti		
	positive) and carbon	ы С	0.5 n				kg	cles	& tv		
	shells (H-TiO ₂ @C as	4/Li	³ at (kW	0 cy	ling		
	negative) electrodes with	(PV_{i})	cm	-	1 ⁻³	-1	³ (45	5,00	benc		
	separator, PVA-LiCl gel	m^{-3}	70 F	1 A §	h cn	kg_	cm_	fter	y in		
	and LiCl aqueous	ΕC) 0 C	at 1.	Mm	Wh	Ň	2% a	deca		
	electrolytes	0.68	GCI	a^{-1}	0.3	(20	0.23	91.2	No (

Carbon cloth (2015)	MnO ₂ as cathode, Ti- Fe ₂ O ₃ @PEDOT as anode	n^{-2}	28.8 F	n^{-2})	at 1				0			174
	with NKK separator and PVA-LiCl gel electrolyte	15 F cr	g^{-1} and	mA cn	h cm ⁻³		m^{-3}		ter 6,00			
		Anode 1.	311.6 F	cm^{-3} at 1	Wm 98.(mA cm ⁻²).44 W c		85.4% af	sycles		
Carbon cloth	SnO ₂ @MO _x	e r	<u> </u>			П	<u> </u>			•		175
(2015)	(SnO ₂ @NiO,	ed th	cm^{-2}						O an			
	SnO ₂ @Co ₃ O ₄ ,	how	mF						@Ni	3.3%		
	SnO ₂ @MnO ₂)	ode s	(980						$nO_2($	in 58		
	heterostructures grown on	ectro	nce						О4, S	ainta		
	carbon cloth (CC) with 1	ire el	acita						C03(n m		
	M Na ₂ SO ₄ aqueous	ructu	l cap						$O_2@$	es ca		
	electrolyte (for SnO ₂ and	crosti	areal						s, Sn	trod		
	SnO ₂ @MnO ₂) and 1 M	hete	arge						ycles	elec		
	KOH aqueous electrolyte	$1nO_2$	isch	cm^{-2}					00 c	$1nO_2$		
	(for SnO ₂ @NiO and	2 @N	est d	mA					r 6,0	2 @N		
	SnO ₂ @Co ₃ O ₄)	SnO	high	at 1					Afte	SnO		
Carbon cloth	Two symmetric								es			176
(2015)	freestanding PANI-	or	۲)	\mathbf{V}	3	n^{-2})	245		cycl			
	cobalt-based MOF	n^{-2} f	7-C	10m	h cm	/h cr	-3 (0,		000			
	crystals (PANI-ZIF-67)	ıF cı	JIF-6	le at	mW	мш	Vcm		er 2			
	electrodes with H ₂ SO ₄ -	46 n	Z-IN	ctroc	161	0044	33 V	m^{-2}	% aft			
	PVA as gel electrolyte	2,1	PA	ele	0.0	(0.0	0.8	Wc	806			
Carbon cloth	Assembly of											177
(2015)	MnO ₂ @carbonized										cles	
	Polypyrole (CPPy) as	cm ⁻²							Se		ß cy	
	positive and carbon	mA							cyclé		nibn	
	coated Co_3O_4 ($Co_3O_4@C$)	t 20			n ⁻³				000 ()0 be	
	microsheet as negative	n ⁻³ a			∕h cn		:m ⁻³		er 5,(er 50	
	electrode with PVA-KOH	F cr			ШW		M c		ó afti		6 aft	
	electrolyte	59.5			27.0		1.31		696		6 7 9	

Carbon cloth	Hierarchical structure of									178
(2015)	ALD Co ₃ O ₄ nanolayer	1 ⁻²				-	cles			
	deposited on CVD	F cn				gu) cy(
	derived carbon cloth	.7 m				fadi	0,000			
	(CNT/CC),	416				icity 	er 50			
	(CNTs@Co ₃ O ₄ /CC) with	hest				cape	n aft			
	2M KOH electrolyte	Hig				No	eve			
Carbon cloth	In-situ electrodeposition	at	2							86
(2016)	of MnO_2 and PPy	a^{-1}	nF cm_			0				
	composite on carbon	25 F	28 n - mA			1,00				
	cloth with 1 M Na ₂ SO ₄	de 3.	$(A g^{-1}, 2)$			after				
	aqueous electrolyte	ctro				, %9	cles			
		Ele	0.2 cm			6~	cyc			
Carbon cloth	SnO ₂ nanoparticles,									179
(2017)	CNTs, ethyl cellulose,	mF								
	and terpineol composite	5.68								
	ink screen-printed onto	; put								
	carbon cloth. Furnace-	lat) â								
	calcined SnO ₂ /CNT	⁻² (f								
	electrodes sandwiched	t cm	ent)							
	with PVA-H ₂ SO ₄ gel	1 mł	- ⁻ (þ							
	electrolyte	5.6	cm							
Carbon cloth	Solvothermal MoO ₂	at			Αt	at				180
(2018)	coating of carbon fibres	cm ⁻²		mA	00 n	cles		SS		
	followed by covered and	mF (t 10	at 1) cyc		time		
	interconnect by rGO film	132 r		n ⁻² a	cm ⁻²),00(4	000		
	to form electrode with	le 8,	÷	'h cn	μW	er 3(cm	er 6,		
	$1 \text{ mol } L^{-1} H_2 SO_4 aqueous$	ctroc	_s ∧	μW	022 2	6 aft	mA	6 aft	ling	
	electrolyte	Ele	2 m	143 cm ⁻	15,(cm ⁻	95%	120	%LL	folc	

Carbon cloth	MnO ₂ /cotton derived						181
(2018)	carbon cloth (CDCC) as			.m ⁻²			
	positive, CDCC as	m^{-2}		IW c			
	negative electrode	ıA cı		15 m		cles	
	attached by nickel foams,).1 n		at 0.		00 cy	
	cotton woven separator	² at (2 m ⁻²		5,00	
	sandwiched with 1 M	cm		Wh c		after	
	Na ₂ SO ₄ aqueous	2 mF		1 m/		7% :	
	electrolyte	202		30.		87.	
Carbon cloth	Hydrothermal						182
(2019)	incorporation of Pt into	le),					
	MoS ₂ nanosheets grown	ctroc	(e)			S	
	on carbon cloth as	(ele	devi			ycle	
	electrode in 1 M Na ₂ SO ₄	∖ g ⁻¹	g_1 ()			00 c	
	solution, Pt-doped MoS ₂	0.5 /	4 A			r 3,0	
	and activated carbon	¹ at (at 0.			afte	
	electrodes with PVA-	н Б	ດ			%9 (
	H ₃ PO ₄ electrolyte	250	42 I			87.9	
Carbon fibre	Activated porous CFT	2					183
textile	(APCFT) electrode as	cm	0	n^{-3}		er s	
(CFT)	anode and TiN@MnO ₂	.2 F	cm	/h cr	m^{-3}	y aft ycle	
(2019)	on CFT as cathode with	de 1	mA	ШW	M c	lecay 00 c	
	PVA-LiCl electrolyte	Ano	at 4	4.70	2.29	No d 25,0	
Carbon	Two, RGO enfolded	-,					184
fabric	cobalt (II, III) oxide	A g'	cm ⁻²	nWł	n ⁻² Vh kj	00	
(2019)	nanowires on flexible	at 1	mF	1041	W cr 68 V	r 2,0	
	carbon fabric substrate	ຜ່	33.4	0.01	2 mV t 24.	afteı	
	(CONW-RGO) electrodes	10 I	g^{-1} , cm ⁻³	or or	or 1. n ⁻³ a	.2%	
	with PVA/KOH gel	e 1,1	35 F mF	h kg	V cn	e 94	
	electrolyte and filter	trod	685	8 M	kW J 3 mV	trod ics	
	paper as a separator	Elec	SCand	34.7 cm ⁻²	3.6 J or 2	Elec cycl	

Carbon cloth	One-step hydrothermal										185
(2020)	method to prepare	$A g^{-1}$					00				
	NiMnO3 nanosheets on a	at 1					$\cdot 1,0$	βY			
	carbon cloth (CC) with	ad					after	at 10			
	6M KOH was the	30 F					8%	les a			
	electrolyte	2,3					67.	cyc			
Carbon cloth	MoS ₂ /PANI composite	1	it				at				98
(2021)	material drop casted on	A g	σ^{-}				cles	_			
	functionalized carbon	0.2.	8 В	(e)) cyd	ode)			
	cloth; MoS ₂ /PANI/FCC	-1 at	, 72.	levic			1,000	lectr			
	electrodes soaked with 1	ы Б С	ode)	g ⁻¹ (6			fter	-1 (e			
	M H ₂ SO ₄ electrolyte	32.25	lectr	2 A §			'% aj	A g			
		45	e)	0.			87	10			10.5
Carbon	Cobalt doped MoS ₂ (Co-				1			S			186
fabric	MoS ₂) nanoflower				_kg'	kg ⁻¹	fter	sycle			
(2022)	electrode with 1 M KOH	ω			lWh	ΧW	% a	000			
	electrolyte	86 F			4.30	0.6]	98.5	10,0			
Multidimens	Graphene nanosheets				n ⁻²						187
ional	(GNS) and PEDOT: PSS	n^{-2}			Wcn		SS		<u>ц</u>		
hirerarchial	deposited on hierarchical	ıVcı			4 m		cycle		eren		
fabric	fabric via spraying	t 1 n			at 0.		9000		diff		
(2019)	method to fabricate	n ⁻² at			cm ⁻²		10,0		nder	tes	
	flexible SCs with	oFcn			ıWhu		after		n ss	g sta	
	PVA/H ₂ SO ₄ electrolyte	5.5 n			82 µ		6%		re lo	guibr	
		245			21.		83.		me	ber	

Appendix E. Other key properties of SCs for wearable applications



E.1 Flexibility

Figure E.1 Flexibility tests of PPy@MnO₂@rGO-deposited conductive yarns measured in the two-electrode cell a) CV curves of the all-solid-state yarn supercapacitor undergoing consecutive deformations at a scan rate of 100 mV/s b) GCD curves of the all-solid-state yarn supercapacitor undergoing consecutive deformations at a current density of 80 mA/cm³ c) Capacitance ratio under various deformations d) Capacitance retention of the all-solid-state yarn supercapacitor after each deformation. Reproduced with permission.¹⁴⁰ Copyright 2015, American Chemical Society. Influence of bending deformation on CeO₂-ACVF capacitive performance; specific capacitance e) under various bending angles; and f) after different bending cycles. ACVF: activated viscose fabric. Reproduced with permission.⁹¹ Copyright 2020, SAGE Publications.

To be considered as wearable, a SC device must be flexible and durable under the physical movement of the body. The bending of SC devices for hundred to several hundred cycles are performed at various angles to evaluate the flexibility of a SC device. Additionally, twisting, winding, and other deformations are also assessed. Yu et al.,³¹ reported more than 97% capacitance retention after 1000 bending cycles at 90° angle of hierarchically structured carbon nanotube-graphene fibre-based micro SCs. Chen et al.¹¹¹ reported an electrochromic fibre-shaped SC composed of elastic fibre/CNTs/PANI retaining capacitance of 93.8% after 1000 bending cycles at 180°. Choi et al.¹¹⁶ reported a flexible SC made of carbon nanotube

yarn with MnO₂ exhibiting no decrease of capacitance even after 1000 bending at 90°. Ding et al.¹²³ fabricated a graphene/PPy composite fibres for all-solid-state, flexible fibre form SC that exhibiting similar performance.

In addition to bending, the flexibility of SCs for other mechanical deformations was reported by some research groups. For example, a varn-based SC was developed by Lee et al.,¹³⁶ composed of Pt/MWCNTs/PEDOT retained 98% of its initial capacitance after 2,000 bending, 92% after 10,000 winding, and 99% after 10,000 cycles when woven into a glove. Huang et al.¹⁴⁰ reported yarn-based SC of rGO/MnO₂/PPy that retained 80% capacitance after 1,000 cycles at 90° bending, 91% after 1,000 cycles knotting, and 103% after 1,000 cycles twisting, revealing the enhancement of the capacitive performance, *Figure* (a-d). Similarly, the capacitance was increased (107% retention) after 1,000 cycles of twisting for an asymmetric fibre-shaped solid-state SC based on carbon fibre bundle.¹³¹ Additionally, several other articles have reported higher bending cycles. For example, Ye et al.¹⁰² reported a fibreshaped SC by introducing rGO and carbon nanoparticles (CNPs) on commercial cotton threads (CT) using dip-coating technique combined with low-temperature vapour reduction, which resulted in 92.30% capacitance retention after 2000 bending cycles. Liu et al.¹⁰¹ examined their 1D-shaped flexible yarn SC composed of rGO/Ni cotton composite electrodes with PVA/LiCl gel as electrolyte and separator for 4000 cycles, and found 95% of retention after at 180° bending angle. Furthermore, Wu et al.¹¹⁸ reported a flexible fibershaped SCs (FSSCs) by twisting a number of CNT yarns (n) with a Pt filament as current collector and PANI nanowires. They obtained a capacitance retention of 98.17 % after 3,000 cycles and 95.91 % after 5,000 flexing cycles.

For 2D-shaped SC devices, shorter bending cycles were used for testing capacitance retention of textiles SCs. Zhang et al.¹⁶⁶ reported a silk fabric-based SC which retained 98.5% after 100 bending cycles and 96.8% after 100 twisting cycles. Lee et al.¹⁷⁷ developed an asymmetric SC by assembling MnO₂@CPPy and carbon coated Co₃O₄ microsheet (Co₃O₄@C)-decorated carbon cloths with a solid-state PVA/KOH electrolyte, which retained 97% capacitance after 500 bending cycles. Luo et al.⁹¹ reported an increase of 13.4% capacitance after 200 bending cycles for an all-fabric solid-state flexible SC, made of activated carbon fibre fabric, *Figure* (e,f). The capacitance of a solid-state stretchable SC, prepared by assembling VPPyNTs/CNOs@PPyG-textile electrodes with a PVA/H₃PO₄ gel electrolyte into a sandwiched structure, was nearly unchanged after stretching for 500 cycles at a strain of 50%.

However, the capacitance retention ratio decreased slightly to 88% as the strain% was increased to 100%.¹⁶⁹

E.2 Safety issue

To be wearable, the flexible SC device components must be non-toxic to avoid any health concern as well as for the environment when disposed of. The concerns with wearable SCs is raised due to the toxic nanoparticles or metal particles of all sizes entering or generated during manufacturing stages as well as during usage. In addition, the effect of electromagnetic fields, accidental electric shock, and the inability to activate the emergency shut-off in case of malfunctioning are also matter of concerns.¹⁸⁸ In addition, wearable SCs and other wearable electronic devices have limited lifetimes. Therefore, it is also critical to ensure that the waste generated by the SCs does not create new hazards for health and the environment.

A report was published revealing that more than 8 billion batteries enter the US and European markets annually. Also 3 billion alkaline units get discarded each year in North America alone.¹⁸⁹ Another report projected a generation of more than 130 g of battery waste per person each year.¹⁹⁰ Besides battery, increasing usage of mobile, computing and other autonomous electrical devices increases the production and disposal of SC devices exponentially.¹⁸⁹ An estimated 20,000 tonnes of old household batteries end up in landfill every year.¹⁹¹ The challenges of such disposal to the environment is due to the presence of a large number of toxic metals (e.g., Cd, Ni, Pb), F-containing electrolytes and device components, corrosive fluids (e.g., H₂SO₄ and H₃PO₄), and fire hazards from organic electrolytes, which may have negative environmental impacts and may induce numerous health problems such as acute or long-term exposure. The principal issue of the release of metals into landfills is the potential to contaminate the groundwater. Incineration of them may also pose two major potential environmental concerns; the release of metals (mostly mercury, cadmium and lead) into the ambient air and the concentration of metals in the ashes which must be landfilled. The Fcontaining electrolyte salts (tetraethylammonium tetrafluoroborate, Net₄BF₄), carbon particle binder (such as PTFE or PVDF), and electrode separator (often PTFE) are likely to generate volatile fluorocarbons during traditional incineration, which are highly toxic to organisms and are likely to damage incinerators and nearby structures. Acetonitrile solvents, commonly implemented in high-performance devices, are flammable, carcinogenic, and may decompose into highly toxic cyanides upon heating. Some ions commonly used in promising ionic liquids, such as bis(trifluoromethanesulfonyl)imide (TFSI), have been shown to inhibit

cellular respiration. Although some aqueous electrolytes that implement Li₂SO₄ or Na₂SO₄ are expected to be benign to the environment, they still emit SO₂, contributing to acid rain when released during incineration. Although SCs, unlike fuel cells and batteries, contain no noble or heavy metals that are particularly difficult to dispose of, conventional collectors and packaging materials, such as steel and Al, are incombustible and cannot be fully burned without leaving ash residue. Thus, the disposal of SCs not only generates harmful substances but also incombustible waste materials that need to be stored in landfill.¹⁹⁰



E.3 Washability

Figure E.2 a) Schematic diagram of washing test. Reproduced with permission.¹⁹² Copyright 2019, The Royal Society of Chemistry. b) Illustration of graphene-based ink pattern and encapsulation layer on textile substrate c) The change in electrical resistance with number of washing cycles of graphene-based ink printed (without encapsulation) and graphene-based ink-printed (with encapsulation) cotton fabric. Reproduced with permission.⁴⁷ Copyright Copyright 2022, Elsevier. d) Resistances of electrode after being immersed in water for different times. The inset is the photograph of electrode immersed in water for 1 week (scale bar: 20 mm) e) Resistances of same electrode on nylon substrate after being immersed in

water for different times and 2 h for each time. Reproduced with permission.¹⁹³ Copyright 2018, American Chemical Society.

Washability is a product's ability to withstand a predetermined number of cycles of a specified washing process, able to adequately clean the product without loss of functionality and/or serviceability and without resulting security risks for the user.¹⁹⁴ Most e-textiles still suffer from the poor wash ability, reducing the reliability of e-textiles to be ready for the market. Many of experimental wearable e-textiles are not suitable for real life applications because of this problem. The hydrophobic textile substrate, due to capillary effect, can still absorb water in the textile bulk making the device fail. Also, the mechanical stresses incorporated by the washing cycles may destroy the electrical contacts between the conductive thread and the electronic wearable device. Thus, the electric impedance becomes uncontrollable after several washing cycles, making the wearable device unstable and often stops functioning.¹⁹⁵ Therefore, there remains a need for the technology that can provide better wash stability for conductive textiles.

Wash ability is usually reported by the retention of performance after several washing cycles. We encapsulated screen-printed graphene-based conductive pattern on textiles to protect it from being washed away, *Figure b*.⁴⁷ The sheet resistance before and after encapsulation of the printed pattern was evaluated. It was found that the bare pattern had an increase of 10 times resistance, whereas the encapsulated pattern exhibited only 3.5 times increase in the sheet resistance after 10 wash cycle, *Figure c*. Cao et al.¹⁹³ also reported a screen-printed washable e-textile electrodes, which were tested after being immersed in water for different times, showed very negligible variation, after soaking in water repeatedly and for longer duration, *Figure* (d,e). A KAIST research team, fabricated a self-powered washable textilebased wearable display module on real textiles that integrate polymer solar cells (PSCs) with organic light emitting diodes (OLEDs), exhibiting little change in characteristics after 10 minutes-long 20 washings cycles.¹⁹² Qiang et al.⁴⁹ demonstrated a super-hydrophobic conducting fabric with graphene and hexagonal boron nitride inks. The different fabrics were then integrated to engineer an all-textile-based capacitive heterostructure that sustained 20 cycles of repeated washing. Barazekhi et al.¹⁶⁴ also reported a negligible decrease in conductivity after 20 laundry cycles for rGO - PPy based polyester textile SC.

Appendix F. Supporting information of Chapter 3. Fully printed and multifunctional graphene-based wearable e-textiles for personalized healthcare applications F1. Characterization of screen-print ink



Figure F1. Thixotropic behaviour of the graphene-based print ink.

F2. Effect of annealing of fabric

Table F1. Comparison of tensile properties of annealed substrate in comparison with the untreated substrate according to EN ISO 13934-1.

Parameters	Un-treated substrate	Annealed (at 170°C)	% Change
Breaking force (N)	613.30	596.18	-2.79
Standard deviation	46.73	52.76	-
Elongation (mm)	34.63	33.43	-3.47
Standard deviation	1.55	1.61	-



Figure F2. Force-elongation curves for untreated and annealed substrate.

F3. Scanning electron microscope (SEM) images



Figure F3. SEM image of graphene-ink printed (1 layer) fabric (×2000).



Figure F4. SEM image of graphene-ink printed (4 layer) and encapsulated fabric (×2000).



Figure F5. SEM image of graphene-ink printed and washed cotton fabric without encapsulation (×1000).

F4. Digital photographs of graphene-based textiles



Figure F6. Digital photograph of a) printed (4 layer), b) encapsulated, c) washed without encapsulation, and d) washed after encapsulation of graphene-ink printed cotton fabric.

F5. Flexibility tests of graphene-based textiles





Figure F7. The cord lengths during bending (concave down) and compression (concave upward) of graphene-ink printed textiles.



Figure F8. The variation in resistance of the bending sensor in backward (bending back) direction.



Figure F9. The variation in resistance of the compression sensor in backward (compression back) direction.



Figure F10. The variation in resistance under 30 outward (printed pattern outside) folding-releasing cycles.



Figure F11. The variation in resistance of the compression sensor (forward direction) while used as new sample versus a bended sample.



Figure F12. The variation in resistance of the compression sensor (backward direction) while used as new sample versus a bended sample.



F5. Digital photographs of activity monitoring

Figure F13. Activity monitoring with the printed sensors a) finger bending b) wrist bending c) elbow bending and d) knee bending.

F6. Schematic of interdigitated supercapacitor pattern



Figure F14. Schematic of graphene-ink printed supercapacitor device.

F6. Comparison of capacitance performance with literature

Table F2. Comparison of the electrochemical performance of the graphene-ink printed

 energy storage textile with others reported in the literature.

SI.	Assembly of energy storage textiles	performance	Energy and power density	Device retention	Flexibility	Ref.
1	Screen printed rGO on cotton	2.5 mF cm ⁻²	-	97 % after	95.6% after	45
	followed by reduction with			10000 cycle	folding 180°	
	PVA-H ₂ SO ₄ solid electrolyte				for 2000	
					cycles	
2	Graphene film with PVA-	2.7 mF cm ⁻²	-	-	-	196
	H ₂ SO ₄ solid electrolyte					
3	Textiles fully printed	426.3 mF cm^{-2}	0.0337	90.8%	86.2%	168
	Ag@PPy@MnO2 on Ag	(cathode)	mWh cm ⁻²	retention	retention after	
	cathode electrode and activated		at 0.38	after 5000	40%	
	carbon on Ag anode electrode		mWcm ⁻²	cycles	stretching	
	with PVA-Na ₂ SO ₄ electrolyte				strain	
4	PPy electrochemically	1117 mF cm ⁻²	0.0658	100% after	98.3% after	108
	deposited on rGO painted	at a current	mWh cm ⁻²	10 000	1000 bending	
	SnCl ₂ modified polyester	density of 1	at 1 mA	cycles	cycles	
	textiles with PVA/H ₂ SO ₄ gel	$mA cm^{-2}$	cm ⁻² and			
	electrolyte		0.5 mW			
			cm^{-2}			
5	Coating of poly-cotton textiles	2.7 mF cm ⁻²	-	98% after 15	98% after	50
	with graphene ink			000 cycles	150 cycles of	
					bending at	
					180°	
6	Kevlar fibres, coated in gold,	Areal	2.7×10-5	-	-	197
	and then grew ZnO nanowires	capacitance	mWh cm ⁻²			
	with PVA- H ₃ PO ₄ electrolyte	2.4 mF cm ⁻²				
7	CNT on Ti wire with PVA-	Areal	0.16×10 ⁻³	80 % after	-	198
	H ₂ SO ₄ solid electrolyte	capacitance	mW h cm ⁻³	1000 cycles		
		1.84 mF cm ⁻³	and 0.01			
			mW cm ⁻³			100
8	SnS/S doped graphene on PET	Areal	-	99% after	-	199
	with PVA/H ₂ SO ₄ solid	capacitance		10000 cycle		

	electrolyte	2.98 mF cm ⁻²				
9	N-Doped rGO on PET with	Areal	0.3 mWh	98% after	-	200
	PVA-H ₃ PO ₄ solid electrolyte	capacitance	cm ⁻³ at 0.2	2000 cycles		
		3.4 mF cm ⁻²	W cm ⁻³			
10	Graphene ink screen printed on		0.28 mWh	95% after	-	This
	cotton textiles with PVA-H ₂ SO ₄	3.2 mFcm ⁻²	cm ⁻² at	10000		work
	gel electrolyte		3 mW cm ⁻² .	cycles		

Appendix G. Supporting Information of Chapter 4. Scalable production of 2D material heterostructure-based wearable textile supercapacitors

G1. Configuration of coating samples

Configurations	G-M-G	G-M-G	G-M-G	G-M-G	G-M-G
Graphene	1-0-0	2-0-0	3-0-0	4-0-0	5-0-0
coating layers	6-0-0	7-0-0	8-0-0	9-0-0	10-0-0
MoS_2	0-1-0	0-2-0	0-3-0	0-4-0	0-5-0
coating layers	0-6-0	0-7-0	0-8-0	0-9-0	0-10-0
	0-1-1	0-2-1	0-3-1	0-4-1	0-5-1
	0-1-2	0-2-2	0-3-2	0-4-2	0-5-2
bi-layers	0-1-3	0-2-3	0-3-3	0-4-3	0-5-3
	0-1-4	0-2-4	0-3-4	0-4-4	0-5-4
	0-1-5	0-2-5	0-3-5	0-4-5	0-5-5
	1-1-1	1-2-1	1-3-1	1-4-1	1-5-1
Graphene-MoS ₂ -	2-1-2	2-2-2	2-3-2	2-4-2	2-5-2
graphene tri-	3-1-3	3-2-3	3-3-3	3-4-3	3-5-3
layers	4-1-4	4-2-4	4-3-4	4-4-4	4-5-4
	5-1-5	5-2-5	5-3-5	5-4-5	5-5-5

Table G1. Graphene (G)- MoS₂ (M) – Graphene (G) layer deposition

G2. Scanning Electron Microscope images of the coated textiles



a. Un-coated cotton textiles (x 1000)



b. MoS₂ 1 coated textiles, M1 (x 1000)



c. MoS₂ 10 coated textiles, M10 (x 1000)

Figure G1. Scanning electron microscope (SEM) image of the a. Un-coated cotton textiles, b. MoS_2 Ilayer coated textiles c. MoS_2 10 layer coated textiles (Scale bar: 40 μ m)

G3. Flexibility of coated textiles



a. The variation in resistance of graphene coated textiles during bending and compression (left) and the variation in resistance of graphene coated, MoS₂-graphene coated and graphene-MoS₂-graphene coated textiles during 10 folding-releasing cycles (right)



b. Graphene coated textiles and fabricated supercapacitor $(1 \text{ cm} \times 1 \text{ cm})$



c. MoS₂ coated textiles and fabricated supercapacitor $(1 \text{ cm} \times 1 \text{ cm})$



d. MoS₂-graphene bi-layer coated textiles and fabricated supercapacitor $(1 \text{ cm} \times 1 \text{ cm})$



e. Graphene-MoS₂-graphene tri-layer coated textiles and fabricated supercapacitor

 $(1\ cm \times 1cm)$

Figure G2. Flexibility of the 2D materials-heterostructure coated textiles and as-fabricated supercapacitors



G4. Electrochemical characterization of graphene-based textile supercapacitors

Figure G3. Graphene-coated textile supercapacitor a. Change of areal capacitance with increase of graphene coating layers b. Cyclic voltammetry curves of the G10 coated textile supercapacitor at various scan rates c. Charge-discharge profile of the G10 coated textile supercapacitor at different current densities d. Electrical impedance spectroscopy of the device at high frequency range (inset shows the response of the supercapacitor device at low frequency range) e. Capacitance retention of the G10 coated textile supercapacitor device up to 1000 cycles, inset shows the CV curves at 1st, 100th, 500th and 1000th cycles and f. cyclic test of the supercapacitor (from 990th to 1000th cycles).





Figure G4. MoS₂-coated textile supercapacitor a. Change of areal capacitance with increase of MoS₂ coating layers b. Cyclic voltammetry curves of the M10 coated textile supercapacitor at various scan rates c. Charge-discharge profile of the M10 coated textile supercapacitor at different current densities d. Electrical impedance spectroscopy of the device at high frequency range (inset shows the response of the supercapacitor device at low frequency range) e. Capacitance retention of the M10 coated textile supercapacitor device up to 1000 cycles, inset shows the CV curves at 1st, 100th, 500th and 1000th cycles and f. cyclic test of the supercapacitor (from 990th to 1000th cycles).

G6. Comparison of capacitance performance of the heterostructure supercapacitors

Table G2. Overview of capacitance comparison of some reported supercapacitors in the literature

Electrode	Electrolyte	Areal capacitance	Energy and Power	Ref
			density	
Graphene	PVA-	3.2 mFcm ⁻²	$0.28 \text{ mWh } \text{cm}^{-2} \text{ and } 3$	201
	H_2SO_4		mWcm ⁻²	
NiO/MoS2/rGO	1 M KCl	7.38 mFcm ⁻² (at 25 mVs ⁻¹)		202
rGO-GO-rGO	0.1 M	2 mFcm ⁻²	$2-5.4 \times 10^{-4}$ Whcm ⁻² and	33
	NaClO ₄		3.6–9×10 ⁻² Wcm ⁻²	
Light scribed GO	PVA-	2.9 mFcm ⁻² (at 50 mVs ⁻¹)		203
	H_2SO_4			
rGO-CNT	3 M KC1	6.1 mFcm ⁻² (at 10 mVs ⁻¹)	0.68 mWhcm^{-3} and 77	204
			Wcm ⁻³	
CNT/MoO3	1 M HCl	4.86 mFcm ⁻² (at 10 mVs ⁻¹)	2.70 μ Whcm ⁻² and 0.53	205
			mWcm ⁻²	
Graphite oxide	1 M	$0.51 \text{ mFcm}^{-2} (\text{at } 20 \text{ mVs}^{-1})$		206
	Na ₂ SO ₄			
MoS ₂ nanoparticles	0.5 M	29 μ Fcm ⁻² (at 0.5 mVs ⁻¹)		207
	H_2SO_4			
Graphene/CNT	1 M	2.16 mFcm ⁻² (at 100 mA	2.42 mWhcm ⁻³ and 115	208
	Na ₂ SO ₄	cm ⁻²)	Wcm ⁻³	
Exfoliated MoS ₂	6 M KOH	2 mFcm ⁻² (at 10 mVs ⁻¹)		209
Graphene/textile	PVA-	80.185 mFcm ⁻² (at 1 mVs ⁻¹)	44.547 μ Whcm ⁻² and	This
	H_2SO_4		581.05 μWcm ⁻²	study
MoS ₂ /textile	PVA-	7.1 mFcm ⁻² (at 1 mVs ⁻¹)	3.944 μ Whcm ⁻² and 3550	This
	H_2SO_4		$\mu W cm^{-2}$	study
MoS ₂ -graphene bi-	PVA-	63.73 mFcm ⁻² (at 1 mVs ⁻¹)	$35.405 ext{ }\mu\text{Whcm}^{-2}$ and	This
layer/textile	H_2SO_4		8497.333 μWcm ⁻²	study
Graphene-MoS ₂ -	PVA-	105.08 mFcm ⁻² (at 1 mVs ⁻¹)	58.377 μ Whcm ⁻² and	This
graphene tri-	H_2SO_4		1604.274 μWcm ⁻²	study
layer/textile				



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